

Diffusion of Carbon in SiO₂ Films and Its Segregation at Si/SiO₂ Interface

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Diffusion of carbon and its segregation at Si/SiO₂ interface were investigated using carbon-incorporated BPSG films and carbon-implanted SiO₂ films. It was found that carbon atoms diffuse in SiO₂ film at a temperature as low as 500°C, and they segregate at the Si/SiO₂ interface. Annealing in oxidizing ambient stimulates the carbon diffusion in SiO₂ and is effective for removing carbon from the films and for the suppression of carbon segregation. FE-TEM and EELS observations revealed that carbon atoms exist on the SiO₂ side of the interface and another carbon-rich phase is formed in SiO₂.

1. INTRODUCTION

Interlayer-dielectrics, such as SiO₂ and BPSG films, have come to be deposited with organic sources, such as TEOS, by LPCVD, because it provides better coverage than films deposited with non-organic sources, such as SiH₄, by APCVD. However, in the case of films with organic sources, carbon cannot be eliminated from the films during the deposition. It has been pointed out that carbon atoms segregate at the Si/SiO₂ interface and degrade the isolation characteristics, because segregated carbon atoms act as positive charge¹⁾. Therefore, it is important to investigate the characteristics of carbon in SiO₂ films on Si substrates. In this paper, diffusion characteristics of carbon in SiO₂ films and the state of segregated carbon at the Si/SiO₂ interface were investigated.

2. EXPERIMENT

LP-BPSG films were deposited on thermally grown SiO₂ films using TEOS (Si(OC₂H₅)₄), TMB (B(OCH₃)₃), phosphine (PH₃) and oxygen (O₂) at 600°C by LPCVD at 0.8torr. Carbon ion implanted SiO₂ films were also fabricated. The samples were annealed in a nitrogen or oxygen ambient. Diffusion characteristics of carbon in BPSG and in SiO₂ were measured by SIMS. Electrical influence of carbon was examined with C-V measurements for capacitors and with Id-Vg measurements for Al-gate field transistors. Cross-sectional FE-TEM and EELS were performed for the analysis of segregated carbon at the interface.

3. EFFECT of CARBON SEGREGATION

Figure 1 shows a typical SIMS profile for as-deposited and annealed samples with LP-BPSG films. In as-deposited films, a concentration of about 10²⁰cm⁻³ of carbon atoms was incorporated in the BPSG films. However, after annealing, most of the carbon atoms were segregated at the Si/SiO₂ interface. Figure 2 shows the positive charge and surface state density induced by the interfacial carbon in the annealed samples. Samples were fabricated by the deposition of various thickness of BPSG films on thermally grown SiO₂. Positive charge was generated by the segregation of carbon¹⁾, and the charge density was proportional to the carbon

concentration, while surface state density did not depend on the carbon concentration.

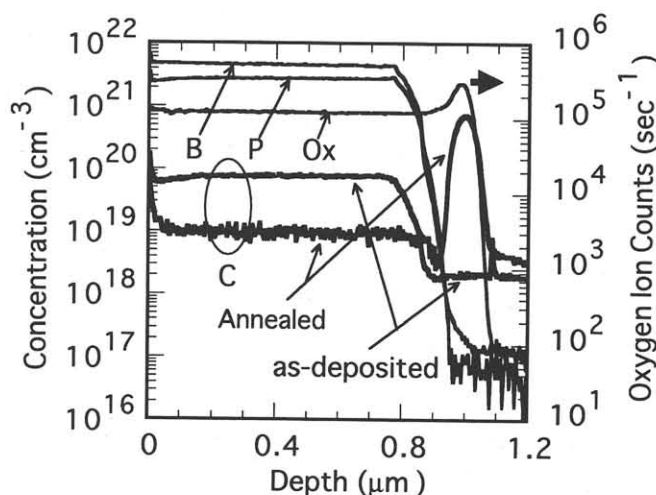


Fig.1 SIMS profiles for as-deposited and annealed samples. 900nm-thick BPSG was deposited on 100nm-thick SiO₂ films. The sample was annealed at 900°C for 30 min.

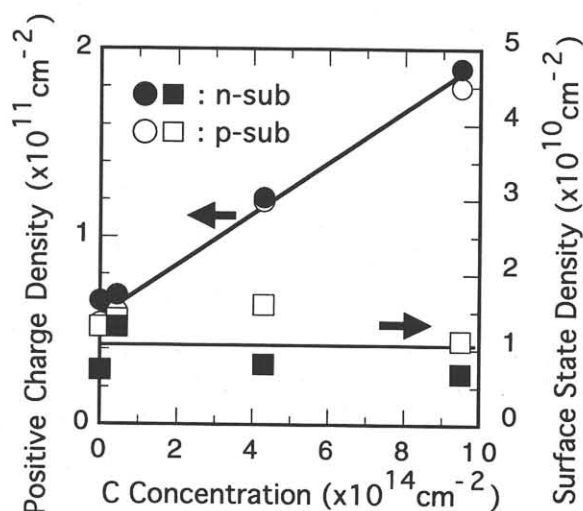


Fig.2 Positive charge and surface state density induced by the interfacial carbon.

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4. DIFFUSION of CARBON in SiO₂

Diffusion of carbon in SiO₂ was examined with carbon implanted samples in order to examine the diffusion characteristics in a wide temperature range. Carbon ions were implanted with a dose of $3 \times 10^{15} \text{ cm}^{-2}$ at 50keV into thermally grown SiO₂ films. Figure 3 shows depth profiles of carbon for samples annealed in a nitrogen ambient for 30 minutes. Some amount of carbon diffuses from the initial depth to the surface and to the Si/SiO₂ interface. This suggests that carbon atoms can make stable bonds at the surface and the interface. Figure 4 shows the segregated carbon concentration at the Si/SiO₂ interface for the samples annealed in the temperature range between 500°C and 800°C. Figure 4 indicates that even at a temperature as low as 500°C, carbon atoms diffuse in SiO₂ films and segregate at the interface.

Annealing in oxygen ambient showed quite different diffusion characteristics. Figure 5 shows carbon profiles for samples annealed in oxygen ambient. Concentrations of carbon atoms at the surface and in the SiO₂ film were much lower than those shown in Fig.3, when the annealing temperature was higher than 700°C. Bonds between carbon and silicon atoms in SiO₂ films were broken and carbon atoms were eliminated as CO or CO₂ molecules. Thus, the carbon at the surface disappeared and the interfacial carbon density was lowered.

Adopting this oxygen annealing process, the influence of carbon on the positive charge can be suppressed. Figure 6 shows the threshold voltages for field transistors using LP-BPSG films. Without the annealing in an oxygen ambient, the interfacial carbon concentration was about $4 \times 10^{15} \text{ cm}^{-2}$, and the threshold voltages for NMOS field transistors were lowered to about 0 V due to the carbon segregation even though the oxide thickness was 1.4μm. On the other hand, almost the same threshold voltages about 40 V were obtained for both PMOS and NMOS transistors, belonging to the low carbon concentration group, which were fabricated with the oxygen annealing process.

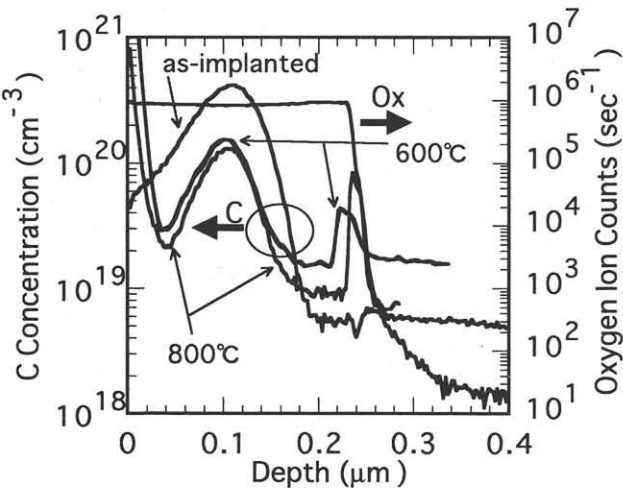


Fig.3 Depth profile of carbon for carbon-implanted samples annealed in a nitrogen ambient.

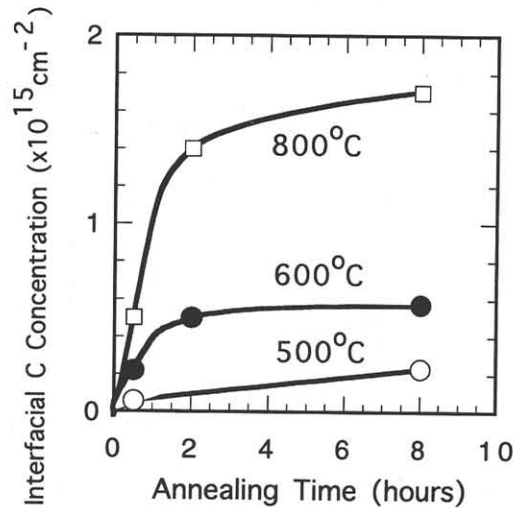


Fig.4 Segregated carbon concentration at the Si/SiO₂ interface for the annealed samples.

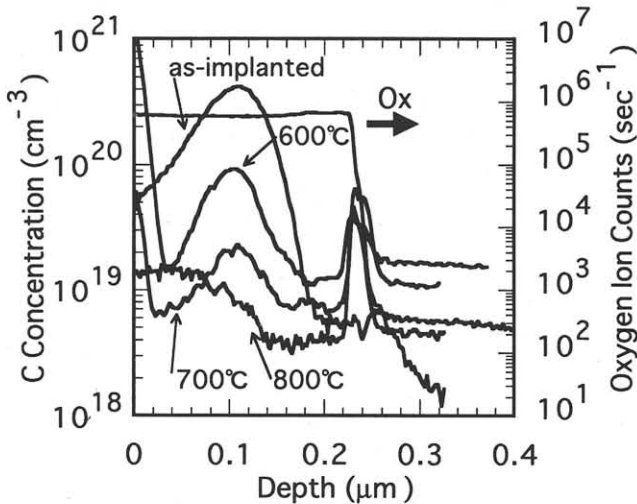


Fig.5 Depth profile of carbon for samples annealed in an oxygen ambient.

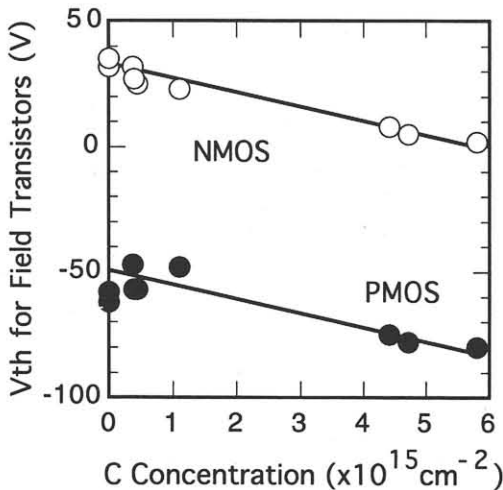


Fig.6 Threshold voltages for field transistors. Total thickness of BPSG and SiO₂ was 1.4μm.

5. SEGREGATED CARBON at Si/SiO₂ INTERFACE

FE-TEM and EELS measurements by HITACHI HF-2000 clearly indicated that carbon atoms segregate on the SiO₂ side and that another carbon related phase is formed. Figure 7 (a) and (b) shows the cross-sectional TEM at the Si/SiO₂ interface, where the interfacial carbon density was $1.9 \times 10^{16} \text{cm}^{-2}$ and $2.2 \times 10^{16} \text{cm}^{-2}$, respectively. Round areas having different contrast to the surrounding SiO₂ are found for both samples. The diameter of the area was about 5nm for the higher carbon concentration sample, while smaller areas were observed for the sample with lower segregated carbon concentration. EELS spectra were measured at the three points, as indicated in Fig.8: (A) Si side of the interface, (B) SiO₂ side of the interface and (C) round area observed in SiO₂. On the Si side of the interface, no carbon was observed as shown in Fig.9(A). On the contrary, on the SiO₂ side,

carbon atoms were detected as a peak at 284 eV, as shown in Fig.9(B). Much higher intensity of carbon was detected at the round area as shown in Fig.9(C). These results indicate that carbon atoms segregate on the SiO₂ side of the interface and some carbon atoms form another carbon-rich phase, such as SiC, in SiO₂ films.

6. CONCLUSION

Carbon atoms diffuse in SiO₂ films at a temperature as low as 500°C. Annealing in oxygen stimulates the carbon diffusion in SiO₂ and is effective for removing carbon from the films and for decreasing the positive charges. They segregate on the SiO₂ side of the Si/SiO₂ interface, and a part of carbon atoms form carbon-rich phase in SiO₂.

Reference

- 1) H.Mori et al., Ext.Abst. SSDM p.904 (1994).

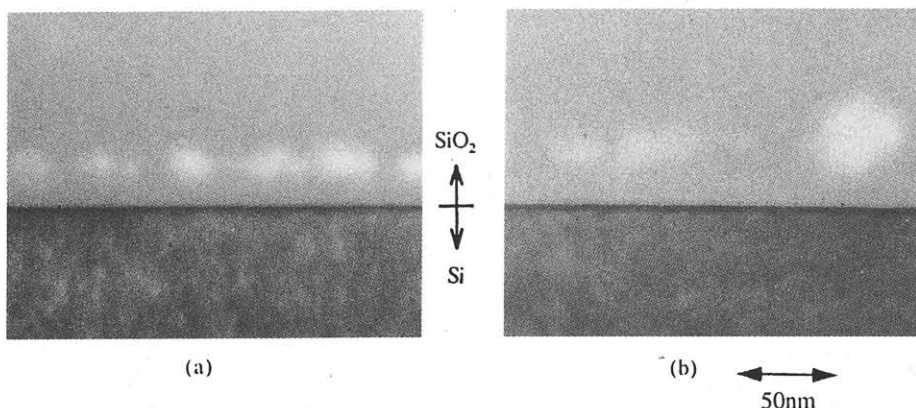


Fig.7 Cross-sectional TEM at the Si/SiO₂ interface for two samples with different interfacial carbon concentration: (a) $1.9 \times 10^{16} \text{cm}^{-2}$ and (b) $2.2 \times 10^{16} \text{cm}^{-2}$.

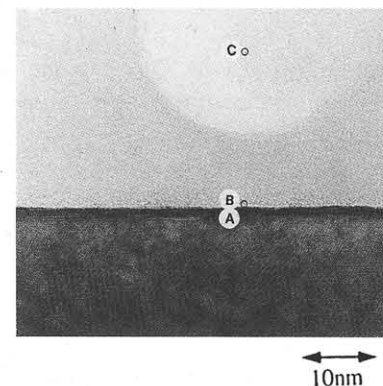


Fig.8 Cross-sectional TEM at the Si/SiO₂ interface where carbon atoms are segregated. Three points indicated as A, B and C were analyzed by EELS.

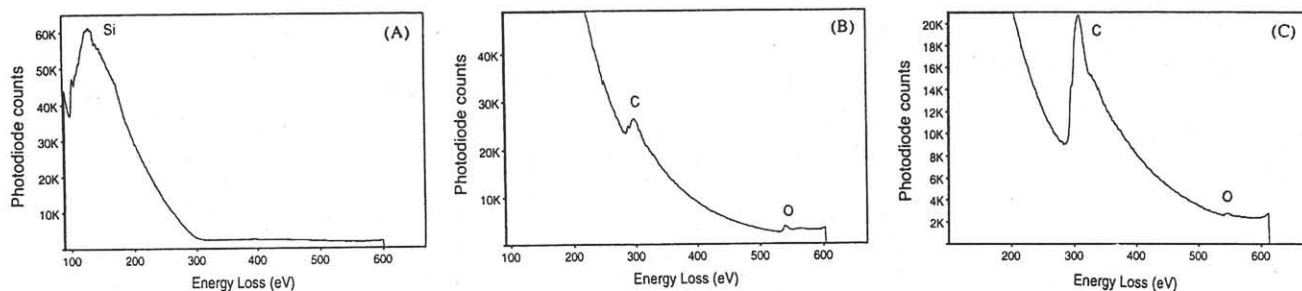


Fig.9 EELS spectra of (A) the Si side, (B) the SiO₂ side of the interface and (C) the round area in SiO₂.