

## Development of MgB<sub>2</sub> composites by spark plasma sintering as a superconductor and as a novel material for biomedical applications

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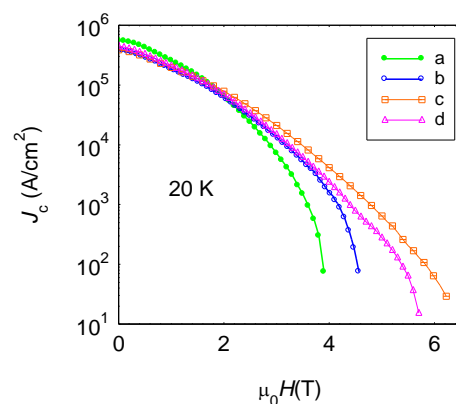
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MgB<sub>2</sub> is prized as a superconducting material. It has advantages over low (LTS) and high temperature (HTS) superconductors: MgB<sub>2</sub> is a light-weight material, it is a simple compound, it is available, it has a low price, and it has the highest critical temperature  $T_c$  for the binary superconductors (39K). The coherence length of MgB<sub>2</sub> is relatively large if compared with the values for HTS. The consequences are that for the enhancement of the critical current density  $J_c$ , epitaxial samples with low angle boundaries and their expensive fabrication technologies as for HTS are not necessary in the case of MgB<sub>2</sub>. Moreover, in MgB<sub>2</sub> grain boundaries are recognized to be transparent to the super-current-flow and they play the role of efficient pinning centers, enhancing  $J_c$ . Grain boundaries can be modified by introducing additions which are not substituting in the crystal lattice of MgB<sub>2</sub> and by processing. Substitutions into the crystal lattice of MgB<sub>2</sub> are also of interest for the improvement of the superconducting parameters.

In our work we use the *ex-situ* spark plasma sintering (SPS). Samples are mixtures of MgB<sub>2</sub> and additive powders. During SPS a uniaxial pressure and a pulsed current are applied on the mold system containing the powder mixture. Although still under debate, between the particles of the powder, unconventional activation processes may occur, accelerating consolidation processes and modifying the grain boundaries. The bulk density of our MgB<sub>2</sub> samples is above 90%. The following additives

classified into 4 groups were tested: (1)- approximately inert such as c-BN, h-BN and graphene, (2)- reactive with formation of  $M_yB_z$  such as RE<sub>2</sub>O<sub>3</sub>, RE = La, Eu, Ho, (3)- reactive with formation of  $Mg_uM_v$  such as Sb, Bi, Te and their oxides, and (4)- additives which are source of carbon substituting for boron in the crystal lattice of MgB<sub>2</sub> such as fullerene (F), F + c-BN, SiC, B<sub>4</sub>C, SiC + Te and Ge<sub>2</sub>H<sub>10</sub>C<sub>6</sub>O<sub>7</sub>. Some additives such as Te, Ge<sub>2</sub>H<sub>10</sub>C<sub>6</sub>O<sub>7</sub> or c-BN significantly increase  $J_c$  and the irreversibility field  $H_{irr}$ , while suppression of  $J_c$  at low magnetic fields is minimized (Fig. 1).

Recently, we have proposed MgB<sub>2</sub> composites added with RE<sub>2</sub>O<sub>3</sub> as a new material for biomedical applications. We paid attention to biodegradation and corrosion aspects as well as to *in-vitro* biocompatibility and antibacterial activity against Escherichia coli, and Staphylococcus aureus. Results suggest that MgB<sub>2</sub>-based materials deserve attention for implants or sterile medical instruments.



**Fig. 1**  $J_c$ - $H$  curves for samples: a-MgB<sub>2</sub>; b-MgB<sub>2</sub>(Te)<sub>0.01</sub>, c-MgB<sub>2</sub>(Ge<sub>2</sub>C<sub>6</sub>H<sub>10</sub>O<sub>7</sub>)<sub>0.0014</sub>; d-MgB<sub>2</sub>(c-BN)<sub>0.01</sub>.