

# First-Principles Calculation of Bandgap Bowing Parameter for Wurtzite InAlGa<sub>N</sub> Quaternary Alloy using Large Supercell

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## 1. Introduction

Solid state ultraviolet (UV) light emitters have drawn a great deal of interests in recent years as the light sources for solid-state lighting, future high density optical storage, photo-catalytic disinfection and so on. At present, GaN-based UV emitters are the only viable choices for such applications. Although most of the demonstrated UV emitters use the AlGa<sub>N</sub> ternary alloy with high Al content to emit lights with shorter wavelengths, increase of the AlGa<sub>N</sub> thicknesses and Al contents is limited because of the epitaxial cracks caused by the lattice mismatch between the AlGa<sub>N</sub> and the underneath GaN. Addition of In into the ternary alloy can relax the material strain and thus formed InAlGa<sub>N</sub> alloy is reported to increase the UV emission intensities[1]. The quaternary alloy can be lattice-matched with wide variation of the bandgaps by choosing the Al/In composition to be 4.7[2]. Understanding of material parameters of the quaternary InAlGa<sub>N</sub> is critical as a part of device design including the bandgap bowing parameters. First principle calculation predicted the bowing parameter of zinc-blende InAlGa<sub>N</sub> to be as high as 2.59 eV, however, it was not on practical wurtzite structure and their calculation is limited to small 16 atoms supercell[3].

In this study, we calculate the bowing parameter of wurtzite InAlGa<sub>N</sub> using large supercells by first-principles technique. Neighboring effect of the group-III atoms in the 192-atoms supercell is examined. The calculated bowing parameters are over 2 eV regardless of the atomic positions which support the experimental results.

## 2. Calculation model

We employ 192-atoms ( $3a \times 3a \times 2c$ ) supercell models for Ga<sub>96</sub>N<sub>96</sub>, Al<sub>1</sub>Ga<sub>95</sub>N<sub>96</sub>, In<sub>1</sub>Ga<sub>95</sub>N<sub>96</sub> and In<sub>1</sub>Al<sub>1</sub>Ga<sub>94</sub>N<sub>96</sub> to extract the bowing parameter of InAlGa<sub>N</sub>. In this supercell, the minimum composition ratio of substitutional elements (Al and In) is 1.04 % (=1/96), which is small enough precise calculation. Neighboring effect between Al and In in GaN host material is investigated using two types of interatomic configurations for the InAlGa<sub>N</sub>. The Al and In are placed at 2nd nearest neighbor position (abbreviated to 2NN, the distance between In and Al is 3.2 Å) and at the farthest position (FAR; 10.8 Å).

The Vienna Ab initio Simulation Package[4,5] based on the Kohn-Sham density-functional theory using a

plane-wave basis set is used for the first-principle calculations. Ultrasoft Vanderbilt pseudopotentials[6] using the generic gradient approximation[7] are employed for the exchange-correlation functional. The cutoff energy of the basis set is set at 21 Ry.

After completing the structural relaxation for each model, the bandgaps are calculated.

## 3. Calculation

Fig. 1 shows the calculated bandgaps for the four compositions. The difference of the bandgap between the alloy and Ga<sub>96</sub>N<sub>96</sub> is plotted as a function of the ( $x$ - $y$ ), where the  $x$  and  $y$  are the number of In and Al atoms in the unit cell, respectively. By simple linear interpolation between the bandgap of In<sub>1</sub>Ga<sub>95</sub>N<sub>96</sub> and that of Al<sub>1</sub>Ga<sub>95</sub>N<sub>96</sub>, the bandgap of In<sub>1</sub>Al<sub>1</sub>Ga<sub>94</sub>N<sub>96</sub> *without* bowing parameter can be estimated as seen in the cross mark in Fig. 1. On the contrary, the calculated bandgaps of In<sub>1</sub>Al<sub>1</sub>Ga<sub>94</sub>N<sub>96</sub> for the 2nd nearest neighbor and the farthest are far lower than the interpolated value, implying a large bowing parameter.

As shown in Fig.1, the difference of the bandgap between the two quaternary models is negligibly small compared with another quaternary alloy, such as GaInNAs. This can be explained by the following two reasons: (1) Al and In atoms are connected only with N atom, (2) the physical properties of Al, Ga and In are very similar. On the contrary, In-N clusters in GaInNAs alloy cause strong neighboring effect[8]. The total energies of the two models are almost same ( $\Delta E < k_B T$ ), which implies Al and In atoms can be randomly placed in GaN host material keeping the almost same value of the bandgap.

The bowing parameter of InAlGa<sub>N</sub> can be derived from the following equation[9],

$$E_g^{\text{InAlGa}_N} = \left[ xE_g^{\text{InN}} + (1-x)E_g^{\text{Ga}_N} \right] + \left[ yE_g^{\text{AlN}} + (1-y)E_g^{\text{Ga}_N} \right] - E_g^{\text{Ga}_N} - (x+y)(1-x-y)b \quad (1)$$

where  $b$  is the bowing parameter, and  $E_g^{\text{material}}$  denotes the bandgap of each binary material. The first and second terms denote the bandgaps of In <sub>$x$</sub> Ga <sub>$1-x$</sub> N and Al <sub>$y$</sub> Ga <sub>$1-y$</sub> N, respectively. Using the calculated bandgaps shown in Fig. 1, the bowing parameter of InAlGa<sub>N</sub> quaternary alloy for the two models is,

$$b = \begin{cases} 2.49 & \text{(2nd nearest neighbor)} \\ 2.19 & \text{(farthest)} \end{cases} \quad (2)$$

The above two bowing parameters are very close reflecting the close bandgaps shown in Fig. 1.

#### 4. Comparison with Experimental Results

In order to verify the calculated results, lattice-matched InAlGa<sub>N</sub> films were successfully grown on GaN (2.5 μm thickness) templates by metalorganic chemical vapor deposition (MOCVD). The thickness of InAlGa<sub>N</sub> films was around 100 nm. The bandgap and the composition ratio of each element were evaluated by cathodoluminescence and electron-probe microanalysis, respectively.

Fig. 2 shows the typical reciprocal space mapping of InAlGa<sub>N</sub> film by x-ray diffraction. A single peak is observed at the same diffraction angle of GaN, which implies the InAlGa<sub>N</sub> film is lattice-matched to the underlying GaN layer.

Among the various InAlGa<sub>N</sub> grown films, closely lattice-matched samples with small strain ( $\Delta a/a_{\text{GaN}} < 0.4\%$ ) were picked up and plotted as open circles in Fig. 3. Bandgaps of the lattice-matched InAlGa<sub>N</sub> using the calculated bowing parameters in Eq. (2) are also shown in the same figure. The calculated bandgap does not much depend on the interatomic configuration explained in the previous section. As shown in Fig. 3, the experimental values well agree with the calculated bandgaps supporting the large bowing parameters predicted in the first-principle calculations.

#### 5. Conclusion

We calculate the bowing parameter of wurtzite InAlGa<sub>N</sub> using large 192-atoms supercells by first-principles technique. Atomic neighboring effect of the group-III atoms is examined and the calculated bandgaps are not much dependent on the atomic positions. The bandgap bowing parameters are estimated to be 2.19-2.49 eV, which well agree with our experimental results.

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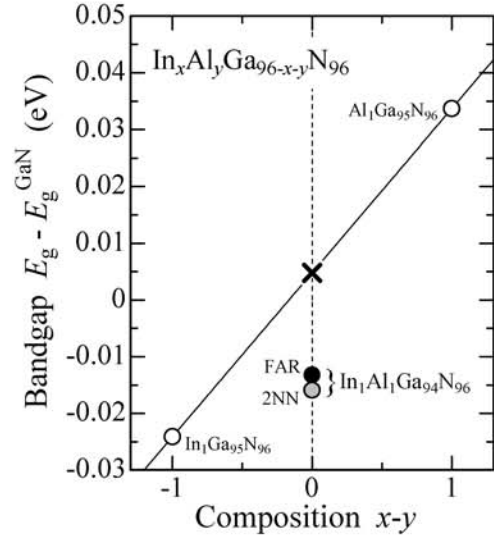


Fig. 1. Calculated bandgaps of AlGa<sub>N</sub>, InGa<sub>N</sub> and InAlGa<sub>N</sub> using first-principles calculation.

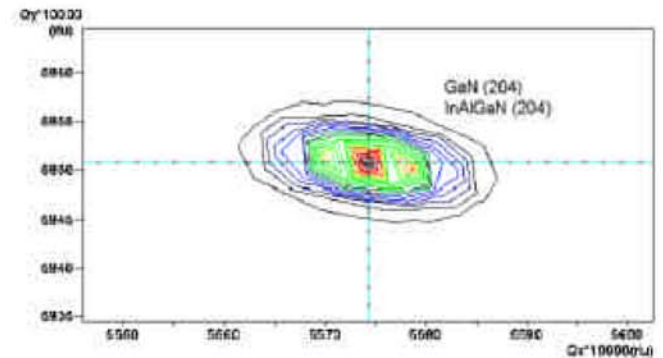


Fig. 2. Reciprocal space mapping of InAlGa<sub>N</sub> film by x-ray diffraction.

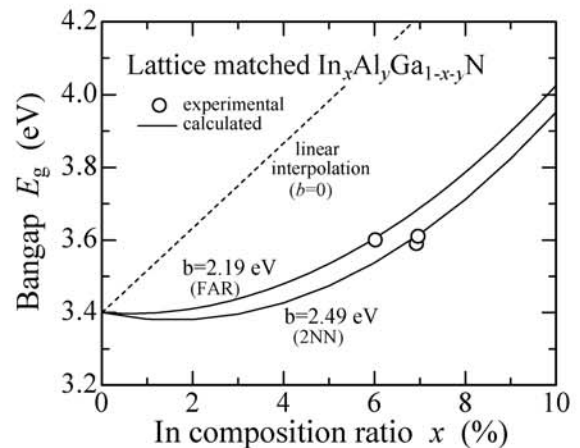


Fig. 3. Comparison of the InAlGa<sub>N</sub> bandgap between experimental results and calculated bowing curves.