# Highly Reliable SiO<sub>2</sub> Films Formed by UV-O<sub>2</sub> Oxidation

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## 1. Introduction

According to the further integration of ULSI devices, the gate oxide becomes thinner and thinner. Thus the gate oxide, which determines device reliability, is used on hostile conditions. Furthermore, for the shrinking of the ULSI devices, the channel profile control of MOSFETs requires the low temperature process even on the gate oxidation. Some new oxidation methods have been given a lot of attention as candidates for thin gate dielectrics. In this work, we studied the UV-O<sub>2</sub> oxidation method.

## 2. Sample Preparation

MOS capacitors were fabricated on p-type (100) substrate with resistivity of  $8.5 - 11.5 \Omega$ cm and n-type (100) substrate with resistivity of 10 - 20  $\Omega$ cm. Gate oxides were grown in dry O<sub>2</sub> ambient at 600 - 900 °C under ultraviolet irradiation (UV-O<sub>2</sub> Oxidation) and were grown in dry O<sub>2</sub> ambient at 900 °C (dry oxidation). The wave lengths of ultraviolet were 185 and 254 nm. The electrodes were formed from in-situ phosphorous-doped polycrystalline silicon.

## 3. Results and Discussions

Figure 1 shows the weibull plots in constant current TDDB measurement for the 63 Å oxides formed by UV-O<sub>2</sub> and dry oxidation at 900 °C. The charges to breakdown (Q<sub>BD</sub>) in the UV-O<sub>2</sub> oxides and the dry oxide are 2.3 and 0.9 C/cm<sup>2</sup>, respectively. Figure 2 shows the Q<sub>BD</sub> value at 50% cumulative failure ( $50\%Q_{BD}$ ) as a function of oxide thickness.  $50\%Q_{BD}$  in the oxide formed by UV-O<sub>2</sub> at 900 °C (shown as  $\bigcirc$ ) is larger than that in the dry oxide (shown as  $\bigcirc$ ) in the thicknesses ranging form 50 to 90 Å.



Fig. 1 Weibull plots in constant current TDDB measurement for oxides formed by UV- $O_2$  and dry oxidation.



Fig. 2  $Q_{BD}$  at 50% cumulative failure in constant current TDDB measurement as a function of oxide thickness.

Figure 3 shows the gate current density as a function of the oxide field for positive and negative polarities of samples with 64 Å oxides before and after F-N stress. The stress induced leakage current (SILC) in UV-O<sub>2</sub> oxide is less than that in dry oxide for both polarities. These results indicate that UV-O<sub>2</sub> oxides are stable against the electrical stresses as compared with the dry oxides.



Fig. 3  $J_G$ - $E_{OX}$  characteristics for (a) positive and (b) negative polarities of samples with 64 Å oxides formed by UV- $O_2$  and dry oxidation.

Figure 4 shows the gate voltage shift in the UV-O<sub>2</sub> oxide and the dry oxide during constant current stressing of -0.1  $\mu$ A/cm<sup>2</sup>. The gate voltage shift in UV-O<sub>2</sub> oxide is less than that in the dry oxide. The shift of gate voltage in the negative gate voltage direction indicates that hole trapping occur in the oxides. These facts indicate hole trap density of UV-O2 oxides smaller than that of dry oxide. Figure 5 shows the proposed defects in SiO<sub>2</sub> films. The Si-Si bond or oxygen vacancy (A), the Si dangling bond (B), the Si-H bond, and the Si-OH bond exist as defects in SiO<sub>2</sub> films. It has been reported that the oxygen vacancy acts as the hole trap [1], and the hole trap causes the SILC and the dielectric breakdown in SiO<sub>2</sub> [2]-[5]. The binding energies of those bonds, O=O bond, and UV-rays are shown in Table. 1. The binding energies of all bonds in SiO<sub>2</sub> and O=O bonds in O<sub>2</sub> gas are lower than the energy of UV-rays. It indicates that these bonds are cleaved by the UV-rays and the Si- bonds generated from these bonds react with activated oxidant to form SiO<sub>2</sub>. These suggest that the amount of Si-Si bonds in the  $UV-O_2$  oxide is less than that in the dry oxide.

Figure 6 shows the weibull plots in constant current TDDB measurement for the positive gate voltage stressing. The lifetimes in the UV-O<sub>2</sub> oxides is larger than that in the dry oxide. These indicate that the high reliable oxide can be formed by UV-O<sub>2</sub> oxidation even at 600 °C. This indicates that the UV-O<sub>2</sub> oxidation technology is very useful for the formation of the future devices which require low-temperature processing.



Fig. 4 Gate voltage shift of MOS capacitors during constant current stressing of -0.1A/cm<sup>2</sup>.



Fig. 5 The proposed defects near the Si-SiO<sub>2</sub> interface.

Table 1 Binding energies for bonds in  $SiO_2$  films and  $O_2$  gas and the energy of UV-rays.

	Binding Eenrgy
Si-Si	2.0 eV
Si-H	3.3 eV
Si-O	4.8 eV
O=0	5.2 eV
UV rays (185 / 254 nm)	6.7 / 4.9 eV



Fig. 6 Weibull plots in constant current TDDB measurement for oxides formed by UV-O<sub>2</sub> at 600  $^{\circ}$ C and dry oxidation.

# 4. Conclusions

We have developed and evaluated  $SiO_2$  films formed by UV-O<sub>2</sub> oxidation. It has been found that TDDB, SILC and hole-trapping characteristics in  $SiO_2$  formed by the UV-O<sub>2</sub> oxidation are superior to those formed by conventional oxidation. The UV-O<sub>2</sub> oxidation is very useful for the formation of the future devices.

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### References

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