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# The Formation of High Temperature Stable Co-Silicide from Co<sub>1-x</sub>Ta<sub>x</sub>/Si Systems

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# 1. Introduction

Co- SALICIDE process has been widely used in deep submicron CMOS device fabrications owing to the low resistivity and easy formation on narrow Si lines.[1, 2] The employment of the Co-salicide process, however, is limited upto 800°C, above which thermal degradation occurs due to the agglomeration of the Co-silicide films. Therefore, for the application to the merged process with logic and memory devices, the high temperature Co-salicide process is required.[3] In order to improve thermal stability of Cosilicide, we investigated Co-silicide formation using Co<sub>1-x</sub>Ta<sub>x</sub>/Si systems, and studied their thermal stability upon annealing.

### 2. Experiments

The Si(100) substrates were chemically cleaned by standard RCA clean, followed by dipping in a HF solution to remove the native oxide. Co1-xTax films were co-deposited by DC magnetron sputtering using Co and Ta targets. In order to prevent oxidation during silicidation reaction, Ti films were deposited on top of Co<sub>1-x</sub>Ta<sub>x</sub> films as a capping layer. Samples were annealed by RTA at various temperatures in N2 ambients, after which, unreacted metals were selectively etched by a chemical solution (H2SO4:H2O2=6:1). After chemical etching, 2nd RTA was performed at 850°C for 30s in N2 ambients. In order to test the thermal stability of the Co1-xTax/Si systems, samples were annealed in the furnace at 950°C for 30, 60 and 120min in Ar ambients. The Ta contents were analyzed by RBS. Microstructures and electrical properties were analyzed by XRD, AES, TEM and four-point probe.

#### 3. Results and Discussion

The Ta contents in Co/Ta alloy films were 8 atomic%, as determined from RBS analyses.(Fig. 1) Fig. 2 shows the changes of sheet resistance values of the Co/Si and the  $Co_{0.92}Ta_{0.08}/Si$  systems as a function of 1<sup>st</sup> RTA temperature. Compared with Co/Si systems, the transformation temperature from CoSi to CoSi<sub>2</sub> increased in the  $Co_{0.92}Ta_{0.08}/Si$  systems. Fig. 3(a) show that as the RTA temperature

increases, the  $CoSi_2(220)$  and (111) peaks appear in the Co/Si systems. Compared with Co/Si systems, the  $CoSi_2(200)$  peak is observed in the  $Co_{0.92}Ta_{0.08}/Si$  system (Fig. 3(b)), which indicates the presence of strong (200) preferred orientation in the Co-silicide films from  $Co_{0.92} Ta_{0.08}/Si$  systems.

Fig. 4 shows AES depth profiles of the  $Co_{0.92}Ta_{0.08}/Si$  systems after RTA process at 740 °C, followed by the 2<sup>nd</sup> RTA process at 850 °C. Co-silicide is formed on Si substrate, and Ta compound is observed to be formed at the top surface. Fig. 5 is TEM micrograph of the  $Co_{0.92}Ta_{0.08}/Si$  systems after the 2<sup>nd</sup> RTA process. Ta rich phases are observed on the surface of  $CoSi_2$  films.

Fig. 6 shows the sheet resistance of the Co-silicide films as a function of annealing time for the Co/Si and the  $Co_{0.92}Ta_{0.08}/Si$  systems. As annealing time increases, the sheet resistance of Co-silicide from Co/Si systems increases significantly, while the Co-silicide from  $Co_{0.92}Ta_{0.08}/Si$ systems maintain at low sheet resistance values. In Fig. 7(a), TaSi<sub>2</sub> phases are observed at the grain boundaries and at the surface of CoSi<sub>2</sub> films by HRTEM and EDS analyses (Fig. 7, 8) These results indicate that the agglomeration of CoSi<sub>2</sub> films is reduced by the formation of TaSi<sub>2</sub> at the grain boundaries of CoSi<sub>2</sub> films.

# 4. Conclusions

We obtained thermally stable  $\text{CoSi}_2$  films on Si(100) substrate from  $\text{Co}_{1-x}\text{Ta}_x/\text{Si}$  systems. The improvement of thermal stability of Co-silicide formed from alloyed Co/Ta thin films is due to the formation of the TaSi<sub>2</sub> phase at the grain boundaries of CoSi<sub>2</sub> films

#### References

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Fig. 1 RBS spectra from as-deposited Ti/Co<sub>0.92</sub>Ta<sub>0.08</sub>/Si systems.



Fig. 2 Sheet resistance values of Co-silicide films as a function of RTA temperature for 30s in  $N_2$  ambients for Co/Si systems and  $Co_{0.92}Ta_{0.08}/Si$  systems.



Fig. 3 XRD spectra of (a) Co/Si systems and (b)  $Co_{0.92}Ta_{0.08}/Si$  system after 1<sup>st</sup> RTA for 30s in N<sub>2</sub> ambients.



Fig. 4 AES depth profile of  $Co_{0.92}Ta_{0.08}/Si$  systems (a) 1<sup>st</sup> RTA at 740°C and (b) after 2<sup>nd</sup> RTA at 850°C.



Fig. 5 Cross-sectional TEM micrographs of  $Co_{0.92}Ta_{0.08}/Si$  systems after  $2^{nd}$  RTA at 850  $^\circ\!C$  .



Fig. 6 Sheet resistance values of Co-silicide films as a function of annealing time at 950  $^{\circ}$ C in Ar ambients with Co/Si systems and Co<sub>0.92</sub>Ta<sub>0.08</sub>/Si systems.



Fig. 7 Cross-sectional TEM micrographs of  $Co_{0.92}Ta_{0.08}$ /Si systems after annealing at 950°C in furnace. (a) Low magnification, (b) High resolution (at grain boundary), and (c) High resolution (at surface).



Fig. 8 EDS spectra for  $Co_{0.92}Ta_{0.08}/Si$  systems after annealing at 950 °C in furnace. (a) TaSi<sub>2</sub> at grain boundaries, (b) TaSi<sub>2</sub> at surface of CoSi<sub>2</sub> films, and (c) CoSi<sub>2</sub> films.