# Critical Thickness of Heteroepitaxial $Si_{1-x}Ge_x/Si$ Layers as Studied by Atomic Force Microscopy

Toshiyuki NAKAMURA, Morifumi OHNO, Kinya ASHIKAGA and Seigo OHNO

Semiconductor Tech. Lab., Oki Electric Industry Co., Ltd.

550-5 Higashiasakawa-cho, Hachioji-shi, Tokyo 193, Japan

A detailed study has been made on the evaluation of the surface of heteroepitaxial Si1-xGex/Si layers by atomic force microscopy (AFM). It has been found that misfit dislocations in the heteroepitaxial layers can be observed by AFM. On the basis of these results, a new non-destructive method of evaluating the critical thickness of strained heteroepitaxial layer will be proposed in this paper.

#### 1. Introduction

The heteroepitaxial structure of strained Si1-xGex layers on Si substrates attracts attention in recent years since they allow the fabrication of heterobipolar transistors (HBT) compatible with silicon processing technologies. To form HBT, the pseudomorphic growth films which are free from dislocation are required. In the above application, the determination of critical thickness is indispensable and transmission electron microscopy (TEM)<sup>1)</sup> or electron beam induced current (EBIC) technique<sup>2)</sup> is usually used for this purpose. Although TEM has an ability to observe a single misfit dislocation in high resolution, its measurement area is very small. On the contrary, the measurement area of EBIC is large, but the resolution is much lower than that of TEM. In addition, in both methods, specimen must be prepared in a destructive manner for measurements. We have, in detail, investigated the surface of heteroepitaxial Si1-xGex layers by AFM. On the basis of this study, we will propose a new non-destructive and preparation-free method of evaluating the critical thickness of strained heteroepitaxial layers.

### 2. Experimental

Heteroepitaxial Si1-xGex/Si films were prepared on Si wafers with (100) orientation by rapid thermal chemical vapor deposition (RTCVD) as shown in Fig.1. Si wafers were treated with H2SO4 + H2O2. After that, they were cleaned with diluted HF acid, and then put into the vacuum chamber. Si1-xGex films were grown on them from a mixture of SiH4 (10 %) and GeH4 (1%) diluted in H2 at 850 °C. Just before Si1-xGex layer deposition, the substrate surface was pretreated, in the reactor, with SiH4 (10 %) in H2 at 900 °C for a few seconds<sup>3)</sup>.



The film thickness was evaluated from step height measurements. Film compositions were determined by high resolution X-ray diffraction (HXRD) by use of non-relaxed Si1-xGex films. The composition x was calculated on the Vegard rule under the condition that the lattice constant was converted into bulk's one using Poisson's ratio  $\nu = 0.28$ . The networks of misfit dislocations at the interface of Si1-xGex/Si were investigated by TEM. The surface of Si1-xGex film was evaluated on a low tension cantilever type (Nanoscope II (Digital Inst.)) atomic force microscope.

### 3. Results and Discussion

Figure 2 shows a cross sectional TEM micrograph of heteroepitaxial Si0.89Ge0.11/Si film. In Fig.2, epitaxial Si layer formed during SiH4 pretreatment and Si0.89Ge0.11 layer are clearly seen. The thicknesses of Si0.89Ge0.11 and epitaxial Si layers were 250 nm and 50 nm, respectively. No defects were observed in both layers. The crystallinity of films was evaluated by Rutherford backscattering spectroscopy (RBS). The minimum RBS yield of the Si0.89Ge0.11 layer was found to be 3.7 %, which is comparable with that of the Si wafer.

Figure 3 shows plane-view TEM micrographs of heteroepitaxial Si0.89Ge0.11/Si films whose thicknesses are (a)250, (b)140 and (c)100 nm, respectively. In Figs.3(a) and 3(b), the networks of misfit dislocations at interfaces



Fig.2 Cross-sectional TEM micrograph of Si0.89Ge0.11/Si film.

are observed in the <110> direction, but not in Fig.3(c). These findings indicate that the thicknesses of both films in Figs.3(a) and 3(b) are above the critical value of pseudomorphic growth and that of the film in Fig.3(c) is below the critical value, that is, indicate that the critical thickness of this film is between 100 and 140 nm.

Figure 4 shows AFM images of heteroepitaxial Si0.89Ge0.11/Si films. Crosshatched patterns parallel to the <110> direction are observed in Figs.4(a) and 4(b), but not in Fig.4(c). In spite of observations on the



Fig.3 Plane-view TEM micrographs of Sio.89Geo.11/Si films. (a) 250nm, (b) 140nm, (c) 100nm

surface, these crosshatched pattern images obtained by AFM are close resemblance to the network of misfit dislocations observed by TEM (Fig.3). The periods of the crosshatched AFM patterns in Fig.4 are coincident with those of the networks of misfit dislocations shown in Fig.3. These results indicate that AFM observation makes it possible to evaluate misfit dislocations in Si0.89Ge0.11 films in a non-destructive manner.

We tried to determine the critical thickness of strained Si1-xGex/Si heteroepitaxial films by applying this phenomenon. Figure 5 shows the critical thickness determined by the present method as a function of lattice mismatch.







mismatch.

As Fig.5 shows, our data ( $\bullet$  and  $\bigcirc$ ) are in reasonable agreement with those reported previously, where cross symbols show previous experimental data<sup>1),4)</sup> and dotted and dot-dash lines are theoretical ones<sup>4)</sup>.

## 4. Conclusion

We have found that crosshatched patterns are observed, under particular conditions, in the of Si1-xGex/Si measurement images AFM and that these heteroepitaxial layers patterns correspond to the crosshatched networks of misfit dislocations in Si1-xGex/Si layers. We tried successfully to develop a new method of evaluating the critical thickness of strained Si1-xGex/Si heteroepitaxial layers by applying the above results. This method makes it possible to evaluate the density of misfit dislocation and the critical thickness of strained in Si1-xGex/Si heteroepitaxial films а non-destructive manner.

### References

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