

Perpendicular magnetic anisotropy in as-deposited La/CoFeB/MgO layered structures

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Perpendicular magnetic anisotropy (PMA) is one of the most important issues to develop perpendicular magnetic tunnel junctions (p-MTJs). In the case of CoFeB that is widely used for p-MTJs, interface PMA with MgO plays an important role, since it does not show PMA in a bulk form [1]. Despite the continuous efforts to improve the PMA characteristics in CoFeB/MgO heterostructures, the mechanism of PMA has not fully been understood. First-principles calculations show that the hybridization of Fe-3d_{z²} and O-2p_z states plays a crucial role in the occurrence of the interface PMA [2], where O atoms are positioned just above Fe atoms at the interface. Such an atomic-scale interface structure is likely essential, which is supported by the fact that PMA energies are dramatically increased by post-annealing in most of Fe-based alloy/MgO heterostructures [3-6]. On the other hand, interface PMA can also be observed even for the as-deposited CoFeB/MgO layered structures, depending on the underlayer materials [3,7]. In this study, we investigated PMA in as-deposited La/CoFeB/MgO layered structures, since the PMA in as-deposited layered structures was obtained depending on underlayer materials with a small electronegativity such as Ta or Zr. Note that La also exhibits a typical small electronegativity material that is likely to cause interatomic charge transfer, as well as to be easily oxidized.

CoFeB (t_{CFB})/MgO (2nm) bilayer structures with a 2-nm-thick La underlayer were prepared by rf sputtering on a W-buffered thermally oxidized Si substrates. The whole stacking structures are Si/SiO₂-subs./W-buffer(3nm)/La(2nm)/CoFeB(1nm)/MgO(2nm)/W-cap(1nm) which are grown at room temperature. The thickness of CoFeB layer, t_{CFB} , was varied ranging from 0.8 nm to 1.4 nm. No annealing process was performed in the sample preparation procedures. Magnetic properties were examined by a vibrating sample magnetometer, and x-ray magnetic circular dichroism (XMCD) was measured for the element-specific characterization.

Figure 1 shows the areal PMA energy density $K_{\text{eff}}t_{\text{CFB}}$ as a function of t_{CFB} . Around $t_{\text{CFB}} = 1.0$ nm, definite PMA is obtained for the as-deposited La/CoFeB/MgO layered structures. The PMA energy density reaches $K_{\text{eff}}t_{\text{CFB}} = 0.2 \text{ mJ/m}^2$ ($K_{\text{eff}} = 0.2 \text{ MJ/m}^3$). In the XMCD measurements, it was found that the normal component of the orbital magnetic moment in Fe is slightly larger than the in-plane one, being qualitatively the same as the results in a previous XMCD study for a monocrystalline Fe/MgO layered structure [8]. This suggests that the anisotropy of the orbital magnetic moment brings about the interface PMA even in the as-deposited and therefore presumably amorphous CoFeB layer.

To further investigate the correlation between PMA and the sample structures, La/CoFeB/La and MgO/CoFeB/MgO were prepared. Interestingly, these structures did not show PMA. This means that each interface does not contribute to the occurrence of PMA independently. It is inferred that there is a certain interplay of the roles of under- and over-layers for CoFeB. One possible mechanism would be that the La underlayer works as an excess oxygen absorber to the well-oriented MgO overlayer. However, such a simple idea would not explain the entire results obtained in this study. Internal electric field generated at the La/CoFeB interfaces locally due to the large difference in electronegativity might influence the observed PMA.

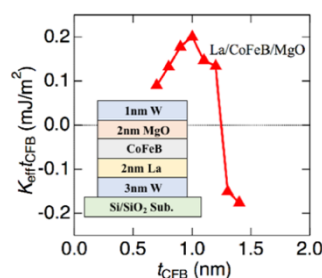


Figure 1, $K_{\text{eff}}t_{\text{CFB}}$ as a function of t_{CFB} in La/CoFeB/MgO layered structures. Inset shows the sample structure.

References:

- [1] S. Ikeda *et al.*, Nat. Mater. **9**, 721 (2010).
- [2] H. X. Yang *et al.*, Phys. Rev. B **84**, 054401 (2011).
- [3] M. Yamanouchi *et al.*, J. Appl. Phys. **109**, 07C712 (2011).
- [4] W. X. Wang *et al.*, Appl. Phys. Lett. **99**, 012502 (2011).
- [5] H. Meng *et al.*, J. Appl. Phys. **110**, 033904 (2011).
- [6] J. W. Koo *et al.*, Appl. Phys. Lett. **103**, 192401 (2013).
- [7] W. Skowronski *et al.*, Phys. Rev. B **91**, 184410 (2015).
- [8] J. Okabayashi *et al.*, Appl. Phys. Lett. **105**, 122408 (2014).