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# Effects of Oxynitride-Based Interface Control on HfO<sub>2</sub> MIS Properties and FET Performance

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### **1. Introduction**

Control of high-k/Si interface is a key factor to obtain the robust structure for sub-100nm high-k gate CMOS. Recently, nitride-based interface control was applied to form thermally stable interfaces between high-k and Si[1,2]. However, the influence of oxynitride-based (including nitride-based) interface control on electrical properties of MIS structures has not been studied systematically.

In this paper, electrical properties of the HfO2 MIS structures with the oxynitride-based interfaces prepared by various processes are examined under the fixed HfO2 deposition and post-deposition processes.

## 2. Experimental Procedure

Various interface control (see Table I) on p-type Si (100) (resistivity 1-5  $\Omega$ cm) substrates was performed immediately after removing the chemical oxide layer formed by a conventional RCA cleaning. The composition and the physical thickness of prepared interface-control layers (ICLs) were measured using Auger electron spectroscopy (AES). The dielectric constant k of the interface oxynitride layer was evaluated using an experimental relationship between x and k, where x was defined as (SiO<sub>2</sub>)<sub>x</sub>(Si<sub>3</sub>N<sub>4</sub>)<sub>1-x</sub> for the SiON system [3].

2 nm-thick HfO<sub>2</sub> films were deposited onto ICLs at 200 °C by KrF excimer laser ablation in 0.1 Torr N2 ambient, followed by 15 s RTA in a N2 atmosphere (>99.99995 % purity) at 800°C. Au electrodes (200µm in diameter) and Al were deposited onto the HfO2 films and the backside of the substrates, respectively.

#### 3. Results and Discussion

We first discuss leakage current, then we describe influence of nitrogen density in ICLs on equivalent oxide thickness (EOT), flat-band voltage  $V_{FB}$ , and  $V_{FB}$  hysteresis of the HfO2 MIS structures, as well as performance of FETs.

Figure 1 shows the leakage current  $(J_g)$  versus EOT of the HfO2 MIS structures with various interface control. EOT was extracted from the comparison between measured C-V curves and ideal C-V curves [4]. For all the samples, Jg values are several orders magnitude lower than those of SiO2 [5].

Figure 2 shows EOT of the HfO2 MIS structures versus nitrogen density  $(N_N)$  in the ICLs. The results show a clear trend that the process induced EOT increase is suppressed with increasing N<sub>N</sub>, though a process dependent variation can still been seen. From this plot,  $N_N \approx 4 \times 10^{15} \text{ cm}^{-2}$  is considered to be the minimum N<sub>N</sub> to prevent the EOT for the sample structure and RTA condition described above. TEM observation also supports that the ICLs with higher nitrogen than this value are thermally stable in the RTA process (Fig. 3).

We have also found that  $V_{FB}$  is linearly reduced with  $N_N$ (Fig. 4). We estimated the fixed charge densities in the interfacial layer  $(N^{fix}_{int})$  and the HfO<sub>2</sub> layer  $(N^{fix}_{HfO2})$ , under the assumption that distribution of the fixed charges in each of these layers is uniform. As a result, the interface fixed charge  $N_{ini}^{fix} = \alpha N_N$  (positive,  $\alpha = 1.8 \times 10^{-3}$ ) and the HfO<sub>2</sub> fixed charge  $N_{HO2}^{fix} = 7.5 \times 10^{13}$  cm<sup>-2</sup> (negative) were obtained.

Furthermore, the result shown in Fig. 5 indicates that the flat-band voltage hysteresis  $V_{hys}$  changes with  $N_N$ . This fact suggests that the contribution of carrier-injection from the substrate to  $V_{hys}$  might become larger than that from the Au electrode with the increase of surface nitrogen density  $N_N$ .

Based on the experimental results, we now discuss the impact of ICLs on the performance of the HfO2 MISFETs as shown in Fig. 6. As a figure-of-merit for the current-drivability of the MISFETs, we introduce  $\mu_{eff}/CET_{gate}$ . Here,  $\mu_{eff}$  and  $CET_{gate}$  are the effective mobility and capacitance equivalent thickness of the gate stack, respectively. We estimate the optimum composition x of ICL at  $N_N$ =4x10<sup>15</sup> cm<sup>-2</sup> according to the following procedures.

First,  $CET_{gate}$  was estimated as  $CET_{gate} = EOT_{ICL} +$  $EOT_{HfO2} + EOT_{inv}$ , where the 2nd and 3rd terms are EOT of HfO<sub>2</sub> and inversion layer [6], respectively. As shown in Fig. 7, EOT of the oxynitride layer ( $EOT_{ICL}$ ) varies along the bold curve by changing x. Second, the mobility degradation was calculated by taking account of the remote charge scattering [7] from the fixed charges both from ICL and HfO<sub>2</sub>. Figure 8 shows the calculated  $\mu_{eff}$  and  $\mu_{eff}/CET_{gate}$  as a function of SiO<sub>2</sub> composition x in ICL at  $N_N=4x10^{15}$  cm<sup>-2</sup>. With x decrease, the calculated  $\mu_{eff}$  significantly decreases because of physical proximity of fixed charges in ICL and HfO<sub>2</sub> to the Si interface. Although  $\mu_{eff}/CET_{gate}$  is determined by the competing effects of mobility degradation and increase of 1/CETgate with decreasing of x, the result in Fig. 8 indicates that it has the maximum value at x = 0.

Figure 7 shows that  $EOT_{ICL}$  at x = 0 (and  $N_N = 4 \times 10^{15}$ cm<sup>-2</sup>) is 0.4 nm. This EOT corresponds to the physical thickness of 0.76 nm (about 2 monolayer (ML) of Si<sub>3</sub>N<sub>4</sub>). Thus, below 2 ML Si<sub>3</sub>N<sub>4</sub>, it is difficult to suppress the surface reaction, while above 2ML Si<sub>3</sub>N<sub>4</sub>, EOT<sub>ICL</sub> increases due to the increase of the Si<sub>3</sub>N<sub>4</sub> thickness itself. This result suggests that the interface with about 2 ML Si<sub>3</sub>N<sub>4</sub> is effective for obtaining the optimum FET performance.

# 4. Conclusions

We have performed a series of experiments in which both interface nitridation method and nitrogen density have been changed in HfO<sub>2</sub> MIS capacitors. We have found that EOT,  $V_{FB}$ , and  $V_{hys}$  of the HfO<sub>2</sub> MIS structures vary systematically with the sheet density of nitrogen in as-prepared ICLs. By using the experimentally obtained N<sub>N</sub> criterion, the optimum performance of HfO<sub>2</sub> MISFET has been expected in the interface layer with about 2ML of Si<sub>3</sub>N<sub>4</sub>.

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Table I Interface control layers (ICLs) examined and their preparation method .

ICL (T <sub>phy</sub> )	Process	Marker
Hydrogen- terminated	HF-last	м
SiO <sub>2</sub> (0.5-0.6nm)	Annealing in O <sub>2</sub> UV irradiation in O <sub>2</sub>	▲ △
Oxynitride (0.4-1.1nm)	Annealing in NO Remote plasma(N <sub>2</sub> /He) Direct plasma (N <sub>2</sub> /He)	
Nitride (0.4-1.5nm)	Annealing in NH <sub>3</sub> RF nitrogen radical	•



N<sub>N</sub> =2.7x10<sup>15</sup> cm<sup>-2</sup> N<sub>N</sub> =7.2x10<sup>15</sup> cm<sup>-2</sup>

Fig. 3 The cross-sectional TEM images for the samples after RTA at 800°C. (a) and (b) correspond to the sample (a) and (b) in Fig.2, respectively. (a): The interfacial layer is much thicker than that of the initial ICL. (b): Thickness of the interfacial layer is almost the same as that of the initial ICL. Note that  $HfO_2$  thickness is unchanged for both cases after RTA.



Fig. 6 Schematic diagram of the interface controlled  $HfO_2$  MISFET.  $N^{flx}_{HfO2}$  and  $N^{flx}_{ICL}$  denote the fixed charge density in  $HfO_2$  and ICL, respectively.



Fig. 1 The leakage current  $J_g$  versus EOT of the HfO<sub>2</sub> MIS structures with various interface control layers.



Fig. 4 Flat-band voltage  $V_{FB}$  as a function of  $N_N$ . The flat-band voltage shift is defined as the shift from  $V_{FB}$  of an Au/SiO<sub>2</sub>/Si MIS structure prepared in our experiment.



Fig. 7 EOT of oxynitride layer plotted as a function of x in  $(SiO_2)_x(Si_3N_4)_{1-x}$  as a parameter of the surface nitrogen density  $N_N(\text{cm}^{-2})$ .  $N_N < 4x10^{15}$  cm<sup>-2</sup> leads to the increase of EOT in the gate stack (see Fig. 2) in the post-deposition process.  $N_N > 4x10^{15}$  cm<sup>-2</sup> gives rise to the increase EOT of as-prepared ICL at a given x.

#### References

- [1] C. H. Lee, et al.: IEDM Tech. Dig. 2000, (2000) pp. 27-31.
- [2] R. Choi et al.: Symp VLSI Tech. 2001, (2001) pp. 15-16.
- [3] N. Yasuda et al.: Ext. Abstract of SSDM 2001, (2001) pp. 486-487.
- [4] N. Yasuda, and H. Satake: Ext. Abstract of SSDM 2001, (2001) pp. 202-203.
- [5] M. Fukuda, et al.: Jpn. J. Appl. Phys. 37 (1998) L1534.
- [6] S. Takagi, and A. Toriumi: IEEE Trans. on Electron Devices, 42 (1995) 2125.
- [7] F. Stern and W. E. Howard: Phys. Rev. 163 (1967)816.







sample (a) and (b) are shown in Fig. 3.

Nitrogen density N<sub>N</sub> (x 10<sup>15</sup>cm<sup>-2</sup>)

Fig. 5 Flat-band voltage hysteresis  $V_{hys}$  as a function of  $N_{N}$ .  $V_{hys} > 0$  and  $V_{hys} < 0$  mean hysteresis in clockwise and counterclockwise direction, respectively.



Fig. 8  $\mu_{eff}$  (left) and  $\mu_{eff}/CET_{gate}$  (right) as a function of x.  $\mu_{eff}$  at effective field of 0.4 MV/cm is plotted. The maximum  $\mu_{eff}/CET_{gate}$  is obtained at x = 0. EOT of ICL at x = 0 is 0.40 nm (see Fig. 7).