Epitaxial Growth of $Al_xGa_{1-x}N$ by MOVPE

Y.Koide, H.Itoh, N.Sawaki, and I.Akasaki

Department of Electronics, School of Engineering, Nagoya University, Furocho, Chikusa-ku, Nagoya 464

M. Hashimoto

Toyota Central R&D Laboratories Inc., Nagakute-cho, Aichi 480-11

Wide band gap semiconductor GaN and AlN have attracted much attention as the materials for optical devices in short wave length region. However, very little work on the solid solution Al Ga $_{\rm l}$ N has been reported. This is almost an ideal alloy system because both Ga and Al have nearly the same covalent radius. In this paper,we report the epitaxial films of Al Ga $_{\rm l}$ N grown on sapphire(0001) and Si(111) substrates by MOVPE using TMG, TMA and NH $_{\rm l}$ as source materials with an ambient H $_{\rm l}$ gas of normal pressure. These organometallic compounds react with NH $_{\rm l}$ at room temperature and form complex addition compounds [1],[2],which make this method much complicated. In order to reduce these parasitic reactions, as shown figure 1,organometallic compounds and NH $_{\rm l}$ were mixed just before the reactor and were fed through the delivery tube to the substrate with the velocity of the gas stream being 110cm/sec. This enabled us to control the solid composition of Al Ga $_{\rm l}$ N fairly well. Figure 2 shows a plot of the alloy composition x versus $^{\rm X}$ 1-x $^{\rm X}$; where X'=[TMA]/([TMG]+[TMA]) i.e. the ratio of the TMA to total group 111 input. The Al distribution coefficient defined as x/X', was found to be near to unity and was insensitive to the substrate temperature and the kinds of substrates

Figure 3 shows the RHEED patterns of Al_{0.1}Ga_{0.2}9 N grown on (0001)sapphire for the azimuth [$1\overline{2}10$](fig.3a) and [$10\overline{1}0$](fig. 3b): The RHEED patterns showed that single crystal films had been obtained with alloy composition $0 \le x \le 0.4$ at substrate temperature 1020 °C on sapphire and 1050 °C on Si substrate: the crystals were of wurtzite type as expected and c-axis was aligned normal to the substrate surface.

The lattice constant of Al $_{\rm X}^{\rm Ga}$ N films grown on sapphire, was measured double crystal X-ray diffraction for the (0006) planes. Figure 4 shows the lattice constant C as a function of alloy composition x. From the figure, it is clear that C decreases linearly with the alloy composition satisfying Vegard's law, which holds in many lll-V alloys but contradicts the results for samples prepared by MBE [3]. This contradiction will be considered to be concerned with the difference of the growth method.

In conclusion, epitaxial layers of Al Ga_1 N were grown on sapphire (0001) and Si(111) substrate by MOVPE. By reducing the parasitic reactions of organometallic compounds with NH3, the alloy composition of Al Ga_1 N layers could be controlled fairly well. Single crystal films were obtained up to x=0.4 at substrate temperature 1020 °C on sapphire and 1050 °C on Si substrate. The change of the lattice constant was proportional to the alloy composition.

The authors wish to thank Dr.Y.Toyoda of Matsushita Electric Ind.Co. for his help in the measuring of the lattice constant using double crystal X-ray diffractmetry and for his valuable discussions.

reference

- [1] H.M.Manasevit, F.M. Erdamann, and W.I. Simpsons
 - J.Electrochem.Soc. 118(1971)1864
- [2] M.Morita, N. Uesugi, S. Isogai, K. Tsubouchi, and N. Mikoshiba Japan. J. Appl. Phys. 20(1981)17
- [3] S.Yoshida, S, Misawa, and S, Gonda J. Appl. Phys. 53(1982)6844

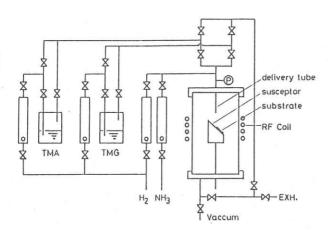


Figure 1 Schematic diagram of growth apparatus.

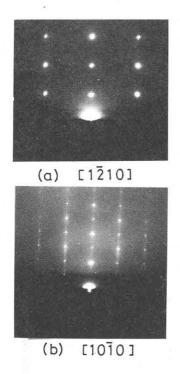


Figure 3 RHEED patterns of Al $_0$.1 Ga $_0$.9 N film on (0001)sapphire substrate for the azimuth [1 $\overline{2}$ 10](a) and [10 $\overline{1}$ 0](b).

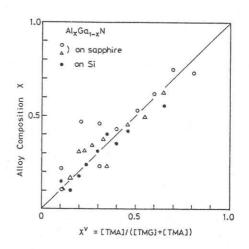


Figure 2 The alloy composition of Al Ga_N layer versus vapor composition XV, at different temperature. (o:1020°C, A:1120°C on sapphire, •: 1050°C on Si)

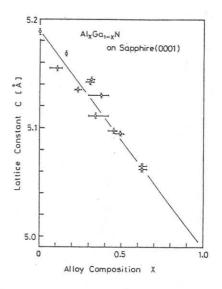


Figure 4 Change of the lattice constant C with the alloy composition x of Al $_{\rm Ga}^{\rm Ga}$ N films grown on sapphire at 1120 $_{\rm C}^{\rm SC}$ 1-x