High Performance Germanium n⁺/p Shallow Junction for the Scaled nMOSFET

Chen Wang, Cheng Li*, Wei Huang, Songyan Chen, Hongkai Lai

Department of Physics, Semiconductor Photonics Research Center, Xiamen University, Xiamen, Fujian 361005, People's Republic of China Phone: +86-13850087150, *E-mail: lich@xmu.edu.cn.

Abstract

In this work, we study excimer laser annealing (ELA) on phosphorus-implanted germanium with implantation energies and doses of 30 keV, 5×10^{15} cm⁻², and 10 keV, 5×10^{14} cm⁻², respectively. A well-behaved Ge n⁺/p shallow junction with a record rectification ratio of ~ 10^7 and low leakage current density of 8.3×10^{-5} A/cm² is achieved by a combination of low temperature pre-annealing (LTPA) and ELA, which is great beneficial to the scaled Ge nMOSFET technology.

1. Introduction

Germanium is a promising alternative channel material for extremely downscaled complementary metal oxide semiconductor (CMOS) technology due to its higher carrier mobility and lower processing temperature compared to silicon [1]. However, the difficulty to achieve a high activation n-type doping made it hard to realize excellent n^+/p shallow junctions for S/D in the scaled nMOSFET [2].

In this work, we investigate the effect of laser energy density on the phosphorus (P) diffusion in P-implanted Ge substrate and make high performance Ge n^+/p shallow junctions.

2. Experiments

A p-type Ge (100) wafer with a resistivity of 0.088 Ω ·cm was used in this study. Ge n⁺/p junctions were made by P⁺ implantation at 30 keV/5x10¹⁵ cm⁻² or 10 keV/5x10¹⁴ cm⁻², and one pulse ELA with or without LTPA process. Ge n⁺/p diodes and the contact of Al/n⁺Ge were fabricated by conventional etching and lift-off process. All of the contact electrodes were aluminum.

3. Results and Discussion

ELA on P-implanted Ge was studied firstly with the implantation energy and dose of 30keV, $5x10^{15}$ cm⁻². A significant diffusion of P after ELA at 200 and 300 mJ/cm² can be seen in Fig. 1. The amorphous Ge (a-Ge) induced by ion implantation can be efficiently re-crystallized after ELA at 200 mJ/cm² and above (Fig. 2). The contact resistivity (ρ_C) is low to $1.61x10^{-6} \Omega$ cm² (Fig. 3). Compared to our previous results for the samples annealed by rapid thermal annealing [3], the value of ρ_C is improved by about three orders of magnitude. Ge n⁺/p diodes (I_{on}/I_{off} ~ 2x10⁵ with an ideality factor η ~1.28) realized using ELA (Fig. 4). As laser fluence is increased from 0 to 250 mJ/cm², the reverse current I_{off} of n⁺/p diode decreases due to the improvement of crystallization (Fig. 4).

The fabricated n^+/p junction has a large junction depth with only ELA. A novel approach is proposed with a combination of LTPA and ELA to achieve Ge n^+/p shallow junction. The phosphorus implantation energy and dose is about 10 keV, $5x10^{14}$ cm⁻².

As shown in Fig. 5 and Fig. 6, the temperature of LTPA and the fluence of ELA are optimized by the J-V characteristics of Ge n^+/p junction diodes. The rectification ratio of Ge n⁺/p diodes and the $\rho_{\rm C}$ of Al/n⁺Ge are extracted (Fig. 7). A well-behaved Ge n^+/p junction with a record $I_{on}/I_{off} \sim 10^7$, low $I_{off} \sim 8.3 \times 10^{-5}$ A/cm² and an ideality factor η ~ 1.07 has been achieved when the samples are pre-annealed at 400 °C-10 min plus ELA at 150 mJ/cm². The SIMS profiles after one pulse ELA with or without LTPA can be well fitted by the diffusion model [4] (Fig. 9). The diffusion coefficient of phosphorus extracted in the samples after only ELA is about 3.5×10^{-4} cm²/s, and $2.9 \times 10^{-4} \text{ cm}^2/\text{s}$ for the samples with LTPA. It is worth noting that the LTPA process can significantly suppress the diffusion of P in Ge during ELA (Fig. 8 and Fig. 9). Moreover, the carrier concentration reaches to $6x10^{19}$ cm⁻³ and the junction depth is only 44 nm at 1×10^{18} cm⁻³ (Fig. 8). The TEM images of the samples before and after thermal treatments are shown in Fig. 10. After the sample LTPA at 400 °C-10 min, the implantation damages are healed preliminarily (Fig. 10 (b)), and plus ELA at 150 mJ/cm², no obvious defects can be observed in the film (Fig. 10 (c)).

4. Conclusion

High performance Ge n^+/p shallow junctions have been realized by using a combination of LTPA and ELA, which is immensely beneficial to the scaled Ge nMOSFET applications.

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Reference

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Fig. 1 Concentration profiles (SIMS) of as-implanted and laser annealed samples. (ion implantation $- P^+$, 30 keV, $5x10^{15}$ cm^{-2} , Laser Fluence – 200 mJ/cm², 300 mJ/cm^{2}).



Fig. 2 TEM Micrographs.(ion implantation - P^+ , 30 keV, $5x10^{15}$ cm⁻², Laser Fluence - 0 and 200 mJ/cm²).



Fig. 4 Effect of laser fuence on the diode's off current. I_{off} decreases and then saturates, $-P^+$, 30keV, 5x10¹⁵cm⁻², Laser Fluence -0, tion $-P^+$, 10 keV, 5x10¹⁴ cm⁻²). 150, 200 and 250 mJ/cm²).



Fig. 5 J-V characteristics of Ge n⁺/p junction diodes formed by ELA (150 mJ/cm²) and as laser fluence increases. (ion implantation LTPA at different conditions. (ion implanta-



Fig. 3 CTLM resistance measured as a function of CTLM contact pad spacing. The y-intercept is an indication of ρ_C for Al/n⁺Ge. (ion implantation- P⁺, 30 keV, 5×10^{15} cm⁻², Laser Fluence – 200, 250 and 300 mJ/cm^2).



Fig. 6 J-V characteristics of Ge n⁺/p junction diodes formed by ELA (100, 150, 200 and 250 mJ/cm²) with LTPA at 400 °C-10 min. (ion implantation $- P^+$, 10 keV, $5x10^{14} \text{ cm}^{-2}$).



Fig. 7 Rectification ratio of junction diodes as a function of laser fluence with or without LTPA at 400 °C-10 min. The inset represents the $\rho_{\rm C}$ of Al/n⁺Ge at different annealing conditions. (ion implantation $-P^+$, 10 keV, $5 \times 10^{14} \text{ cm}^{-2}$).



Fig. 8 Concentration profiles (SIMS and SRP) of as-implanted and laser annealed samples. (ion implantation $-P^+$, 10 keV, $5 \times 10^{14} \text{ cm}^{-2}$, Laser Fluence – 150 and 200 mJ/cm² with or without LTPA at 400 °C-10 min).



Fig. 9 Concentration profiles (SIMS) of as-implanted and laser annealed samples. (ion implantation $-P^+$, 10 keV, $5x10^{14}$ cm⁻², Laser Fluence – 150 and 200 mJ/cm² with or without LTPA at 400 °C-10 min). The continuous black lines represent best fits from the diffusion model [4].



Fig. 10 TEM Micrographs. (ion implantation $-P^+$, 10 keV, $5x10^{14}$ cm⁻²).