

## Effect of Nitridation for SiO<sub>2</sub>/SiC Interface on Defects Properties near Conduction Band Edge

◦W. Takeuchi<sup>1</sup>, K. Yamamoto<sup>2</sup>, M. Sakashita<sup>1</sup>, T. Kanemura<sup>2</sup>, O. Nakatsuka<sup>1</sup>, and S. Zaima<sup>1,3</sup>

<sup>1</sup>Graduate School of Engineering, Nagoya University,  
Furo-cho, Chikusa-ku, Nagoya 464-8603, Japan

Phone: +81-52-789-3819 E-mail: wtakeuti@alice.xtal.nagoya-u.ac.jp

<sup>2</sup>DENSO CORPORATION,

500-1 Minamiyama, Komenoki-cho Nisshin, Aichi 470-0111, Japan

<sup>3</sup>EcoTopia Science Institute, Nagoya University

Furo-cho, Chikusa-ku, Nagoya 464-8603, Japan

### Abstract

We have investigated the effect of NO-annealing for SiO<sub>2</sub>/4H-SiC interface properties. The electrical properties of NO-annealed sample are different from that of the wet-annealed sample. The NO-annealing generates positive charge in the insulator. The  $D_{it}$  from  $E_i+1.5$  eV ( $E_c-0.1$  eV) to the near the conduction band edge increases by the NO-annealing. The type of  $D_{it}$  near the conduction band edge in the nitrated SiO<sub>2</sub>/SiC interface is identified as donor. It is considered that Coulomb scattering causes decreasing the n-channel mobility in the nitridation of the SiO<sub>2</sub>/SiC interface by the NO-annealing.

### 1. Introduction

4H-SiC semiconductor is a wide bandgap semiconductor whose properties are well-suited for high-power applications. However, it is difficult to realize the ideal on-resistance of 4H-SiC MOSFET owing to its low channel mobility. It is considered that the mobility is limited by trapped inversion carriers at interface states and near the interface, which cause Coulomb scattering. In order to decrease the interface state density ( $D_{it}$ ), nitridation and wet annealing processes for a SiO<sub>2</sub>/SiC interface are often performed. The wet annealing process decreases the  $D_{it}$  near the conduction edge and the electron mobility increased up to 244 cm<sup>2</sup>/Vs [1]. Also, the nitridation annealing process improves an SiO<sub>2</sub>/SiC interface properties, being effective for increasing the channel electron mobility as high as 100 cm<sup>2</sup>/Vs [2]. However, in the channel mobility at a nitrated interface is still much lower than a value expected from the bulk electron mobility of 4H-SiC (> 800 cm<sup>2</sup>/Vs) [3]. To further improve the mobility, it is important to reveal mobility-limiting factors at the nitridation SiO<sub>2</sub>/SiC interface. In this study, we investigated the effect of NO-annealing on SiO<sub>2</sub>/SiC interface properties compared with wet-annealed samples.

### 2. Experimental

Gate SiO<sub>2</sub> layers were formed by using low pressure chemical vapor deposition and wet-oxidation methods on n-type 4H-SiC(11-20) face substrate with an n-type epitax-

ial layer (epitaxial layer thickness: 15 μm, electron concentration: 4-6 × 10<sup>15</sup> cm<sup>-3</sup>). These SiO<sub>2</sub> thicknesses are ranging from 75 nm to 100 nm. After the formation of gate SiO<sub>2</sub> layers, annealing was performed under the NO gas atmosphere or the wet atmosphere. Here, the wet and NO atmosphere annealing are referred to as Depo.+Wet and Depo.+NO, respectively. Also, the sample of wet atmosphere annealing for the wet-oxidized SiO<sub>2</sub>/SiC sample is named Oxidation+Wet. In order to estimate the  $D_{it}$ , MOS capacitors were fabricated with a Ti / n<sup>+</sup>-type poly-Si electrode on the gate SiO<sub>2</sub> layer. The  $D_{it}$  of the MOS capacitor was estimated by using the conductance method, Gray-Brown method, and deep level transient spectroscopy (DLTS) method. Surface potential fluctuation (SPF) was also estimated by using the conductance method.

### 3. Results and discussion

Figure 1 shows capacitance-voltage (C-V) characteristics of MOS capacitors prepared with various conditions. These C-V curves of Depo.+Wet and Oxidation+Wet samples show similar characteristics. The flatband voltage ( $V_{fb}$ ) of wet-annealed samples was estimated to be 4.7 V, which shift to positive side compared with the ideal- $V_{fb}$  (0.2 V).

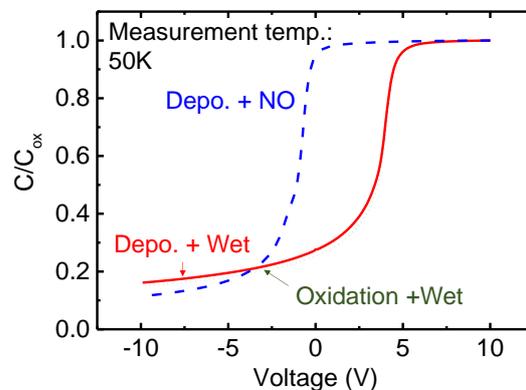
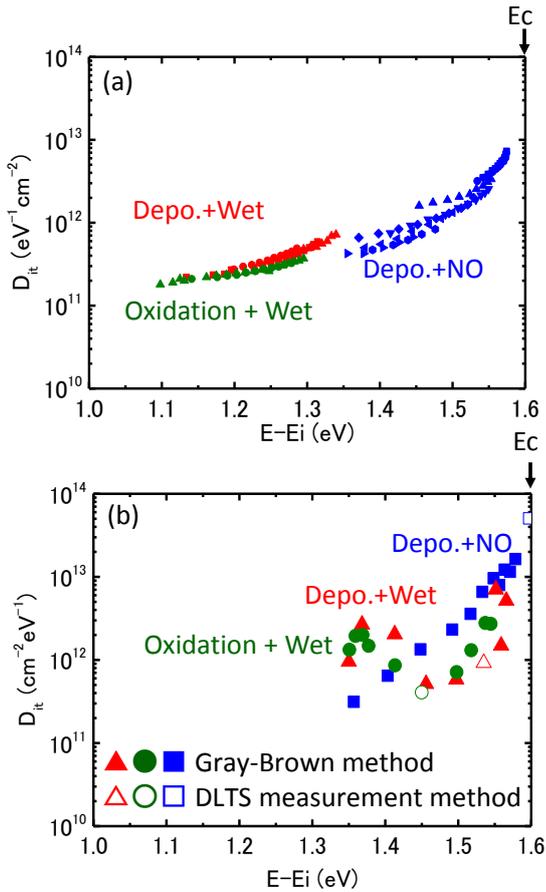


Fig. 1. C-V characteristics of MOS capacitors with wet- and NO-annealing for deposited- and wet-oxidized-SiO<sub>2</sub> samples measured at 50 K.

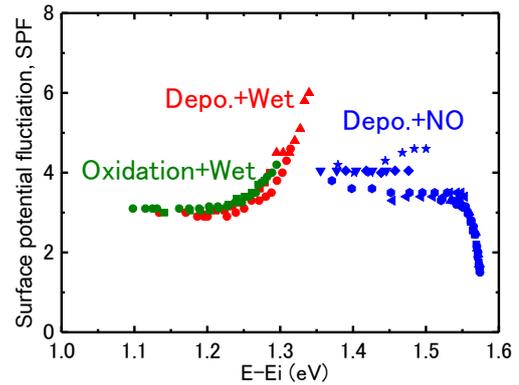


**Fig. 2.** Interface state densities evaluated by using (a) the conductance method (b) Gray-Brown and DLTS methods.

This means that the wet-annealing generates positive charges. In contrast, the C-V characteristic of Depo.+NO sample is different from the wet-annealed samples. The  $V_{fb}$  of NO-annealed sample was estimated to be  $-0.35$  V. This means that the NO-annealing generates positive charges.

Figures 2 (a) and 2(b) show the  $D_{it}$  distributions estimated by the conductance method and Gray-Brown and DLTS measurement methods, respectively. The  $D_{it}$  distributions of wet-annealed samples are similar, meaning that characteristics of the deposited- and the oxidized-SiO<sub>2</sub> are not different after wet-annealing. In the case of Depo.+NO sample, the  $D_{it}$  distribution estimated by the conductance method coincides with that of Gray-Brown and DLTS results. On the other hand, the  $D_{it}$  distribution of the wet-annealed sample is different from that of NO-annealed sample. The  $D_{it}$  of the NO-annealed sample from  $E_i+1.5$  eV ( $E_c-0.1$  eV) to near the conduction band edge is higher than that of wet annealed samples.

Figure 3 shows the energy distribution of the surface potential fluctuation (SPF) of MOS capacitors. In the wet-annealed samples, the SPF increases toward the conduction band edge. This means that the type of  $D_{it}$  is identified as acceptor [4]. In contrast, the SPF of NO-annealed sample decreases toward the conduction band edge. This means that the type of  $D_{it}$  is donor.



**Fig. 3.** Energy distribution of the surface potential fluctuation (SPF) on MOS capacitors with Wet- and NO-annealing for deposited- and wet-oxidized- SiO<sub>2</sub> samples.

It is well known that nitrogen is introduced at the SiO<sub>2</sub>/SiC interface by the NO-annealing [5]. The donor level of nitrogen in 4H-SiC is  $E_c-0.05$  eV on the hexagonal site and  $E_c-0.10$  eV on quasicubic site [6]. Thus, it is considered that increasing the  $D_{it}$  in the NO-annealed sample would be caused by the nitrogen introduction. In the case of NO-annealing, the mobility might be limited by Coulomb scattering which is caused by charges from interface states near the conduction band edge at the nitrated SiO<sub>2</sub>/ SiC interface.

#### 4. Conclusions

We investigated the effect of NO- and wet-annealing on SiO<sub>2</sub>/4H-SiC interface properties. The characteristics of the deposited- and the oxidized-SiO<sub>2</sub> samples are not different after wet-annealing. The wet- and NO-annealing cause the voltage-shift of C-V characteristics with negative and positive charges, respectively. The  $D_{it}$  in NO-annealed samples around  $E_i+1.5$  eV ( $E_c-0.1$  eV) and near the conduction band edge is higher than that of the wet-annealed sample. The  $D_{it}$  type of wet- and NO-annealing are identified as acceptor and donor types, respectively. Thus, it is considered that Coulomb scattering causes decreasing the n-channel mobility in the nitridation of the SiO<sub>2</sub>/ SiC interface by the NO-annealing.

#### References

- [1] T. Endo et al., *Materials Science Forum* **600**, 691 (2009).
- [2] K. Yamamoto et al., *Materials Science Forum* 821-823 (2015) pp 713-716.
- [3] H. Matsunami and T. Kimoto, *Mater. Sci. Eng.* **R20**, 125 (1997).
- [4] E. H. Nicollian and J. R. Brews, *MOS (Metal Oxide Semi-conductor) Physics and Technology*, Wiley, New York, 1982, pp. 307-311.
- [5] K. McDonald et al., *J. Appl. Phys.* **93**, 2257 (2003).
- [6] A.O. Ewvaraye et al., *J. Appl. Phys.* **79**, 7726 (1996).