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Epitaxial Growth of $\mathrm{Al}_{\mathrm{x}} \mathrm{Ga}_{1-\mathrm{x}} \mathrm{N}$ by MOVPE

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Wide band gap semiconductor GaN and A1N have attracted much attention as the materials for optical devices in short wave length region. However, very little work on the solid solution $\mathrm{Al} \mathrm{Ga}, \mathrm{N}$ has been reported. This is almost an ideal alloy system because both ${ }^{1-X} G a$ and $A 1$ have nearly the same covalent radius. In this paper, we report the epitaxial films of $A 1 \mathrm{Ga}_{1-\mathrm{x}} \mathrm{N}$ grown on sapphire(0001) and $\operatorname{Si}(111)$ substrates by MOVPE using TMG, TMA ${ }^{1}$ and $\mathrm{NH}_{3}$ as source materials with an ambient $\mathrm{H}_{2}$ gas of normal pressure. These organometallic compounds react with $\mathrm{NH}_{3}$ at room temperature and form complex addition compounds [1], [2], which make this method much complicated. In order to reduce these parasitic reactions, as shown figure 1,organometallic compounds and $\mathrm{NH}_{3}$ were mixed just before the reactor and were fed through the delivery tube to the substrate with the velocity of the gas stream being $110 \mathrm{~cm} / \mathrm{sec}$. This enabled us to control the solid composition of $A 1 \mathrm{Ga}_{1-\mathrm{x}}^{\mathrm{N}} \mathrm{f}_{\mathrm{V}} \mathrm{fairly}$ well. Figure 2 shows a plot of the alloy composition $x$ versus $x^{x}{ }^{1-x} X^{v}$; where $X=[T M A] /([T M G]+[T M A])$ i.e. the ratio of the TMA to total group 111 input. The Al distribution coefficient defined as $x / X^{V}$, was found to be near to unity and was insensitive to the substrate temperature and the kinds of substrates

Figure 3 shows the RHEED patterns of $\mathrm{Al}_{0.1} \mathrm{Ga} .^{\mathrm{N}}$ grown on (0001) sapphire for the azimuth [15 10 ] (fig. 3a) and [10 $\overline{1} 0]\left(f i g .{ }^{1} 3 b\right.$ ): The RHEED patterns showed that single crystal films had been obtained with alloy composition $0 \leq x<0.4$ at substrate temperature $1020^{\circ} \mathrm{C}$ on sapphire and $1050^{\circ} \mathrm{C}$ on Si substrate: the crystals were of wurtzite type as expected and c-axis was aligned normal to the substrate surface.

The lattice constant of $\mathrm{A} 1_{\mathrm{X}}^{\mathrm{Ga}}{ }_{1-\mathrm{x}} \mathrm{N}$ films grown on sapphire, was measured double crystal $X$-ray diffraction ${ }^{x}$ for $^{1-x}$ the (0006) planes. Figure 4 shows the lattice constant $C$ as a function of alloy composition $x$. From the figure, it is clear that $C$ decreases linearly with the alloy composition satisfying Vegard's law, which holds in many $111-\mathrm{V}$ alloys but contradicts the results for samples prepared by MBE [3]. This contradiction will be considered to be concerned with the difference of the growth method.

In conclusion, epitaxial layers of $\mathrm{Al} \mathrm{Fa}_{1-\mathrm{x}} \mathrm{N}$ were grown on sapphire(0001) and $\mathrm{Si}(111)$ substrate by MOVPE. By reducing the parasitic reactions of organometallic compounds with $\mathrm{NH}_{3}$, the alloy composition of $\mathrm{Al} \mathrm{Ga}_{1-\mathrm{N}} \mathrm{N}$ layers could be controlled fairly well. Single crystal films were ob爻aineđ up to $\mathrm{x}=0.4$ at substrate temperature $1020^{\circ} \mathrm{C}$ on sapphire and $1050^{\circ} \mathrm{C}$ on Si substrate. The change of the lattice constant was proportional to the alloy composition.

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Figure 1 Schematic diagram of growth apparatus.

(a) $[1 \overline{2} 10]$

(b) $[10 \overline{1} 0]$
 the azimuth [1210](a) and [1010](b).


Figure 2 The alloy composition of A1 Ga ${ }_{1-\mathrm{N}} \mathrm{layer}$ versus vapor composi-xion $X^{v}$, at different temperature. (o: $1020^{\circ} \mathrm{C}, \Delta: 1120^{\circ} \mathrm{C}$ on sapphire, $\bullet$ : $1050^{\circ} \mathrm{C}$ on Si )


Figure 4 Change of the lattice constant $C$ with the alloy composition x of $\mathrm{Al} \mathrm{Xa}_{1-\mathrm{x}} \mathrm{N}$ films grown on sapphire
at $1120^{\circ} \mathrm{C}$.

