Si 基板上への Si/Gd₂O₃ヘテロ構造の MBE 成長 MBE growth of Si/Gd₂O₃ heterostructure on Si substrates NTT 物性研¹, NTT ナノフォトニクスセンタ² Viviana Fili¹, Wojciech Szuba¹, 徐学俊¹、稲葉智宏¹, ⁰俵毅彦^{1,2}, 尾身博雄^{1,2}, 後藤秀樹¹ NTT Basic Research Labs.¹, NTT Nanophotonics Center² V. Fili¹, W. Szuba¹, X. Xu¹, T. Inaba¹, ^oT. Tawara^{1,2}, H. Omi^{1,2}, H. Gotoh¹ E-mail: takehiko.tawara.tn@hco.ntt.co.jp

Single crystalline Er-doped rare earth oxides (REO) are appealing solid-state materials for the realization of optical quantum memories in the telecommunication band [1,2]. In order to realize these devices, a high-quality rare-earth oxide crystal and a following implementation of a basic waveguide structure with large overlap between light and REO are desirable. Recently we have successfully grown a-Si/Gd₂O₃ horizontal-slot-waveguides on SOI substrates by molecular beam epitaxy (MBE) using molecular oxygen (O₂) as an oxygen precursor [3]. However, some important problems that should be settled were left; those are very slow growth rate of Gd_2O_3 (< 20 nm/h) and uniformity of refractive index of Si cap layer grown on Gd_2O_3 . In this paper, we examine new growth method of Gd_2O_3 with faster growth rate and Si cap with single crystal phase on it.

 Gd_2O_3 and Si cap layers were grown by MBE on Si or SOI (111) substrates. In order to obtain faster growth rate of Gd_2O_3 , we used oxygen radical (O*) with stronger oxidizability in substitution for O_2 molecules. After the optimization of growth condition (growth temperature, flux ratio of Gd/O* and so on), we obtained the Gd_2O_3 layers with 2D growth mode as shown in Fig.1. In this case the growth rate and surface roughness are about 100 nm/h and 0.3 nm, respectively. These results indicate that the Gd_2O_3 growth with O* is effective

both for obtaining higher growth rate and maintaining crystal quality. Single crystalline Si cap layers were also grown on this Gd₂O₃ surface with various growth conditions. However, epitaxial 2D growth of Si on Gd₂O₃ cannot achieved straightforward (Fig.2), because usually the (111) surface energy of rare earth-oxides is about two times lower than those of Si $(\gamma_{Gd2O3} = 500 - 700 \text{ mJ/m}^2, \gamma_{Si} = \sim 1500$ mJ/m^2) [4]. That means Si does not wet Gd₂O₃. Considering this evidence, the a-Si layer is more suitable for the Si cap on Gd₂O₃ rather than the epitaxial Si layer.

The resultant heterostructure of the a- $Si/Gd_2O_3/Si$ must be promising for a following doping with Er ions and for the



Fig.1 (a) Cross-sectional TEM image and (b) surface AFM image of grown Gd_2O_3 with O^* .



Fig.2 Cross-sectional TEM image of $\rm Si/Gd_2O_3$ heterostructures with different growth condition.

realization of an optical waveguide in rare earth doped oxides.

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