

Ge Selective Growth in Micron and Submicron Trenches with UHV-CVD

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1. Introduction

Ge is an indispensable material in silicon photonics because its direct bandgap energy corresponds to 1.55 μm light and it can be processed with Si-CMOS technology. Recently many Ge optical devices have been reported, such as modulator[1], detector [2] and laser [3]. These Ge devices were made by selective epitaxial growth (SEG). SEG is an indispensable process for device fabrication. However, Ge-SEG is not well understood compared to III-V compounds[4] or Si [5]. Facet formation and growth species migration occur in SEG of these materials, so the epi-layer height is different from non-SEG layer. In Ge SEG, these phenomena should occur because Ge SEG is similar to Si SEG[6]. From these reasons the purpose of this research is to understand the mechanism of Ge-SEG in narrow areas, focusing on the Ge width and height, such as facet formation and growth species migration.

2. Experiments

We used a 4-inch (100) surface p-Si wafer, 525 μm thick with 150nm SiO₂ layer. Then we made trenches on the SiO₂ layer to expose Si with electron beam lithography and reactive ion etching (CHF₃). The trench width ranged from 0.5~100 μm , along with the <110> direction. At last we grew Ge by ultra-high-vacuum chemical vapor deposition (UHV-CVD) at 600°C for 110 min. using GeH₄ gas. Before the growth, we grew pure Ge at 370°C as a buffer layer. After the growth we observed the cross-sectional view of Ge by scanning electron microscopy (SEM).

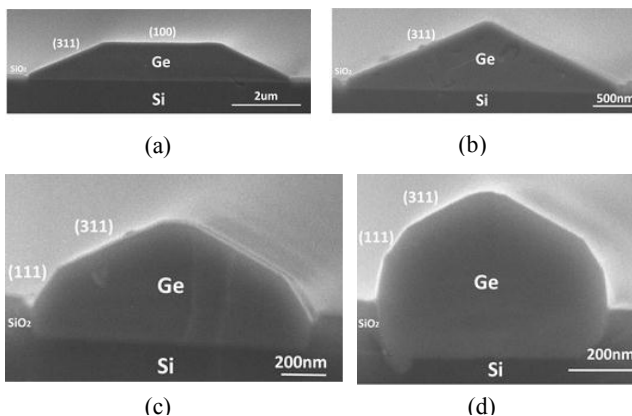


Fig. 1: Cross-sectional view of SEG-Ge by SEM

(a): 7.5 μm wide (b): 3.5 μm wide (c): 1.0 μm wide (d): 0.6 μm wide

3. Results

Fig. 1 shows that the Ge shape changes from mesa to tetragonal through triangle with narrowing Ge width, and Ge height becomes lower. Fig. 2 shows the Ge height as a function of the Ge width.

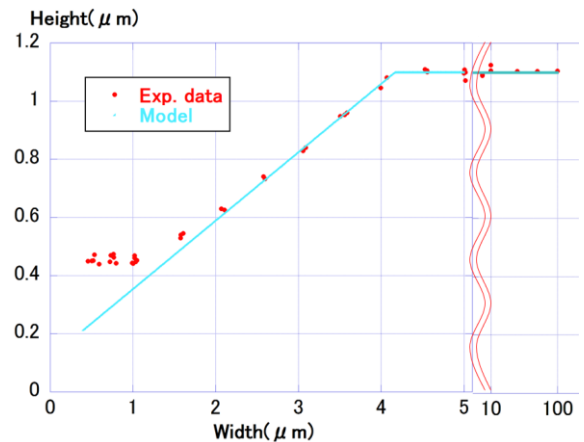


Fig. 2: Ge Width and Height

The Ge height is constant (1.1 μm) when the Ge width is wider than 4 μm . This is because the Ge shape is mesa in the wider cases and the height is defined by the mesa height. The mesa height is identical to that of the blanket Ge layer.

The Ge height decreases in proportion to the width despite of the same growth time when the Ge width is narrower than 4 μm . This is because the (100) plane disappears and the (311) facets dominate Ge height in the narrower cases. The Ge shape is triangle in the narrower cases and the triangle height is proportional to the width. Along with the triangle height, growth on (311) facet is also the factor to determine the Ge height. From the growth mechanism, we made a formula of Ge height.

$$H = \frac{\tan\theta_{311}}{2} \times W + \frac{GR_{311}}{\cos\theta_{311}} \times t. \quad (1)$$

H and W mean the Ge height and width respectively. θ_{311} is 25.2°, i.e., the angle between Si (100) surface and Ge (311) facet. GR_{311} is the growth rate on the (311) facet and t means the growth time. The solid line in Fig. 2 shows this formula, assuming 1.0 nm/min as GR_{311} . The formula fits the experimental data well when the width is wider than 2 μm , but the formula does not fit the data when the width is narrower than 2 μm . This is because we assumed that GR_{311} was constant in Eq. (1), but actually GR_{311} is faster in narrower cases.

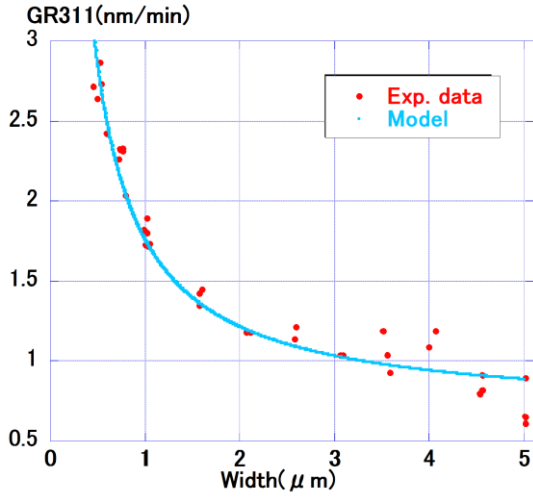


Fig. 3: Ge width and GR₃₁₁

Fig. 3 shows the Ge width and GR₃₁₁. We calculated GR₃₁₁ by dividing Ge thickness on (311) by growth time. This suggests that GR₃₁₁ would be inversely proportional to Ge width. Generally in CVD growth, growth rate is the same in all cases because chamber flux is constant. To explain Fig. 3, we made a model that Ge atoms or GeH_x migrated from SiO₂ in addition to the chamber flux. The trench in this model is 1cm long and W cm wide. We assumed that the SiO₂ height did not affect growth rate. In this experiment, the gap between trenches is 100μm so that it is much wider than trench width. Thus we assumed that the number of Ge atoms migrating from SiO₂ into the trench is constant (Ge_{SiO_2}). Since the trench area is $W \text{ cm}^2$ and the chamber flux is $F_{chamber} \text{ cm}^{-2}$, total Ge atoms (Ge_{all}) are

$$Ge_{all} = F_{chamber} \times W + Ge_{SiO_2}. \quad (2)$$

By dividing total Ge atoms by trench area ($W \text{ cm}^2$), we obtain the number of Ge atoms per width (Ge°) as below.

$$Ge^\circ = F_{chamber} + \frac{Ge_{SiO_2}}{W}. \quad (3)$$

In a high vacuum condition, GR₃₁₁ is proportional to GeH₄ pressure [7]. Thus we assumed that GR₃₁₁ is proportional to the number of Ge atoms on (311). From this assumption, we fit Eq. (3) to the experimental data and obtain GR₃₁₁ as below.

$$GR_{311(nm/min)} = 0.67 + \frac{1.09}{W}. \quad (4)$$

The solid line in Fig. 3 shows Eq. (4). This good fitting strongly suggests that the Ge migration model is right. We applied Eq. (4) to Eq. (1), and the final model is as Fig. 4.

4. Discussion

In Eq. (4), the first term corresponds to the chamber flux and the second term corresponds to Ge atoms from SiO₂. If the width is 0.8μm, the ratio is 1:2.

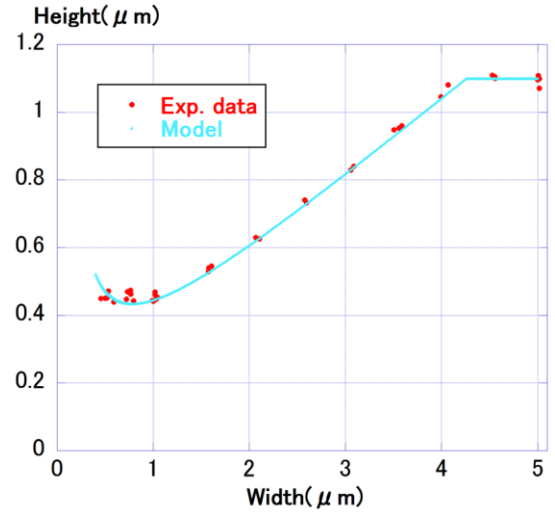


Fig.4: Final model of Ge height

This means that Ge atoms migrating from SiO₂ contribute to growth twice more than chamber flux in a 0.8μm trench. Thus we can say Ge from SiO₂ should not be negligible in submicron wide Ge-SEG with UHV-CVD. Since we need submicron scale Ge to fabricate single mode optical devices, we have to know the behavior of Ge on SiO₂.

In [8] the ratio of GR₁₀₀ : GR₃₁₁ is reported 1:0.58, but in our growth system the ratio is 1:0.1~0.3. We considered this is because the pressure of GeH₄ is lower than [8].

5. Conclusion

We have studied Ge SEG with UHV-CVD using SiO₂ mask. Ge height is lower in narrower trenches because of facet formation. In narrower trenches Ge height is higher than our prediction due to Ge migrating from SiO₂. The Ge migration model strongly suggests that Ge migrating from SiO₂ should not be negligible in submicron wide Ge-SEG with UHV-CVD. This is very important to fabricate submicron scale Ge devices.

Acknowledgement

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