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## VO<sub>2</sub> and V<sub>2</sub>O<sub>3</sub> Films Fabricated on (1000) or (10 $\bar{1}$ 0) Al<sub>2</sub>O<sub>3</sub> by Reactive RF-Magnetron Sputter Deposition and Annealing Processes

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### 1. Introduction

Vanadium dioxide (VO<sub>2</sub>) and vanadium sesquioxide (V<sub>2</sub>O<sub>3</sub>) are known as Mott-insulators showing abrupt metal-insulator transition (MIT) near 340 and 170K, respectively [1]. These vanadium oxides also show the structural phase transition (SPT) near the MIT temperatures (T<sub>C</sub>). It has been reported that the MIT temperature can be varied by applying electric field or by irradiating infrared light [2]. Although it remains controversial as to whether the MIT and the SPT are isochronous, the variability provides a large potential to the MIT materials for programmable thermal or photo sensors.

Vanadium cations can exist in one of several oxidation states, such as V<sup>3+</sup>, V<sup>4+</sup>, or V<sup>5+</sup>, to form several different vanadium oxides. They can also convert from one oxidation state to another, and easily coexist [3]. Therefore, it is very difficult to obtain stoichiometric VO<sub>2</sub> or V<sub>2</sub>O<sub>3</sub> films of single chemical state using a simple deposition method without subsequent annealing in a well-controlled environment [4,5]. The VO<sub>2</sub> or V<sub>2</sub>O<sub>3</sub> films have been fabricated using the reactivity through which V<sub>2</sub>O<sub>5</sub>, the oxide of the highest oxidation state (V<sup>5+</sup>) among the vanadium oxides, can be reduced to VO<sub>2</sub> or V<sub>2</sub>O<sub>3</sub> in appropriate environment [4-7]. Especially, earlier works had reported that annealing temperatures higher than 700°C had been required to reduce V<sub>2</sub>O<sub>5</sub> to V<sub>2</sub>O<sub>3</sub> in an environment flowing reducing agents [6,7]. In a previous work, the authors reported that the annealing in a very low pressure with no reducing agent could reduce sol-gel VO<sub>2</sub> films to V<sub>2</sub>O<sub>3</sub> at a temperature much lower than the earlier reported temperatures [5].

In the present work, the preparations of VO<sub>2</sub> and V<sub>2</sub>O<sub>3</sub> films were investigated using the oxide film initially deposited by reactive RF-magnetron sputter deposition technique. The difference of the films formed on two different substrates, (1000) and (10 $\bar{1}$ 0) Al<sub>2</sub>O<sub>3</sub>, was also briefly discussed.

### 2. Experiments

The reactive RF sputter deposition of vanadium oxide films was carried out using V-metal target of 4-inch diameter (99.9%, Kojyundo Kagaku) in the atmosphere of

O<sub>2</sub> gas mixed with Ar. The O<sub>2</sub> gas fraction was fixed at 12.3 %, and the operating pressure was 5 mTorr. The operating pressure was selected to have the best film quality through the study on the effect of chamber pressure on MIT characteristics of VO<sub>2</sub> [8]. The RF power was 300W. The films were post-annealed to fabricate stoichiometric VO<sub>2</sub> at 490°C with O<sub>2</sub> pressure of 30 mTorr. V<sub>2</sub>O<sub>3</sub> films were fabricated by reducing high quality VO<sub>2</sub> films in a very low pressure [5]. Two different kinds of sapphire (Al<sub>2</sub>O<sub>3</sub>) substrates, with crystalline orientations of (1000) or (10 $\bar{1}$ 0), were used. The thickness of vanadium oxide films were approximately 110 nm. The resistance was measured by standard four-point probe method.

### 3. Results and Discussion

First, the composition of as-deposited oxide film and annealed films was measured using Auger electron spectroscopy (AES), as listed in Table 1. Only VO<sub>2</sub> reveals MIT near 340K. Therefore, a reference film showing an abrupt MIT behavior near 340K was considered to be stoichiometric VO<sub>2</sub> film in the sensitivity calibration of vanadium and oxygen.

As shown in Table 1, the composition of an as-deposited oxide film was very similar to that of annealed VO<sub>2</sub> film showing MIT near 336K. Because the as-deposited oxide film did not show any MIT behavior, the film was not in the chemical state of VO<sub>2</sub>. XRD measurement (the data not shown here) also indicated that the as-deposited film mostly contained V<sub>2</sub>O<sub>5</sub> crystalline phases in spite of its oxygen deficiency.

Table 1 also listed the composition of a V<sub>2</sub>O<sub>3</sub> film showing MIT near 172K. The O/V ratio in the film, 1.66, was larger than the expected value, 1.5. The result indicates that the film probably contained some amount of different vanadium oxides with higher oxidation states.

Table 1. The composition of as-deposited vanadium oxide, VO<sub>2</sub>, and V<sub>2</sub>O<sub>3</sub> films on (10 $\bar{1}$ 0) Al<sub>2</sub>O<sub>3</sub>

films	Content Ratio, O/V
as-deposited film	2.04
VO <sub>2</sub>	2.08
V <sub>2</sub> O <sub>3</sub>	1.66

Figure 1 presents the resistance (R) vs. temperature (T) curves of VO<sub>2</sub> films formed on (1000) or (10 $\bar{1}$ 0) Al<sub>2</sub>O<sub>3</sub> substrate. As illustrated in Fig. 1(b), the critical temperature, T<sub>C</sub>, is defined as the temperature at which d(-log R)/dT shows the maximum value, and the peak width implicates the abruptness of MIT. The films formed on different substrates showed different T<sub>C</sub> values as much as 5.4K. T<sub>C</sub> of VO<sub>2</sub> on (10 $\bar{1}$ 0) Al<sub>2</sub>O<sub>3</sub> was 336.0K. In spite of the different MIT temperatures, both films showed abrupt MIT as shown in Fig. 1. The results in Fig. 1(a) and 1(b) also indicate that the VO<sub>2</sub> film on (10 $\bar{1}$ 0) Al<sub>2</sub>O<sub>3</sub> reveals more typical metallic behavior after MIT and underwent a sharper MIT than that on (1000)Al<sub>2</sub>O<sub>3</sub>. The VO<sub>2</sub> film on (1000)Al<sub>2</sub>O<sub>3</sub> showed that the resistance kept on decreasing even after MIT as indicated by an arrow in Fig. 1(a).

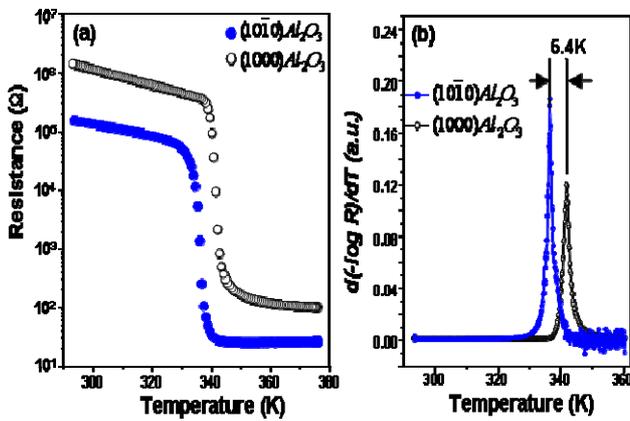


Figure 1. (a) R-T curves of VO<sub>2</sub> films deposited on single crystalline Al<sub>2</sub>O<sub>3</sub> substrates and post-annealed at 490°C; (b) d(-log R)/dT vs. T curves showing T<sub>C</sub>.

The R-T curves and d(-log R)/dT - T curves of V<sub>2</sub>O<sub>3</sub> films on (1000) or (10 $\bar{1}$ 0) Al<sub>2</sub>O<sub>3</sub> substrate are shown in Fig. 2(a) and 2(b), respectively. The films were prepared by reducing VO<sub>2</sub> films shown in Fig. 1. Similarly to our previous work using sol-gel VO<sub>2</sub> films [5], the VO<sub>2</sub> films fabricated in the present work easily lost their MIT characteristics during vacuum-annealing even at a temperature as low as 300°C in the chamber pressure of 2 × 10<sup>-6</sup> Torr. The vacuum-annealing at a temperature higher than 550°C could fabricate V<sub>2</sub>O<sub>3</sub> showing abrupt MIT near 170K.

Figure 2(a) indicates that the VO<sub>2</sub> film on (10 $\bar{1}$ 0) Al<sub>2</sub>O<sub>3</sub> reveals larger resistance change by MIT than that on (1000)Al<sub>2</sub>O<sub>3</sub>. The resistance change of V<sub>2</sub>O<sub>3</sub> films is slower than that of VO<sub>2</sub>, and the T<sub>C</sub> difference of V<sub>2</sub>O<sub>3</sub> films deposited on different substrates are only 2.4K. T<sub>C</sub> values of V<sub>2</sub>O<sub>3</sub> films on (10 $\bar{1}$ 0) and (1000)Al<sub>2</sub>O<sub>3</sub> were 172.5 and 174.9K, respectively.

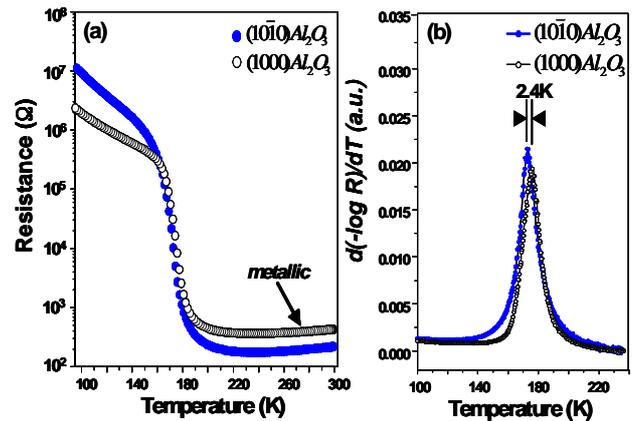


Figure 2. (a) R-T curves of V<sub>2</sub>O<sub>3</sub> films formed on single crystalline Al<sub>2</sub>O<sub>3</sub> substrates and annealed at 600°C; (b) d(-log R)/dT vs. T curves showing T<sub>C</sub>.

#### 4. Conclusion

VO<sub>2</sub> and V<sub>2</sub>O<sub>3</sub> films showing abrupt MIT near their typical T<sub>C</sub> were successfully fabricated by annealing vanadium oxide films deposited by reactive RF sputter deposition process. The annealing was carried out to prepare VO<sub>2</sub> in a low pressure O<sub>2</sub> ambient. V<sub>2</sub>O<sub>3</sub> film was fabricated by further annealing the VO<sub>2</sub> film with no reducing agent at 600°C in 2 × 10<sup>-6</sup> Torr. The annealing temperature to prepare V<sub>2</sub>O<sub>3</sub> film (≥550°C) is much lower than those reported in earlier works.

The dependence of T<sub>C</sub> on substrate was also shown for VO<sub>2</sub> and V<sub>2</sub>O<sub>3</sub> films. After the transition from insulator to metal, the films on (10 $\bar{1}$ 0) Al<sub>2</sub>O<sub>3</sub> showed lower resistance and more typical metallic behavior than those on (1000)Al<sub>2</sub>O<sub>3</sub> substrate.

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