

## Controlling anion composition at MIS interfaces on III-V channels by plasma processing

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### 1. Introduction

Scaling limit of Si LSIs has prompted intensive studies on MISFETs incorporating high-mobility materials such as Ge and III-V semiconductors. In order to form high-quality III-V MIS interfaces, the compositions and bonding structures of the cations and anions at the interface should be properly controlled [1]. We recently reported that deposition of SiO<sub>2</sub> or Al<sub>2</sub>O<sub>3</sub> on nitride InGaAs formed the MIS interfaces that exhibit good capacitor characteristics [2]. Here the nitride layer formed by ECR plasma was thicker than 1 nm. From the viewpoint of EOT scaling, thinner nitride layers are preferable. In this paper, we investigate the effects of plasma cleaning and nitridation (< 0.5 nm) on InGaAs using on-line Auger electron spectroscopy (AES). Electrical characterization has shown that such lightly-nitrided Al<sub>2</sub>O<sub>3</sub>/InGaAs interfaces also show good MIS characteristics.

### 2. Experimental Procedures

Experiments were carried out using a high-vacuum compatible ALD system which was equipped with a remote RF plasma source and an analysis chamber for AES (Fig. 1). In order to fabricate MIS capacitors, n-type In<sub>0.53</sub>Ga<sub>0.47</sub>As(100) epitaxial wafers with doping concentration of  $3 \times 10^{16} \text{ cm}^{-3}$  were etched in an NH<sub>4</sub>OH solution for 1 minute. The InGaAs surface was then subjected to plasma cleaning using H<sub>2</sub> and/or plasma nitridation using N<sub>2</sub> at 250 °C under the RF power of 300 W. ALD of Al<sub>2</sub>O<sub>3</sub> was carried out using trimethylaluminum (Al(CH<sub>3</sub>)<sub>3</sub>) and H<sub>2</sub>O at 250 °C. A high Al(CH<sub>3</sub>)<sub>3</sub> dose was used in the first and second ALD cycles to enhance the interface-forming reaction [3]. Finally, post deposition annealing was done at 400 °C for 2 minutes prior to the Au electrode deposition.

### 3. Experimental results and Discussion

Changes of the InGaAs surface by plasma cleaning/nitridation are shown in Fig. 2. N KLL and O KLL signals were normalized with respect to In MNN intensity. NH<sub>4</sub>OH-etched InGaAs is covered with the surface oxide as shown by the O KLL signal (Fig. 2(a)). H<sub>2</sub> plasma cleaning effectively removes this oxide layer (Fig. 2(b)). Plasma-nitrided surface clearly shows the N KLL signal which partly overlaps with In MNN (Fig. 2(c)). By combining the plasma cleaning and nitridation (Fig. 2(d)),

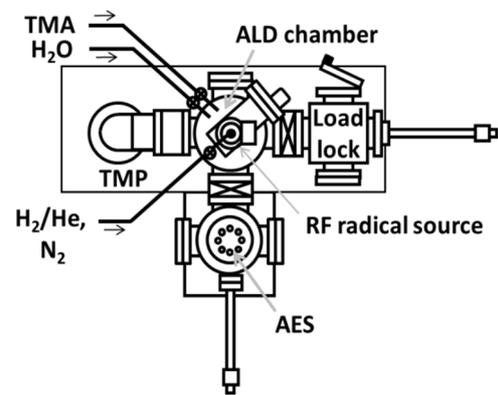


Fig. 1 Experimental setups of a high-vacuum compatible ALD system equipped with AES and remote RF plasma source.

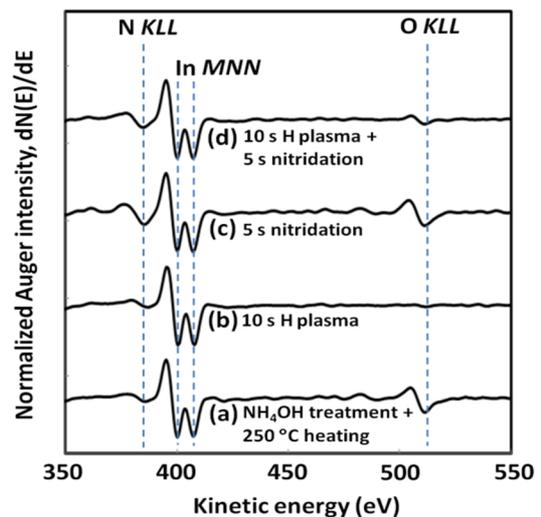


Fig. 2 AES spectra of InGaAs(100) substrate after various surface treatments at 250 °C.

the surface is covered mainly by a nitride layer containing a small amount of oxygen which might come from the residual oxygen in the present plasma chamber.

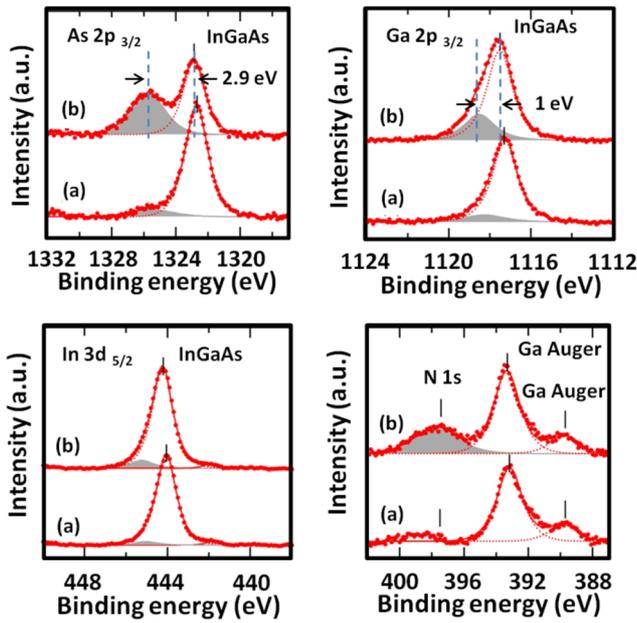


Fig. 3 Ga 2p, As 2p, In 3d and N 1s XPS spectra of Al<sub>2</sub>O<sub>3</sub> (~1 nm)/InGaAs. Initial InGaAs surfaces were treated at 250 °C by (a) 10 s H<sub>2</sub> plasma and (b) 10 s H<sub>2</sub> plasma and 5 s nitridation.

In order to probe the coverage and bonding states of N at the interface, the InGaAs surfaces with 1 nm-thick Al<sub>2</sub>O<sub>3</sub> cap layer were analyzed by XPS (Fig. 3). Spectra (a) and (b) in each plot are for the cases of plasma cleaning only and plasma cleaning/nitridation combination, respectively. N 1s peak, which is overlapped with the tail of the Ga Auger peak, is clearly observed for the nitride interface. The N coverage is estimated to be approximately 2 monolayer (~0.5 nm) from the ratio of N 1s to In 3d<sub>5/2</sub>. Ga 2p<sub>3/2</sub> and As 2p<sub>3/2</sub> peaks have a component with chemical shift of 1.0 and 2.9 eV, respectively. This result indicates that both Ga-N and As-N bonds exist at the lightly-nitrided Al<sub>2</sub>O<sub>3</sub>/InGaAs interface. The chemical shift for Ga 2p<sub>3/2</sub> agrees with that reported for the thick nitride [2], suggesting that N is the dominant anion at the interface.

The plasma cleaning and nitridation affect the MIS properties in distinct manners. Figure 4 compares the C-V characteristics for the Al<sub>2</sub>O<sub>3</sub>/InGaAs interfaces prepared with plasma cleaning only (a) and cleaning/nitridation combination (b). Reduction of the frequency dispersion under accumulation by plasma cleaning and nitridation indicates the smaller interface traps densities ( $D_{it}$ ). Table I summarized the MIS capacitor properties for various surface treatment conditions. The ratio of the 100 and 1 MHz capacitances by measuring at  $V_{fb}+1V$  was used to quantify the frequency dispersion under accumulation.  $D_{it}$  was estimated by high/low frequency method. The MIS interfaces with nitridation shows well-behaved characteristics with nearly ideal  $V_{fb}$  and low  $D_{it}$ . H<sub>2</sub> plasma cleaning without nitridation degraded the properties as evidenced by a large positive shift in  $V_{fb}$  and increases in  $D_{it}$  and frequency dispersion under accumulation. These degraded properties can be recovered by adding the nitridation treatment as seen in Table I.

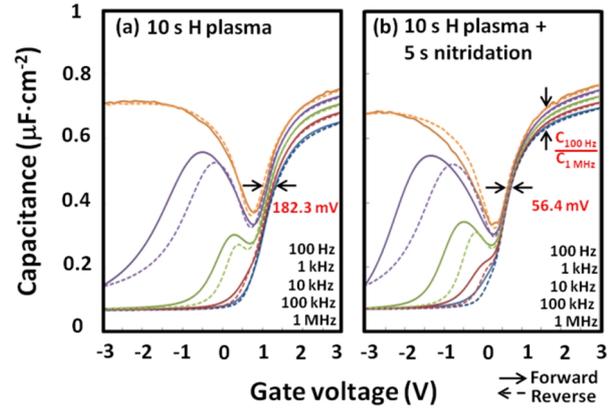


Fig. 4 C-V characteristics of Au/Al<sub>2</sub>O<sub>3</sub>(6 nm)/ n-InGaAs(100) capacitors with (a) 10 s H plasma and (b) 10 s H plasma with 5 s nitridation at 250 °C.

Table I. Summary of MIS capacitor properties.  $V_{fb}$  is calculated for the 1 MHz data (Ideal  $V_{fb} = +0.55$  V).  $D_{it}$  was estimated by high/low frequency method.

Conditions	$V_{fb}$ shift (V)	$D_{it}$ minimum ( $\times 10^{12}$ cm <sup>-2</sup> eV <sup>-1</sup> )	$C_{100\text{ Hz}}/C_{1\text{ MHz}}$ @ $V_{fb} + 1$ V
No plasma	+0.22	2.0	1.11
10 s H plasma	+0.47	3.5	1.62
5 s nitridation	0	1.3	1.10
10 s H plasma + 5 s nitridation	-0.07	1.2	1.09

## Conclusions

The Al<sub>2</sub>O<sub>3</sub>/InGaAs capacitors with ~2 monolayer nitride interfaces showed well-behaved C-V characteristics. The H<sub>2</sub> plasma cleaning, which effectively removed the surface oxides of InGaAs, degraded the electrical properties, whereas the subsequent nitridation restored the MIS characteristics.

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## References

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