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# The Initial Growth Mechanism of LPCVD-Si<sub>3</sub>N<sub>4</sub> on Si and SiO<sub>2</sub>

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Ultra thin (less than 5nm)  $\text{Si}_{3}\text{N}_{4}$  film is one of the key material for 256Mb DRAM capacitor. In order to use ultra thin film, it is significant to coninitial growth of the film. In this study, we make clear trol the the initial growth mechanism of LPCVD-Si $_3N_4$  on SiO $_2$  and Si. On SiO $_2$ , Si $_3N_4$  nuclei are generated and then Si $_3N_4$  film grows. On Si, on the other hand, Si $_3N_4$  film grows without nuclei generation. The nuclei generation rate is limited by gas pressure, and the film growth rate is limited by surface reaction rate.

#### 1.Introduction

The composite  $SiO_2/Si_3N_4$  film, formed by oxidation of low pressure chemical vapor deposition(LPCVD)  $Si_3N_4$  film, has been widely used as the storage capacitor dielectric for dynamic random access memory (DRAM). With increase in the integration of DRAM, it becomes impossible to obtain sufficient storage capacitance using conventional capacitor cell and Si<sub>3</sub>N<sub>4</sub> film. To overcome this problem, advanced cell structure which increase effective surface area of storage electrode has been studied[1].

And as for Si<sub>3</sub>N<sub>4</sub> deposition process, rapid thermal nitridation and loadlock LPCVD deposition process have been proposed[2][3] [4][5].

To use ultra thin film, it is essential to control the initial growth of the film. The great dependence of the initial Si<sub>3</sub>N<sub>4</sub> film growth on substrate surface was demonstrated [6]; LPCVD-Si<sub>3</sub>N<sub>4</sub> film grows initially on Si surface selectively, it does not on Si02\_surface.

In this paper, the initial growth mechanism of LPCVD-Si $_3N_4$  is proposed, which can explain the effect of gas pressure and substrate material on initial  $Si_3N_4$  growth. And atomic force microscopy(AFM) image of initial  $Si_3N_4$  growth on  $SiO_2$  are shown, which also certifies the growth mechanism.

#### 2.Experimental result

Table.1 shows the condition of LPCVD- $Si_3N_4$  growth which was studied in this paper. Usually the growth of Si<sub>3</sub>N<sub>4</sub> film has incuba-

tion time before linear film growth[1]. We deposited  $Si_3N_4$  film on  $SiO_2$  in various pressure, and measured  $Si_3N_4$  film thickness by ellipsometer. As shown in Fig.1, we found that 1) incubation time depends on the total gas pressure, 2)linear growth rate is independent of the gas pressure.

In order to consider the above phenomena, we prepared two type substrate; Si substrate which has  $SiO_2$  line pattern( $SiO_2/Si$  substrate), and  $SiO_2$  substrate which has poly-Si line pattern( $Si/SiO_2$  substrate). The sharp edge of substrate pattern is more exposed than the flat surface of substrate pattern to the reactive gas. So when the gas pressure limits the  $Si_{3}N_{4}$  growth rate, the  $Si_{3}N_{4}$  film is expected to grow thicker on the sharp edge of substrate pattern.

We deposited  $Si_3N_4$  film on them in the pressure of 0.10Torr, for 8 minutes and for 24 minutes. The  $Si_3N_4$  initial growth on Si and SiO<sub>2</sub>, and on flat surface and sharp edge, was observed by transmission electron microscopy(TEM).

Fig.2(a) shows the TEM cross sectional view of Si<sub>3</sub>N<sub>4</sub> film grown on SiO<sub>2</sub>/Si substrate for 8 minutes. The selective LPCVD-Si3N4 growth is observed. Although 7nm thick Si<sub>3</sub>N<sub>4</sub> film grows on Si, Si<sub>3</sub>N<sub>4</sub> film does not grow on Si02;

<sup>2</sup>Fig.2(b) shows  $Si_3N_4$  film grown for 24 minutes. The selectivity is collapsed.  $Si_3N_4$ grows as thick as 19nm on Si and 12nm on SiO<sub>2</sub>. These results are already reported[6].

The results, which we want to discuss in this work, are the followings.

The roughness of Si<sub>3</sub>N<sub>4</sub> film surface is

quite different between on Si and SiO<sub>2</sub> flat surface. Si<sub>3</sub>N<sub>4</sub> film on Si flat surface has smooth surface. On the other hand, Si<sub>3</sub>N<sub>4</sub> film on SiO<sub>2</sub> flat surface has rough surface.

And Fig.3 shows  $Si_3N_4$  film grown on  $Si/SiO_2$  substrate for 8 minutes. (The selective growth on Si is also observed.) Comparing with Fig.2(b), it is recognized that  $Si_3N_4$  growth is quite different between on Si sharp edge and  $SiO_2$  sharp edge.  $Si_3N_4$  film grows thicker on  $SiO_2$  sharp edge than on  $SiO_2$  flat surface. On the contrary,  $Si_3N_4$  film grows thinner on Si sharp edge than on Si flat surface.

### 3.Discussion

From the TEM observation of  $Si_3N_4$  growth on substrate pattern, we propose the following model as shown in Fig.4. In the stage of  $Si_3N_4$  initial growth on  $SiO_2$ ,  $Si_3N_4$  nuclei are generated, and then  $Si_3N_4$  nuclei grow film. The nuclei generation step corresponds to the incubation time. Besides the generation rate of  $Si_3N_4$  nuclei is supposed to be limited by gas pressure.

So  $Si_3N_4$  film on  $SiO_2$  has rough surface, and  $Si_3N_4$  grows thicker on  $SiO_2$  sharp edge. And the incubation time becomes longer when the gas pressure becomes lower.

On Si, on the other hand,  $Si_3N_4$  film grows directly on Si surface from the first layer without nuclei generation. Besides the linear growth rate of  $Si_3N_4$  film thickness is supposed to be not limited by gas pressure, but by the surface reaction rate on Si or  $Si_3N_4$  surface.

 $Si_3N_4$  surface. So  $Si_3N_4$  film on Si has smooth surface, and grows thinner on sharp edge. And the incubation time becomes zero, which leads the selective growth of  $Si_3N_4$  on Si [6]. This model can also explain the study

This model can also explain the study about oxigen resistance ability of ultra thin  $Si_3N_4$  film[1][6];  $H_20$  or  $0_2$  cannot penetrate bulk  $Si_3N_4$ . But  $Si_3N_4$  film on  $Si0_2$ has no oxigen resistance before a critical  $Si_3N_4$  deposition time.  $Si_3N_4$  film on Si has oxigen resistance from the film thickness is less than 1nm. Ultra thin  $Si_3N_4$  film grown on  $Si0_2$  is cosidered to have pin holes, through which oxigen can pass.

This model is confirmed by atomic force microscopy(AFM) observation, in addition. Fig.5 shows AFM image of  $Si_3N_4$  grown on  $SiO_2$ for 8 minute. Although  $Si_3N_4$  film was not observed by TEM, something like nuclei was observed on  $SiO_2$  surface. The left image is the magnification of the right image. By X-ray probe spectroscopy(XPS), nitrogen is detected on this  $SiO_2$  surface, which suggests that the nuclei observed by AFM are  $Si_3N_4$ growth nuclei. 4.Conclusion

From the above studies, the following are concluded;

1)On SiO<sub>2</sub>, Si<sub>3</sub>N<sub>4</sub> growth nuclei are initially generated, and then Si<sub>3</sub>N<sub>4</sub> film are formed. 2)On Si, Si<sub>3</sub>N<sub>4</sub> film grows from the first layer without nuclei generation.

3) The nuclei generation rate on SiO<sub>2</sub> is limited by gas pressure. And the film growth rate is limited by surface reaction rate.

In conclusion, this study has clarified the initial growth mechanism of  $Si_3N_4$  film. These results are consistent with theoretical prediction[7]. To obtain ultra thin  $Si_3N_4$ film which has less surface roughness and high oxigen resistance, LPCVD-Si\_3N\_4 should be deposited on Si surface after the complete removal of SiO<sub>2</sub>.

## 5.References

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Table.1 The condition of LPCVD-Si₃N₄ film growth.



Fig.2(a) A TEM cross-sectional photograph of Si<sub>3</sub>N₄ growth on SiO₂ /Si substrate (8 minutes).



Fig.3 A TEM cross-sectional photograph of Si<sub>3</sub>N<sub>4</sub> growth on Si/SiO<sub>2</sub> substrate (8 minutes).



Fig.4 The initial growth mechanism of LPCVD-Si₃N₄ on Si or SiO₂.



Si3N4 DEPOSITION TIME (min.) Fig.1 The dependence of the thickness on deposition time as a parameter of gas pressure.



Fig.2(b) (24 minutes)



Fig.5 AFM images of LPCVD-Si₃N₄ growth on SiO₂ (8 minutes).