

Invited

InGaN/AlGaIn Double-Heterostructure Light-Emitting Diodes

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High-brightness InGaN/AlGaIn double-heterostructure (DH) blue-green-light-emitting diodes (LEDs) with a luminous intensity of 2 cd were fabricated by increasing an indium mole fraction of the InGaN active layer. The external quantum efficiency was as high as 2.4 % at a forward current of 20 mA. The peak wavelength and the full width at half-maximum (FWHM) of the electroluminescence (EL) were 500 nm and 80 nm, respectively. Also, high-power InGaN/AlGaIn DH violet LEDs were fabricated. The output power was as high as 1.5 mW. The peak wavelength and the FWHM of the EL were 385 nm and 10 nm, respectively.

1. Introduction

Much research has been done on high-brightness blue-light-emitting diodes (LEDs) for use in full-color displays, full-color indicators and light sources for lamps with the characteristics of high efficiency, high reliability and high speed. For these purposes, II-VI materials such as ZnSe,¹⁾ SiC²⁾ and III-V nitride semiconductors such as GaN³⁾ have been investigated intensively for a long time. However, it was impossible to obtain high-brightness blue LEDs with brightness over 1 cd. Recently, the present authors succeeded in producing for the first time 1-cd-brightness blue InGaN/AlGaIn LEDs suitable for commercial applications.^{4,5)} The characteristics of those LEDs were peak wavelength of 450 nm, forward voltage of 3.6 V and output power of 1.2 mW at 20 mA. From the standpoint of application to traffic lights, the color of blue InGaN/AlGaIn LEDs is too bluish and inadequate. Also, the color of commercially available GaP green LEDs with a peak wavelength of 560 nm is too greenish and the brightness of the GaP LEDs is too low (about 100 mcd) for application to outdoor traffic lights. Therefore, there are no available blue-green LEDs with the peak wavelength between 450 nm and 550 nm.

Recent research on III-V nitrides has paved the way for realization of high-quality crystals of AlGaIn and InGaIn, and p-type conduction in AlGaIn.⁶⁻¹²⁾ Also, the hole-compensation mechanism of p-type AlGaIn has been elucidated.^{13,14)} High-brightness blue LEDs with a luminous intensity of 1 cd have been achieved by using these techniques and are now commercially available.^{4,5)} In order to obtain blue emission centers in these InGaN/AlGaIn double-heterostructure (DH) LEDs, Zn doping into the InGaIn

active layer was performed. This impurity-assisted recombination mechanism has not been elucidated. In this paper, co-doping with both Zn and Si into the InGaIn active layer in InGaN/AlGaIn DH LEDs in order to increase the output power of LEDs is described. Also, in order to achieve longer-wavelength (500nm) emission for the application of traffic lights, indium mole fraction of the InGaIn active layer was increased.

2. Experimental

InGaIn films were grown by the two-flow MOCVD method. Details of the two-flow MOCVD are described in other papers.^{15,16)} The growth was conducted at atmospheric pressure. Sapphire with (0001) orientation (C face), which had a two-inch diameter, was used as a substrate. The structure and growth conditions of LEDs are described in detail in other papers.^{4,5)} Only growth conditions of the InGaIn active layer were changed from previous reports. The temperature of the InGaIn active layer was decreased to 780°C to increase the indium mole fraction of InGaIn to 0.23. During InGaIn growth, both Si and Zn were co-doped with a flow of monosilane (SiH₄) and diethylzinc (DEZ). Fabrication of LED chips was accomplished as mentioned in previous reports.^{4,5)} The characteristics of LEDs were measured under DC-biased conditions at room temperature.

3. Results and Discussions

Figure 1 shows the electroluminescence (EL) spectra of the InGaIn/AlGaIn DH LEDs at forward currents of 0.5 mA, 1 mA and 20 mA. The carrier concentration of the InGaIn active layer in this LED was $2 \times 10^{19} \text{ cm}^{-3}$. A typical peak wavelength and full width

at half-maximum (FWHM) of the EL were 500 nm and 80 nm, respectively, at 20 mA. The peak wavelength varies to a shorter wavelength with increasing forward current. The peak wavelength is 537 nm at 0.5 mA, 525 nm at 1 mA and 500 nm at 20 mA. Figure 2 shows the peak wavelength of the EL spectra as a function of the forward current. When the forward current increases, the peak wavelength becomes shorter. This blue shift of EL spectra with increasing forward current suggests that the luminescence mechanism is the DA pair recombination in the InGa_N/AlGa_N active layer co-doped with both Si and Zn. At 20 mA, a more narrow, high-energy peak emerges around 425 nm in Fig. 1. This peak is due to band-to-band recombination in the InGa_N active layer. This peak becomes resolved at injection levels where the impurity-related recombination is saturated.

The output power of the InGa_N/AlGa_N DH LEDs is 0.6 mW at 10 mA, 1.2 mW at 20 mA and 2.2 mW at 40 mA. The external quantum efficiency is 2.4 % at 20 mA. The typical on-axis luminous intensity of InGa_N/AlGa_N LEDs with 15° cone viewing angle is 2 cd at 20 mA. This luminous intensity is the highest value ever reported for blue-green LEDs. Also, this luminous intensity is so bright that these blue-green InGa_N/AlGa_N LEDs can be used for outdoor applications, such as traffic lights and displays requiring high brightness. The forward voltage was 3.5 V at 20 mA.

Figure 3 shows a chromaticity diagram where blue InGa_N/AlGa_N LEDs and blue-green InGa_N/AlGa_N LEDs are shown. Also, the commercially available green GaP LEDs and red GaAlAs LEDs are shown. From this figure, only blue-green InGa_N/AlGa_N LEDs are within the regions of roadway signals and railway signals. Therefore, InGa_N/AlGa_N LEDs are suitable for those applications from the viewpoint of color.

Figure 4 shows luminous intensity as a function of the peak wavelength of various commercially available LEDs. Judging from this figure, there are no LED materials except for InGa_N that have strong luminous intensity over 1 cd below the peak wavelength of 550 nm. Therefore, InGa_N is one of the most promising materials for LEDs and laser diodes (LDs) of peak wavelengths between 550 nm and 360 nm.

Figure 5 shows the EL spectrum of the InGa_N/AlGa_N DH violet LEDs at a forward current of 20 mA. These violet LEDs were grown under the same conditions as blue-green LEDs, except for the InGa_N active layer. During InGa_N growth, only Si was doped with a flow of SiH₄. The typical output power was 1.5 mW and the external quantum efficiency was as high as 2.3 % at a forward current of 20 mA at room temperature. The peak wavelength and the FWHM of the EL were 385 nm and 10 nm, respectively. Therefore, we can also fabricate high-power violet LEDs using III-V nitride materials.

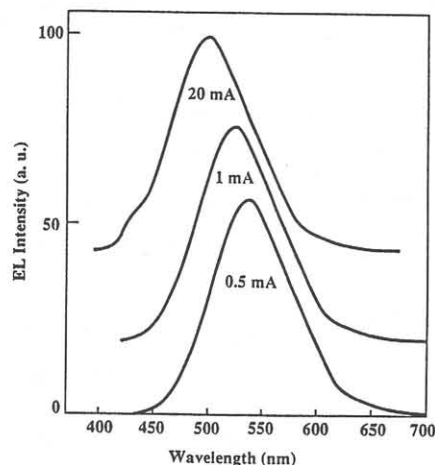


Fig. 1. EL spectra of the InGa_N/AlGa_N DH blue-green LEDs under different forward currents.

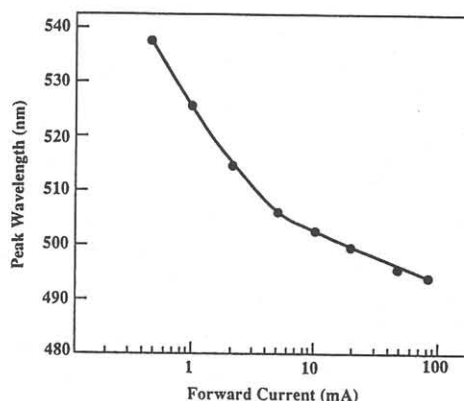


Fig. 2. The peak wavelength of the EL spectra of the InGa_N/AlGa_N DH blue-green LEDs as a function of the forward current.

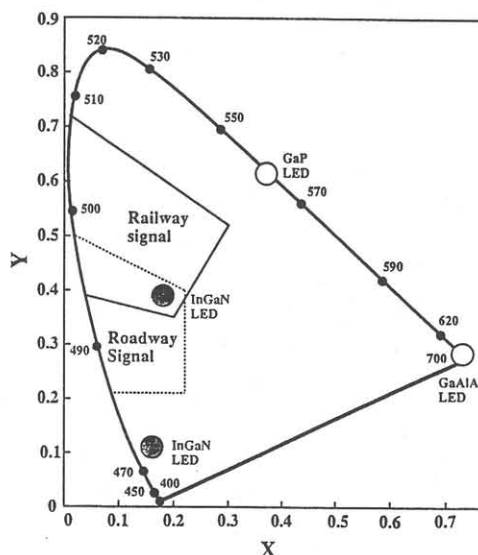


Fig. 3. Chromaticity diagram in which blue InGa_N/AlGa_N LEDs, blue-green InGa_N/AlGa_N LEDs, green GaP LEDs and red GaAlAs LEDs are shown.

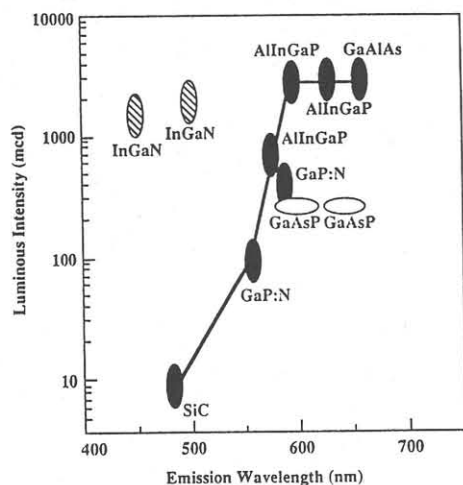


Fig. 4. Luminous intensity as a function of the peak wavelength of various commercially available LEDs.

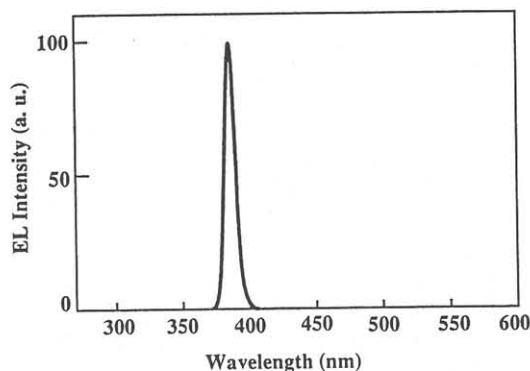


Fig. 5. EL spectrum of the InGaN/AlGaP DH violet LEDs under a forward current of 20 mA.

4. Summary

In summary, 2-cd high-brightness InGaN/AlGaP DH blue-green LEDs were fabricated for the first time. The output power was 1.2 mW and the external quantum efficiency was as high as 2.4 % at a forward current of 20 mA at room temperature. The peak wavelength and the FWHM of the EL were 500 nm and 80 nm, respectively. Traffic lights may prove to be a fertile application for the blue-green LEDs.

References

- 1) W. Xie, D. C. Grillo, R. L. Gunshor, M. Kobayashi, H. Jeon, J. Ding, A. V. Nurmikko, G. C. Hua and N. Otsuka: *Appl. Phys. Lett.* 60 (1992) 1999.
- 2) K. Koga and T. Yamaguchi: *Prog. Crystal Growth and Charact.* 23 (1991) 127.
- 3) J. I. Pankove, E. A. Miller and J. E. Berkeyheiser: *RCA Review* 32 (1971) 383.
- 4) S. Nakamura: *Nikkei Electronics* 602 (1994) 93.
- 5) S. Nakamura, T. Mukai and M. Senoh: *Appl. Phys. Lett.* 64 (1994) 1687.
- 6) H. Amano, M. Kito, K. Hiramatsu and I. Akasaki: *Jpn. J. Appl. Phys.* 28 (1989) L2112.
- 7) T. Matsuoka: *J. Crystal Growth* 124 (1992) 433.
- 8) S. Nakamura and T. Mukai: *Jpn. J. Appl. Phys.* 31 (1992) L1457.
- 9) B. Goldenberg, J. D. Zook and R. J. Ulmer: *Appl. Phys. Lett.* 62 (1993) 381.
- 10) M. A. Khan, J. N. Kuznia, D. T. Olson, M. Blasingame and A. R. Bhattarai: *Appl. Phys. Lett.* 63 (1993) 2455.
- 11) C. Wang and R. F. Davis: *Appl. Phys. Lett.* 63 (1993) 990.
- 12) S. Strite, M. E. Lin and H. Morkoç, *Thin Solid Films* 231 (1993) 197.
- 13) S. Nakamura, N. Iwasa, M. Senoh and T. Mukai: *Jpn. J. Appl. Phys.* 31 (1992) 1258.
- 14) J. A. Van Vechten, J. D. Zook, R. D. Horning and B. Goldenberg: *Jpn. J. Appl. Phys.* 31 (1992) 3662.
- 15) S. Nakamura: *Jpn. J. Appl. Phys.* 30 (1991) 1620.
- 16) S. Nakamura, Y. Harada and M. Senoh: *Appl. Phys. Lett.* 58 (1991) 2021.