## Layer thickness and underlayer dependence of perpendicular magnetic anisotropy in Cu<sub>2</sub>Sb-type (Mn-Cr)AlGe films

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The Cu<sub>2</sub>Sb-type MnAlGe and (Mn-Cr)AlGe are materials showing uniaxial magnetocrystalline anisotropy and relatively small saturation magnetization,  $M_s$ , (~ 300 emu/cm<sup>3</sup>) [1,2], which are attractive for application of the current induced magnetization switching phenomena. Previously we reported magnetic properties of (001)-textured MnAlGe and (Mn-Cr)AlGe films showing perpendicular magnetization fabricated onto thermally oxidized silicon (Si, SiO<sub>2</sub>) substrates for a layer thickness of 100 nm [3]. Although the compatibility with silicon substrates and the relatively large perpendicular magnetic anisotropy energy,  $K_u$  are attractive, the reduction of layer thickness is necessary for application. Thus, in this work, the layer thickness dependence of perpendicular magnetic anisotropy was investigated.

The samples were fabricated using a magnetron sputtering machine. The stacking structure was as follows: Si, SiO<sub>2</sub> sub. | underlayers | MnAlGe or (Mn-Cr)AlGe *t* | MgO 1.5 nm | Ta 5 nm, where *t* is the layer thickness changing from 5 to 30 nm. For the underlayers, "no underlayer" or Ta 5 nm | CoFeBTa 1 nm | MgO 1.5 nm was used, for which SiO<sub>2</sub> of the substrate and the MgO layer acted as templates, respectively. The samples were annealed using a vacuum furnace. The annealing temperature was changed from 300 – 500 °C. The  $K_u$  and magnetic dead layer thickness,  $t_{dead}$ , are summarized for optimum annealing temperatures which are 300 °C for MnAlGe films with the MgO template, and 400 °C for MnAlGe with the SiO<sub>2</sub> template and (Mn-Cr)AlGe films with the MgO templates. The  $t_{dead}$  was 1.7 nm for the MnAlGe films onto the SiO<sub>2</sub> template, while those for the MgO template films were approximately twice larger. From cross-sectional transmission electron microscope images, the oxidization of the (Mn-Cr)AlGe layer was found at the MgO interfaces, which was an origin for the magnetic dead layer. The values of  $K_u$  in 10-nm-thick samples are of the same order as

those reported for the 100-nm-thick film samples in ref. [3], while the  $K_u$  decreased for the 5-nm-thick samples. Details will be discussed in the presentation including other annealing temperatures and as-deposited samples.

- [1] K. Shibata *et al.*, JPSJ **35**, 448 (1973).
- [2] T. Kubota *et al.*, APEX **12**, 103002 (2019).

Table	e I Summary	of $K_{\rm u}$ and	$t_{dead}$ for	r the samp	ples
١	with optimur	n annealii	ng temp	eratures.	

Matarial	Template	t <sub>dead</sub>	$K_{\rm u} (10^6 {\rm erg/cm^3})$		
Material		(nm)	<i>t</i> : 5 nm	<i>t</i> : 10 nm	
Ma AlCa	SiO <sub>2</sub>	1.7	1.1	2.8	
MIIAIGe	MgO	2.9	2.1	3.2	
(Mn-Cr)AlGe		3.5	1.4	5.8	

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