

## Enhancement of dielectric constant due to expansion of lattice spacing in CeO<sub>2</sub> directly grown on Si (111)

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### 1. Introduction

Crystalline oxide is a candidate for gate dielectrics, which can realize equivalent oxide thickness (EOT) less than 0.5 nm. We achieved direct growth of CeO<sub>2</sub> on Si (111) and EOT as thin as 0.38 nm [1]. The dielectric constant ( $\epsilon$ ) of the CeO<sub>2</sub> was 52, which is twice as large as the reported value of bulk (polycrystalline) CeO<sub>2</sub> ( $\epsilon \sim 26$ ). The value of  $\epsilon$  in ionic oxide is strongly related with crystal structures and/or lattice spacings. Enhancement of  $\epsilon$  by changing the crystal structure of Ta<sub>2</sub>O<sub>5</sub> was reported [2], but the crystalline symmetry is unchanged in the CeO<sub>2</sub> case. Anisotropy in  $\epsilon$  may not be existent in fluorite structures such as CeO<sub>2</sub>. The relationship between the lattice spacings and  $\epsilon$  theoretically studied in SrTiO<sub>3</sub> [3] and we supposed that the changes in lattice spacing in CeO<sub>2</sub> may be a considerable reason. In this study, we precisely evaluated the lattice spacings of CeO<sub>2</sub> directly grown on Si (epitaxial CeO<sub>2</sub>) and discussed the  $\epsilon$  enhancement characteristics.

### 2. Sample preparation

CeO<sub>2</sub> was grown on p-Si (111) substrate by molecular beam epitaxy (MBE) using metal Ce and O<sub>3</sub> as source materials at 700 °C. The thickness of CeO<sub>2</sub> was 10 nm. Fig.1 (a) is a cross-sectional TEM image of the CeO<sub>2</sub>/Si (111) interface and (b) is a plane view TEM image of CeO<sub>2</sub>. CeO<sub>2</sub> is confirmed to be single crystalline and directly grown on Si without any interfacial layer.

### 3. Results and discussions

To evaluate lattice spacings in ultra thin films, we performed in-plane XRD measurements [4], whose schematic diagrams are shown in Figs. 2. Reciprocal space maps around (220) planes for (a) Si substrate and (b) CeO<sub>2</sub>/Si were measured to separate Si and CeO<sub>2</sub> peaks as shown in Figs. 3. Ellipses in Figs. 3 indicate FWHM of diffraction peaks for Si and CeO<sub>2</sub>, which are determined by the peak separation method using gaussian curves. In Fig. 3 (b), a broad diffraction pattern of CeO<sub>2</sub> with a sharp Si diffraction pattern was observed. It can be seen that the 2 $\theta$  angle of CeO<sub>2</sub> is lower than that of Si. It means that the lattice spacing of CeO<sub>2</sub> (110) ( $d_{\text{CeO}_2(110)}$ ) is larger than that of Si (110) ( $d_{\text{Si}(110)}$ ). The relative change in  $d_{\text{CeO}_2(110)}$  from Si ( $\Delta d_{\text{CeO}_2(110)}$ ) is calculated to be +0.26 %.

Electron diffraction patterns (EDP) were observed by TEM [5] to examine the lattice spacing parallel to the sample surface ( $d_{\text{CeO}_2(111)}$ ), in addition to  $d_{\text{CeO}_2(110)}$ , as shown in Fig. 4. We can obtain  $\Delta d_{\text{CeO}_2(110)}$  and  $\Delta d_{\text{CeO}_2(111)}$  by comparing the spot positions of CeO<sub>2</sub> with those of Si. Table I summarizes  $\Delta d_{\text{CeO}_2(110)}$  and  $\Delta d_{\text{CeO}_2(111)}$  by in-plane XRD and EDP. The value of  $d_{\text{CeO}_2(111)}$  is also larger than  $d_{\text{Si}(111)}$  ( $\Delta d_{\text{CeO}_2(111)} = +0.3$  %).  $\Delta d_{\text{CeO}_2(110)}$  by EDP is evaluated to be +0.3 %, which agrees with the in-plane XRD result. The lattice constant of

bulk CeO<sub>2</sub> is reported to be 0.541 nm which is smaller than that of Si (0.543 nm) by -0.35 %. It is found that the lattice spacings in epitaxial CeO<sub>2</sub> are isotopically expanded by 0.6 %, compared with those in bulk CeO<sub>2</sub>.

We focused on the lack of oxygen in CeO<sub>2</sub> as a reason for the expansion of the lattice spacings. Because it is reported that Ce oxides have two crystal structures as CeO<sub>2</sub> and Ce<sub>2</sub>O<sub>3</sub> and that the lattice constant of Ce<sub>2</sub>O<sub>3</sub> is larger than that of CeO<sub>2</sub> by 3 % [6]. As the number of oxygen atoms in a unit cell of Ce<sub>2</sub>O<sub>3</sub> is smaller than that of CeO<sub>2</sub>, the coulomb interaction in Ce<sub>2</sub>O<sub>3</sub> would be lowered compared with that in CeO<sub>2</sub>, resulting in expansion of the lattice constant in Ce<sub>2</sub>O<sub>3</sub>. In order to confirm the lack of oxygen in epitaxial CeO<sub>2</sub>, we performed X-ray photoelectron spectroscopy (XPS) measurements. Additional in-gap state is observed in the valence band edge spectrum as shown in Fig. 5, which is determined to be due to the oxygen-defect-induced state [7]. This result indicates that the expansion of lattice spacings in epitaxial CeO<sub>2</sub> is due to the lack of oxygen atoms.

We consider that the enhancement of  $\epsilon$  is due to the expansion of the lattice spacings in epitaxial CeO<sub>2</sub> as described above. This tendency is similar to the result of SrTiO<sub>3</sub>[3] as shown in Fig. 6. However, the enhancement of  $\epsilon$  in CeO<sub>2</sub> is smaller than that of SrTiO<sub>3</sub>. It may indicate that the  $\epsilon$  enhancement mechanism in CeO<sub>2</sub> is different from that in SrTiO<sub>3</sub>; for example, the contribution of ionic polarizations on  $\epsilon$ , or the changes of electron states around Ce atoms due to oxygen defects.

### 4. Conclusions

We found that the lattice spacings in CeO<sub>2</sub> directly grown on Si are expanded compared with those in bulk CeO<sub>2</sub> due to the oxygen defects and that causes  $\epsilon$  enhancement in epitaxial CeO<sub>2</sub>. This result indicates that we can control the dielectric constant with the quantity of oxygen atoms in ionic crystalline oxides.

### References

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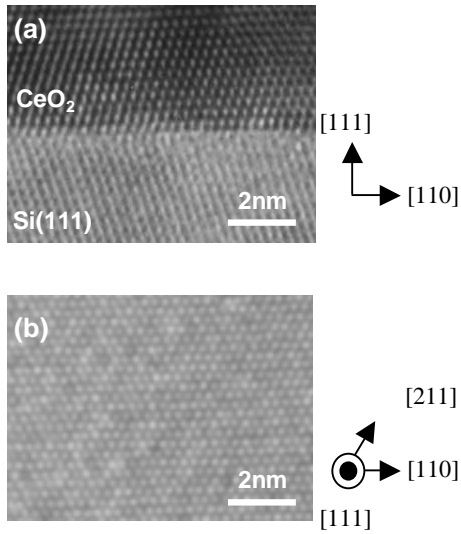


Fig. 1 (a)Cross-sectional view and (b)plane view TEM images of CeO<sub>2</sub>/Si(111).

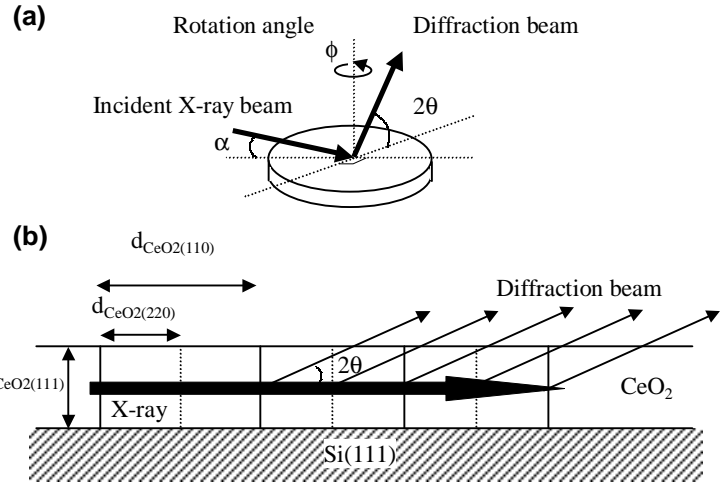


Fig. 2 Schematic diagrams of in-plne X-ray diffraction (XRD) measurements. (a) Geometry for incident and diffraction beams. X-ray is irradiated under critical angle of CeO<sub>2</sub>. (b)Direction of X-ray propagation in CeO<sub>2</sub>/Si(111). Lattice spacings perpendicular to sample surface can be obtained.

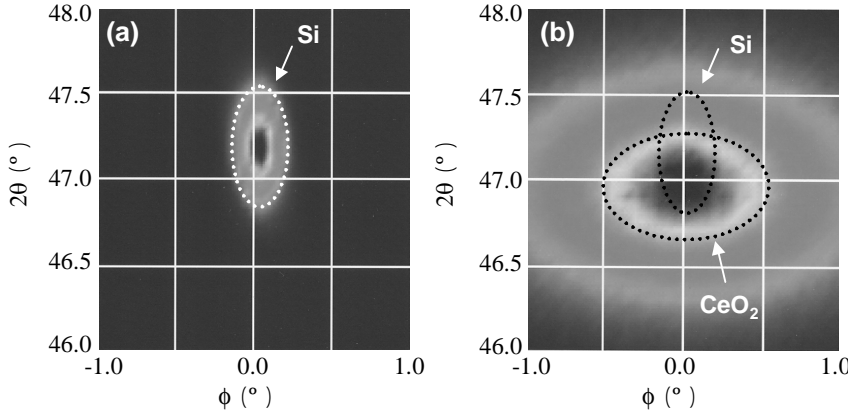


Fig.3 Reciprocal space maps around Si(220) and CeO<sub>2</sub>(220) measured by in-plane XRD for (a) Si and (b) CeO<sub>2</sub>/Si. Ellipses indicate FWHM of diffraction peaks for Si and CeO<sub>2</sub>. Incident X-ray angle ( $\alpha$ ) was 0.2°.

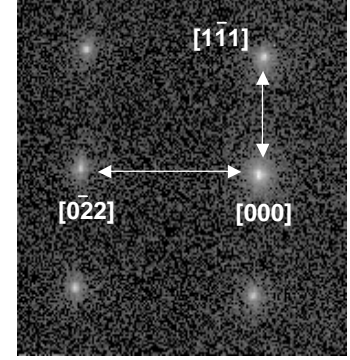


Fig. 4 Electron diffraction pattern (EPD) of CeO<sub>2</sub>. Spot positions of CeO<sub>2</sub> [022] and [111] from [000] were compared with those of Si.

Table I Relative changes in CeO<sub>2</sub> (110) and (111) lattice spacings from Si ( $\Delta d_{\text{CeO}_2(110)}$  and  $\Delta d_{\text{CeO}_2(111)}$  ).

Experimental method	in-plne XRD ( $\pm 0.02\%$ )	EDP ( $\pm 0.1\%$ )
$\Delta d_{\text{CeO}_2(110)}$	+0.26	+0.3
$\Delta d_{\text{CeO}_2(111)}$	-	+0.3

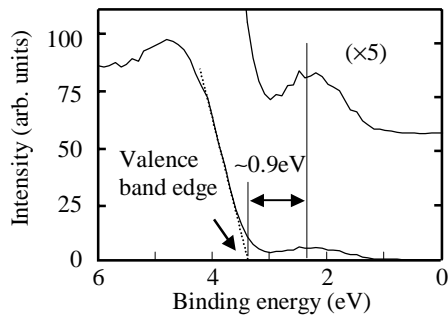


Fig. 5 XPS spectrum of CeO<sub>2</sub> valence band edge. In-gap state (~0.9 eV from the valence band edge) attributed to oxygen defects was observed.

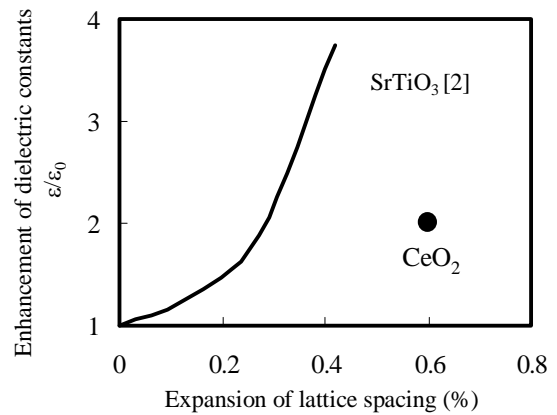


Fig. 6 Relationship between expansion of CeO<sub>2</sub> (111) lattice spacing and enhancement of dielectric constant for CeO<sub>2</sub> compared with SrTiO<sub>3</sub> [2] .