

Hysteresis-Free Like, High Mobility p-Channel Tin Monoxide Thin-Film Transistor

Taikyu Kim¹, Se Eun Kim¹, and Jae Kyeong Jeong¹

jkjeong1@hanyang.ac.kr

¹Department of Electronic Engineering, Hanyang University, Seoul 04763, Republic of Korea

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ABSTRACT

We demonstrate high mobility p-channel tin monoxide (SnO) thin-film transistors with hysteresis free-like behavior. Intermediate alumina encapsulation in the middle of two postdeposition annealing processes significantly facilitates crystal growth, enabling considerable intertwining between crystals. Here, we report a simple method crystallizing the SnO thin-film.

1 Introduction

Since the invention of amorphous indium gallium zinc oxide (a-IGZO) in 2004, amorphous oxide semiconductor (AOS) family, such as IGZO, indium gallium oxide (IGO), and indium gallium tin oxide (IGTO) etc., has been studied due to their fascinating characteristics such as high mobility (μ_{FE}), extremely low off-current, and great uniformity etc. [1,2] Benefiting from these properties, organic light-emitting diodes (OLED) display backplanes which consist of AOS thin-film transistors (TFTs) were practically demonstrated. However, the energetic bombardment which occurs during deposition of indium tin oxide (ITO) thin-film on an organic hole injection layer deteriorates electrical characteristics of inverted OLED stacks. In this respect, it could be a breakthrough to develop high-performance p-type oxide semiconductors. However, valence band (VB) maximum is strongly localized by oxygen 2p orbitals in the oxide system, which hinders the development of high-performance p-type oxide semiconductors to date.

Tin monoxide (SnO) is one of a few p-type oxide semiconductor candidates. [3] It draws significant attention due to its several characteristics such as moderate μ_{FE} and large optical bandgap. For this reason, it has been considerably studied to improve its electrical characteristics with various approaches. [3-5] Nonetheless, μ_{FE} of p-channel SnO TFTs still stays at the level of a few cm^2/Vs , which indicates a necessity to increase the μ_{FE} . Large hysteresis is also a severe hindrance and its origin

in the p-channel SnO TFTs, which should be investigated for its potential applications.

In this study, we report an effect of intermediate encapsulation on p-channel SnO TFTs. Of particular interest is our finding that the encapsulation in the middle of two postdeposition annealing (PDA) prevents a complete oxidation and expedites a significant growth of crystals in the corresponding SnO thin-films. Benefiting from this improved crystallization, the resultant TFTs exhibit high mobility, hysteresis free-like behavior with maximum μ_{FE} of $15.8 \text{ cm}^2/\text{Vs}$ and hysteresis of 0.1 V etc.

2 Experiment

Aluminum oxide (Al_2O_3)/hafnium zirconium oxide (HfO₂) gate dielectric stack was deposited using plasma-enhanced atomic layer deposition (PEALD) process. Then, 20-nm-thick p-type SnO thin-film was deposited on the gate dielectric stack using reactive sputtering. The thin-films were subjected to first PDA (PDA-1) at 100 or 250 °C for 1 hour. Subsequently, ITO thin-films were deposited for source/drain (S/D) electrodes using the sputtering. Note that the patterning was defined using metal shadow masks, and width/length of the patterned channel were $500/100 \text{ }\mu\text{m}$. Furthermore, Al_2O_3 encapsulation layer was deposited on top of the channel layer using the PEALD, followed by second PDA (PDA-2) at 250 °C for 1 hour. Figure 1 shows a schematic of p-channel SnO TFTs and its fabrication procedure.

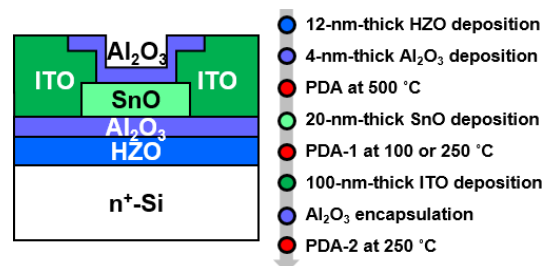


Figure 1. A schematic of p-channel SnO TFT and its fabrication procedure.

3 Results

Crystalline structure of SnO thin-films fabricated through different fabrication procedures were investigated through grazing incidence X-ray diffraction (GIXRD) as shown in Figure 2a. SnO thin-films with PDA-1 at 100 °C show tetragonal Sn phase (JCPDS card No. 04-0673), which could come from the low PDA temperature not enough to effectively oxidize the thin-film. With the temperature of PDA-1 increasing up to 250 °C, the crystalline structure is completely changed to tetragonal SnO phase (JCPDS card No. 06-0395). However, Al₂O₃ encapsulation effectively prevents the complete oxidation of SnO thin-films even after PDA-2 at 250 °C, which leads to the formation of a mixed structure of tetragonal Sn and SnO phases in the encapsulated thin-films with PDA-1 and PDA-2 at 100 and 250 °C, respectively. Note that SnO thin-films with PDA-1 at 250 °C have tetragonal SnO phase regardless of the intermediate encapsulation.

High-resolution transmission electron microscopy (HRTEM) was also conducted to investigate nanoscale differences in the crystalline structure of SnO thin-films (Figure 2b-d). As a result, the encapsulated SnO thin-film with PDA-1 and PDA-2 at 100 and 250 °C, respectively, has the improved surface roughness and crystallinity compared to the other SnO thin-films. Notably, the size of each crystal is significantly enlarged in the corresponding postdeposition condition.

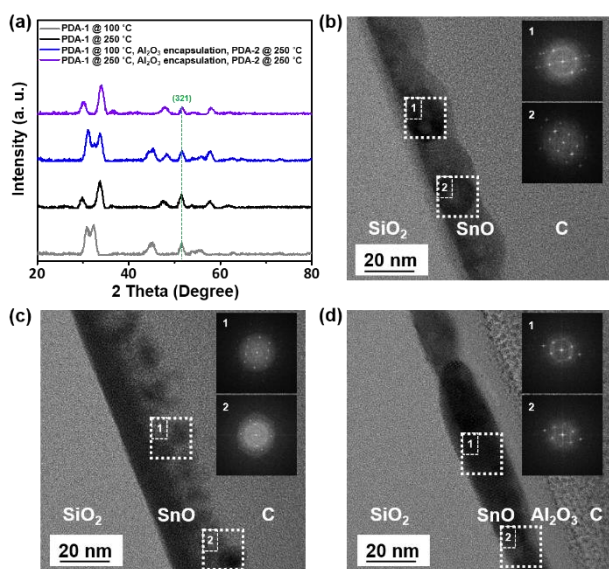


Figure 2. (a) GIXRD patterns of SnO thin-films. (b-d) HRTEM images of SnO thin-films: (b) PDA-1 at 100 °C; (c) PDA-1 at 250 °C; (d) PDA-2 at 250 °C after PDA-1 at

100 °C and subsequent encapsulation.

Then, electrical characteristics of SnO TFTs were examined (Figure 3). SnO TFT with PDA-1 at 100 °C shows the metallic behavior without switching capability. As the temperature of PDA-1 increases to 250 °C, it is changed to semiconducting characteristics with device performances including a current modulation ratio ($I_{ON/OFF}$) of 8.1×10^1 and a μ_{FE} of 1.7 cm²/Vs etc. Surprisingly, the encapsulation considerably improves the device performances of TFT with PDA-1 and PDA-2 at 100 and 250 °C, where an $I_{ON/OFF}$ of 3.4×10^3 , a subthreshold swing (SS) of 0.6 V/dec, a record-breaking μ_{FE} of 14.6 cm²/Vs and a negligible hysteresis of 0.1 V etc. are revealed. This significant enhancement can be attributed to the considerable improvement in the crystal quality. Meanwhile, encapsulated SnO TFT with both PDA-1 and PDA-2 at 250 °C exhibits the conventional device performances with a μ_{FE} of 1.8 cm²/Vs and a negligible hysteresis of 1.0 V etc. Given that the best hysteresis is obtained in the thin-film with the best surface property and crystal quality, bulk defects in SnO thin-films can be a possible origin of the hysteresis.

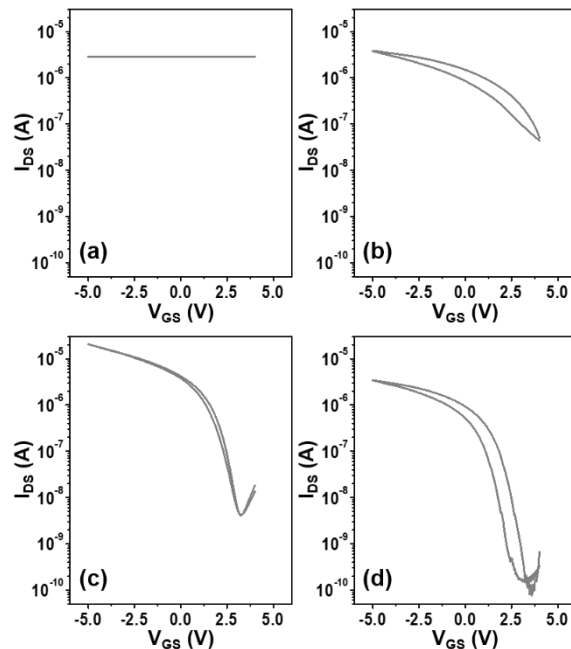


Figure 3. Transfer characteristics of SnO TFTs: (a) PDA-1 at 100 °C; (b) PDA-1 at 250 °C; (c) PDA-2 at 250 °C after PDA-1 at 100 °C and subsequent encapsulation; (d) PDA-2 at 250 °C after PDA-1 at 250 °C and subsequent encapsulation.

4 Conclusions

In this study, we demonstrate the hysteresis free-like, high mobility p-channel SnO TFT using the low temperature PDA-1 and subsequent encapsulation. The significant enhancement in the device performances could be attributed to the improvement in both the crystal quality and surface property by the proposed fabrication method.

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