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**Investigation of the Polarization-Induced Charges in Modulation-Doped Al<sub>x</sub>-Ga<sub>1-x</sub>N/ GaN Heterostructures through Capacitance-Voltage Profiling and Simulation**

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**1. Introduction**

Presently, Al<sub>x</sub>Ga<sub>1-x</sub>N/GaN heterostructure field-effect transistors (HFETs) for high power, high temperature, and microwave applications are being rapidly developed. The polarization-induced electric field in an Al<sub>x</sub>Ga<sub>1-x</sub>N layer on GaN leads to a significant increase of the sheet concentration and the narrower confinement of the two-dimensional electron gas (2DEG) in an Al<sub>x</sub>Ga<sub>1-x</sub>N/GaN heterostructure in comparison with that in an Al<sub>x</sub>Ga<sub>1-x</sub>As/GaAs one. Quantitative characterization of the polarization field is necessary for a better understanding of the piezoelectric and spontaneous polarization and the strain relaxation in Al<sub>x</sub>Ga<sub>1-x</sub>N/GaN heterostructures. In this study, capacitance-voltage (C-V) method is developed to extract the polarization-induced charge in Al<sub>x</sub>Ga<sub>1-x</sub>N/GaN heterostructures.

**2. Experimental**

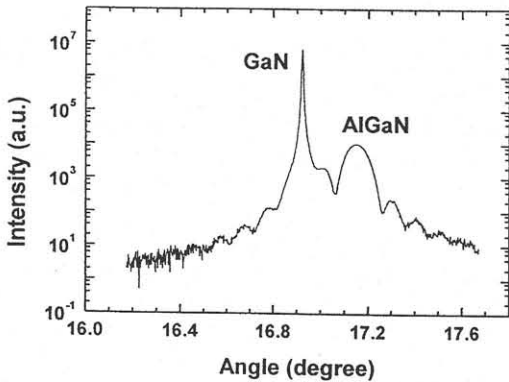


Fig.1.  $\omega/2\theta$  X-ray diffraction rocking curve of the modulation-doped Al<sub>0.22</sub>Ga<sub>0.78</sub>N heterostructure with the Al<sub>0.22</sub>Ga<sub>0.78</sub>N thickness of 45 nm. The scanning is along the GaN [0001] direction.

Three samples of high quality Al<sub>0.22</sub>Ga<sub>0.78</sub>N/GaN heterostructures were grown for this study. 2.0- $\mu$ m-thick unintentionally doped GaN epilayer, an unintentionally doped Al<sub>0.22</sub>Ga<sub>0.78</sub>N (i-AlGaN) layer, and a 25-nm-thick Si-doped Al<sub>0.22</sub>Ga<sub>0.78</sub>N (n-AlGaN) layer were deposited successively for each sample. The thickness of the i-AlGaN layer was 5 nm for Sample 1, 20 nm for Sample 2, and 50 nm for Sample 3. Fig. 1 is the  $\omega/2\theta$  X-ray diffraction rocking curve of Sample

3. Ohmic contact using Ti/Al/Pt/Au multilayers and Schottky contact using Pt/Au bilayers were fabricated on the sample surfaces, forming the structure of the Schottky diodes. C-V measurements were carried out at RT using a HP 4194A LCR meter with a frequency of 100 KHz.

**2. Results and Discussion**

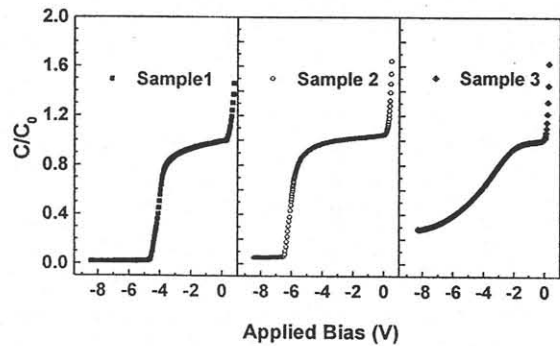


Fig. 2. Normalized C-V profiling of the three samples

Fig. 2 shows the normalized C-V profiling of the three samples. C<sub>0</sub> appeared in the figure is defined as:

$$C_0 = \epsilon_0 \epsilon_{AlGaN} A / d_{AlGaN} \tag{1}$$

where A is the area of Schottky contacts, d<sub>AlGaN</sub> is the barrier thickness,  $\epsilon_{AlGaN}$  is the relative dielectric constant of Al<sub>0.22</sub>Ga<sub>0.78</sub>N.

As shown in Fig. 2, when the reverse applied voltage is high, the capacitance is small and the corresponding boundary of the depletion layer is in GaN. With decreasing the reverse voltage, a capacitance plateau appears, where C/C<sub>0</sub> ≈ 1, corresponding to the depletion of the 2DEG located at the heterointerface. Between the left part and the plateau, there is a transition region, where the capacitance increases rapidly with decreasing the reverse voltage. The voltage positions of the transition region in three samples are much different from each other. It is about -4.0 V for Sample 1, -6.0 V for Sample 2, and -2.5 V for Sample 3. At the right of the plateau, another sharp capacitance slope (right slope) appears, indicating that the depletion layer is in Al<sub>0.22</sub>Ga<sub>0.78</sub>N. The start voltage

position of the right slope is about 0.24 V for Sample 1, 0.44 V for Sample 2, and 0.16 V for Sample 3, respectively.

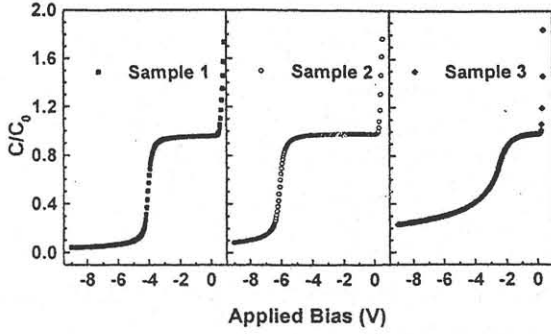


Fig. 3. Simulated C-V characteristics of the three samples

Numerical simulation was made based on the experiment results. The three-dimensional Fermi model was used. According to the simulation, the position of the transition region and the position of the right slope in every curve are determined by the polarization field, the barrier height  $\phi_b$ , and the doping profiles of the n-AlGaIn layer. The polarization field at the heterointerface influences the position of the transition region remarkably, but influences the position of the right slope slightly. Meanwhile,  $\phi_b$  and the doping level of n-AlGaIn influence both the positions of the transition region and the right slope remarkably. Thus, the polarization field can be extracted when the positions of the transition region and right slope in the simulated C-V curve both agree with those in the measured C-V curves. The simulated C-V curves of the three samples are shown in Fig. 2. They fit the measured curves well. The parameters extracted from the simulation are listed in Table I.

Table I. The parameters derived from the simulation, including the Pt/Al<sub>0.22</sub>Ga<sub>0.78</sub>N/GaN Schottky barrier height  $\phi_b$ , the donor concentration in n-AlGaIn  $N_{d1}$ , the donor concentration in GaN  $N_{d2}$ , and the sheet density of the polarization-induced charges  $N_p$ .

Sample No.	$\phi_b$ (eV)	$N_{d1}$ (cm <sup>-3</sup> )	$N_{d2}$ (cm <sup>-3</sup> )	$N_p$ (cm <sup>-2</sup> )
1	1.2	$1.51 \times 10^{18}$	$1.0 \times 10^{16}$	$6.78 \times 10^{12}$
2	1.2	$1.51 \times 10^{18}$	$1.0 \times 10^{16}$	$6.78 \times 10^{12}$
3	1.2	$1.51 \times 10^{18}$	$6.0 \times 10^{16}$	$1.30 \times 10^{12}$

$N_p$  at an Al<sub>x</sub>Ga<sub>1-x</sub>N/GaN heterointerface is determined by both the spontaneous and piezoelectric polarization of Al<sub>x</sub>Ga<sub>1-x</sub>N:

$$N_p(x) = N_{SP}(x) + N_{PE}(x), \quad (2)$$

where  $x$  is the mole fraction of Al in Al<sub>x</sub>Ga<sub>1-x</sub>N, and  $N_{SP}$  and  $N_{PE}$  are the sheet charge density induced by spontaneous and piezoelectric polarization, respectively.  $N_{SP}$  is determined by  $x$ , and  $N_{PE}$  can be expressed as:

$$N_{PE}(x) = N_{PEmax}(x)(1 - R) \quad (3)$$

where  $N_{PEmax}(x)$  is the piezoelectric polarization-induced sheet charge density when the Al<sub>x</sub>Ga<sub>1-x</sub>N layer on GaN is unrelaxed, and  $R$  is the lattice relaxation ratio of an Al<sub>x</sub>Ga<sub>1-x</sub>N layer on GaN. When the Al<sub>x</sub>Ga<sub>1-x</sub>N thickness is less than the critical thickness, the strain in the barrier is unrelaxed, and thus  $R = 0$ . When the Al<sub>x</sub>Ga<sub>1-x</sub>N barrier is thicker than the critical thickness, the strain in the barrier is partially relaxed or fully relaxed, and thus  $R > 0$ . In this case,  $R$  increases with increasing the barrier thickness.

The simulation suggests that  $R$  is equal for Sample 1 and Sample 2, which are different in barrier thickness. Therefore, we think that the Al<sub>0.22</sub>Ga<sub>0.78</sub>N thickness is below the critical thickness in these two samples. We think the Al<sub>0.22</sub>Ga<sub>0.78</sub>N layer in Sample 3 is partially relaxed. Since the strain in Al<sub>0.22</sub>Ga<sub>0.78</sub>N becomes much weaker, the density of the polarization-induced charges becomes much lower in Sample 3 than in Sample 1 or Sample 2.

#### 4. Conclusions

C-V measurement and simulation were performed to investigate the polarization-induced sheet charge density in Al<sub>x</sub>Ga<sub>1-x</sub>N/GaN heterostructures. The simulation based on the experimental results indicates that the sheet density of the polarization-induced charges at the heterointerface is  $6.78 \times 10^{12}$  cm<sup>-2</sup> in the samples with the Al<sub>0.22</sub>Ga<sub>0.78</sub>N thickness of 30 nm or 45 nm. The charge density reduces to  $1.30 \times 10^{12}$  cm<sup>-2</sup> in the sample with the Al<sub>0.22</sub>Ga<sub>0.78</sub>N thickness of 75 nm. It is thought that the reduction of the polarization-induced charges at the heterointerface is due to the partial relaxation of the Al<sub>0.22</sub>Ga<sub>0.78</sub>N layer on GaN.

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