

**B-9-3****Scanning Tunneling Microscopy Study of Hf Silicate Formed by Ultrathin Hf Metal on SiO<sub>2</sub>:Effect of Hf/SiO<sub>2</sub> Thickness Ratio**

Jung-Ho Lee and Masakazu Ichikawa

Joint Research Center for Atom Technology, AIST Tsukuba Central 4, 1-1-1 Higashi, Tsukuba, Ibaraki 305-0046, Japan

Phone: +81-298-61-2597 FAX: +81-298-61-2577 e-mail: [sarangok@popsmail.com](mailto:sarangok@popsmail.com)**1. Introduction**

Recently, many high-k materials [1-9] like Hf(Zr)O<sub>2</sub>, Hf(Zr)Si<sub>x</sub>O<sub>y</sub>, Al<sub>2</sub>O<sub>3</sub>, La<sub>2</sub>O<sub>3</sub>, Y<sub>2</sub>O<sub>3</sub>, Gd<sub>2</sub>O<sub>3</sub>, GdSi<sub>x</sub>O<sub>y</sub>, and Pr<sub>2</sub>O<sub>3</sub> have been introduced as an alternative gate dielectrics to overcome the leakage current issue of SiO<sub>2</sub>. In a view of thermal stability, however, high-k oxides would be avoided due to their practical metastability [1,5] showing silicate formation far below the thermal budget required to decompose based upon thermodynamic data. Particularly, in light of the fact that as closer to oxide/Si interface chemical bonding at the interface should be similar to SiO<sub>2</sub>-Si interface so as to guarantee the interface reliability for successful device applications, silicate would be more probable for alternative dielectrics.

Hf forms the most stable oxide [6] with the highest heat of formation ( $\Delta H_f = 271 \text{ Kcal/mol}$ ) among the elements of transition IV-A group (Ti, Zr, Hf), and HfO<sub>2</sub> also shows a high dielectric constant of 30~40 along with a bandgap of ~5.6eV. These properties make Hf silicate one of the most promising candidates for alternative gate dielectrics.

In this paper, we present Hf silicate formation behaviors based upon ultrathin Hf metal on SiO<sub>2</sub> with annealing in ultrahigh vacuum (UHV) ambient by employing scanning tunneling microscopy (STM).

**2. Experimental**

Clean Si(001) surfaces were thermally oxidized to form 1 monolayer (ML) [10] and 1nm thick SiO<sub>2</sub> [11] by oxygen exposure for 3min at room temperature under  $2 \times 10^{-4} \text{ Pa}$  O<sub>2</sub> pressure and by one for 60min at 740°C under  $2 \times 10^{-2} \text{ Pa}$ , respectively. Electron-beam evaporated Hf was deposited on SiO<sub>2</sub> to make two splits of Hf/SiO<sub>2</sub> stack such as 1.7ML/1ML (Hf-rich) and 1ML/1nm (SiO<sub>2</sub>-rich). Dielectric surfaces were observed using STM by applying a sample bias of ~5V to inject tunneling electrons into the oxide conduction band. Local surface dielectric characteristics along with bandgap were measured with scanning tunneling spectroscopy (STS).

**3. Hf-rich silicate made by Hf/SiO<sub>2</sub> (1.7ML/1ML)**

Figs. 1 (a), (b), (c), and (d) show the as-Hf deposited, 400°C, 500°C, and 780°C annealed STM topographies, respectively. Ultrathin metal deposition on oxide film has already been found to form cluster-like morphology like Fig. 1 (a) on Au/TiO<sub>2</sub> and Pd/MgO [12]. Average diameter of cluster tends to increase with deposited metal coverage, and the density of state near the Fermi level also increases with cluster size implying development of metallic nature. In our case using 1.7ML thick Hf, average cluster diameter of ~3.6nm has been observed with 0.11nm root-mean-square (RMS) roughness. Increasing temperature up to 700°C causes the reaction between Hf and underlying SiO<sub>2</sub> showing disappearance of cluster feature with increasing RMS

roughness.

From previous study [11] focused on SiO<sub>2</sub>, it has been known that the tunneling spectra taken within bias range from -6 to +6V could give useful information about local surface dielectric characteristics as well as surface bandgap. An amount of surface bandgap in 1ML thick SiO<sub>2</sub> is observed to be ~2.5eV compared to bulk value of ~9eV, as shown in Fig. 2.

Dielectric degradation caused by annealing at 400°C for 2min in UHV could be explained by a formation of amorphous silicide based upon previous works [13,14]. Interestingly, further annealing to 500°C restores its dielectric property, and this peculiar behavior might be understood by different activation energies of amorphous silicidation and silicate reaction. Epitaxial Hf silicide phase appears at 780°C for 2min, and tend to align along Si<110> direction. Topographic line scan in Fig.1 (e) describes the formation of Si depletion zone near silicide phases implying that diffusion species for epitaxial silicide phase should be Si atoms. Si depletion zone is usually found along with right angle kinks of Si steps.

**4. SiO<sub>2</sub>-rich Hf silicate made by Hf/SiO<sub>2</sub> (1ML/1nm)**

Figs.3 (a), (b), (c), and (d) illustrate as-Hf deposited, 580°C, 650°C annealed STM topographies in UHV, and additionally 650°C annealed one in oxygen-rich ambient, respectively. Contrary to previous report [12], increased average cluster diameter of ~4.3nm is shown in spite of reduced metal coverage, and might be attributed to increased SiO<sub>2</sub> thickness. Formation of amorphous silicide at 400°C observed in Hf-rich silicate is not observed in SiO<sub>2</sub>-rich one. Lattice image shown in Fig.3 (b) reveals an appearance of crystalline phase in 580°C annealing for 2min, and is believed to form silicate phase from STS data of Fig.4 (a). The amount of silicate bandgap is larger than one of SiO<sub>2</sub>, and the surface bandgap narrowing to 5eV from 9eV, which usually observed when 1nm thick SiO<sub>2</sub> has been characterized by STS, almost disappears in case of silicate. Further annealing to 650°C in UHV drastically degrades a dielectric property due to silicidation, and additional annealing in oxygen-rich ambient enables a partial recovery of dielectric property.

**5. Conclusions**

Hf-rich and SiO<sub>2</sub>-rich Hf silicate were formed by controlling the thickness ratio of Hf/SiO<sub>2</sub> in UHV STM chamber. Reaction temperature for silicidation was found to change with different thickness ratio, and dielectric property of silicate was improved in SiO<sub>2</sub>-rich one compared to SiO<sub>2</sub>.

This work was supported by NEDO.

## References

- [1] G. D. Wilk, et al., J.Appl.Phys., 87, 484 (2000)
- [2] M. Kundu, et al., Appl.Phys.Lett., 78, 1517 (2001)
- [3] J. A. Gupta, et al., Appl.Phys.Lett., 78, 1718 (2001)
- [4] G. Lucovsky, et al., Appl.Phys.Lett., 77, 2912 (2000)
- [5] M. Copel, et al., Appl.Phys.Lett., 78, 1607 (2001)
- [6] B. H. Lee, et al., Tech.Dig.IEDM, 133 (1999)
- [7] L. Kang, et al., Tech.Dig.IEDM, 35 (2000)
- [8] J. Kwo, et al., J.Appl.Phys., 89, 3920 (2001)
- [9] S. J. Wang, et al., Appl.Phys.Lett., 78, 1604 (2001)
- [10] H. Watanabe, et al., Phys.Rev.Lett., 80, 345 (1998)
- [11] H. Watanabe, et al., J.Appl.Phys., 87, 44 (2000)
- [12] C. Xu, et al., J.Vac.Sci.Technol. A15, 1261 (1997)
- [13] S. Zaima, et al., J.Appl.Phys., 74, 6703 (1993)
- [14] C. S. Chang, et al., J.Appl.Phys., 2393 (1987)

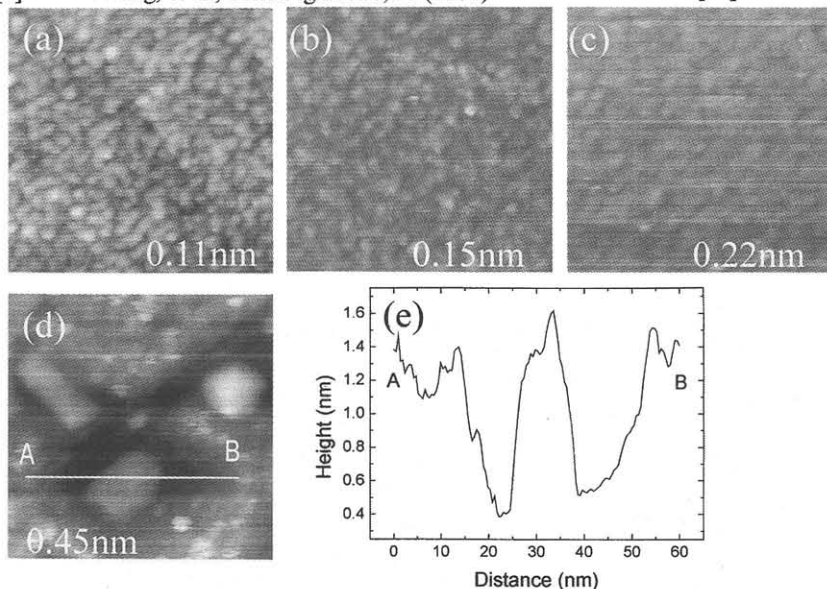


Fig.1 STM surface topographies showing (a) as-Hf deposited, (b) 400°C, (c) 500°C, and (d) 780°C annealed Hf/SiO<sub>2</sub> film stacks for 2min under UHV, respectively, in case of thickness ratio of 1.7ML/1ML. Topographic line scan shown in (e) is along the line AB noted in (d). RMS roughness value for each figure is indicated on the bottom corner. Scanning area is 50nm × 50nm.

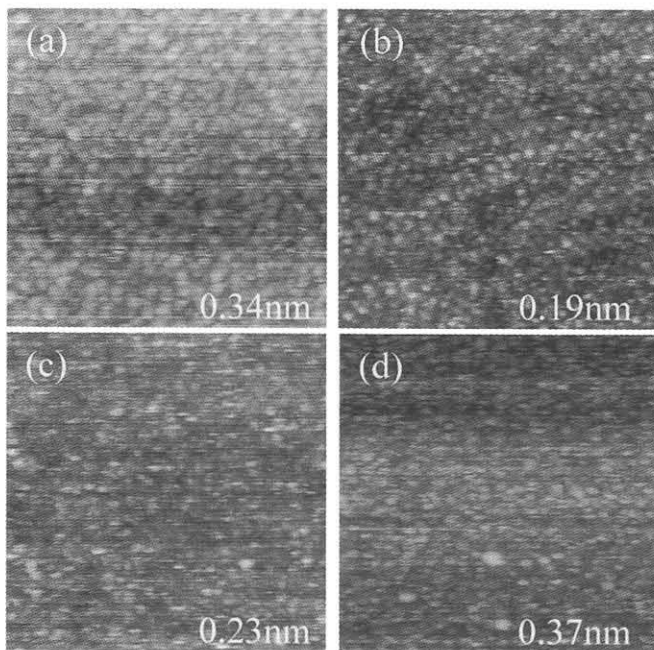


Fig.3 STM surface topographies showing (a) as-Hf deposited, (b) 580°C, (c) 650°C annealed Hf/SiO<sub>2</sub> film stacks for 2min under UHV, and (d) additionally 650°C annealed one for 2min under  $8 \times 10^{-2}$  Pa O<sub>2</sub> pressure, respectively, in case of thickness ratio of 1ML/1nm. Development of crystalline texture observed along with decreased roughness at 580°C is explained by the formation of SiO<sub>2</sub>-rich silicate in assistance with Fig.4 (a). After 580°C, an amount of roughness consistently increases with thermal budget, but the additional annealing in oxygen-rich ambient partially restores crystalline texture in spite of increased thermal budget. Scanning area is 80nm × 80nm.

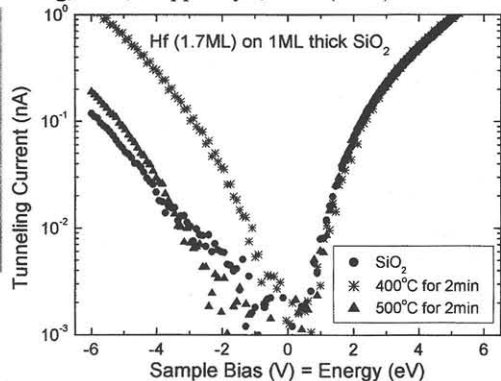


Fig.2 Log-linear I-V spectra of 1ML thick SiO<sub>2</sub> and annealed Hf/SiO<sub>2</sub> film stack measured by STS. The bias width of  $\sim 0$  tunneling current region observed nearby 0V ( $=E_F$ ) corresponds to the amount of surface bandgap of dielectrics. Fermi level of Hf-rich silicate formed at 500°C tends to be pinned to the conduction band edge.

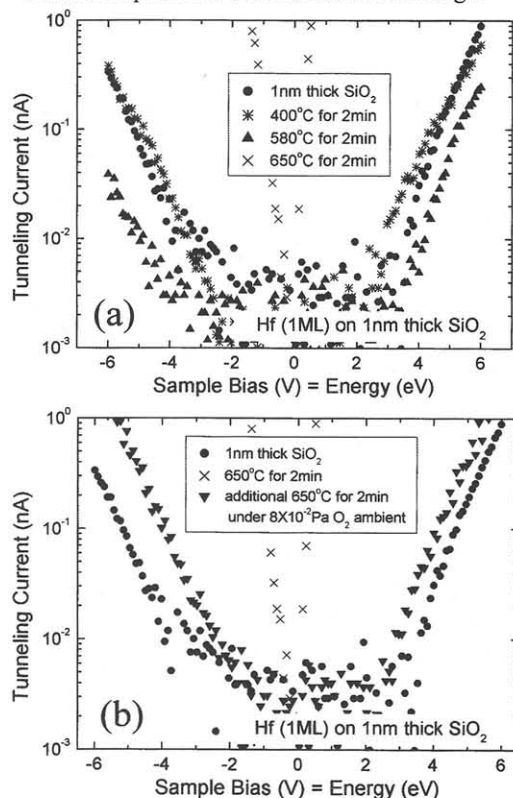


Fig.4 Log-linear I-V spectra of 1nm thick SiO<sub>2</sub> and annealed Hf/SiO<sub>2</sub> film stack measured by STS. Under SiO<sub>2</sub>-rich condition, dielectric degradation at 400°C shown in Fig.2 disappears, and wider ( $\sim 5$ eV) bandgap compared to 1nm thick SiO<sub>2</sub> is observed at 580°C annealed silicate. After additional annealing at 650°C in oxygen-rich ambient, partial recovery of dielectric property is shown in (b), and agrees well with Fig. 3 (d).