Low power switching of Si-doped Ta₂O₅ ReRAM for high density memory application

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Abstract

We report, for the first time, the resistive switching properties of Si-doped Ta_2O_5 grown by atomic layer deposition. The low power operation is successfully demonstrated in the Si-doped Ta_2O_5 ReRAM devices of 50nm tech node. The switching mechanism for the Si-doped Ta_2O_5 resistor is discussed. Si dopants enable a switching layer to have conformal distribution of oxygen vacancy and easily form conductive filament.

1. Introduction

In order to be more competitive for high density memory application, ReRAM cells are required to low operation current below 10µA [1]. Ta₂O₅ is the widely used material for ReRAM due to its good endurance [2-3]. However, the operation current is still high level $(50-100\mu A)$. Recently, it is reported that the good switching performance was obtained using defect engineering such as oxygen reduction or impurity implantation [4-5]. However, these defect engineering technologies may have fundamental limitations for application of multi-plane and 3D vertical type structures. Thus, the defect engineering process during the film deposition is much more desirable. We predict that silicon doping would be advantageous to oxygen vacancy (V_0) formation in Ta₂O₅ film because of the chemical valence difference between Ta⁵⁺ and Si⁴⁺ [5]. In this study, we demonstrated the switching properties of Si-doped Ta₂O₅ using atomic layer deposition (ALD). Moreover, oxygen reduction of Ta₂O₅ by hydrogen plasma treatment (HP) was conducted as a comparison.

2. Device Fabrication and experiment

The amount of Si doping in Ta_2O_5 is controlled by the ratio of Ta_2O_5 and SiO₂ ALD subcycles (Fig. 1). Fig. 2 shows the process flow of the fabricated ReRAM structure with 50nm tech node. Ti_4O_7 is taken for a reservoir layer. Ta_2O_5 reduced by hydrogen plasma treatment (Ta_2O_5 -HP) was also formed to compare with Si-doped Ta_2O_5 .

3. Results and discussion

Table I shows the composition of Si-doped Ta_2O_5 with Ta_2O_5/SiO_2 subcycle ratios. Fig. 3(a) shows XPS Ta5f spectra. In case of Ta_2O_5 -HP, the surface of Ta_2O_5 was reduced. However, there are no noticeable changes in Si-doped Ta_2O_5 with the increase of Si doping concentration. On the other hand, for Si2p XPS spectra as shown in Fig. 3(b), SiO_x was formed and the peak intensity of SiO_x increased as the increase of Si content. This result in turn

confirms the idea that oxygen vacancies can be generated by utilizing ALD SiO_2 and Ta_2O_5 subcycles. Thermodynamic approach explains that Ta could consume oxygen from SiO_2 and therefore form SiO_x having V_0 (Eq. (1), (2)).

DC I-V switching characteristics as a function of compliance current (CC) were measured for the Ta₂O₅-HP and Si-doped (2.4% atomic percent) Ta₂O₅ resistor stacks. In case of Ta₂O₅-HP, on/off current ratio (I_{on}/I_{off}) of 4.4 was achieved at 20 μ A CC but resistive switching did not occur at lower CC, 5 μ A (Fig. 4). Surprisingly, Si-doped Ta₂O₅ stack shows the improved switching (I_{on}/I_{off} >5) not only at 20 μ A CC but also at 5 μ A CC (Fig. 5-6). In addition, the endurance property was not degraded up to 80000 cycles, as shown in Fig. 7.

From linear fitting of high resistance state (HRS), Si-doped Ta₂O₅ shows the better linearity of Poole-Frenkel emission fitting than Ta₂O₅-HP (Fig. 8, 9). Moreover, Si-doped Ta₂O₅ has higher low resistance state (LRS) current than Ta₂O₅-HP, while there is little difference in HRS currents (Fig. 10). This indicates that a conductive filament in Si-doped Ta₂O₅ film is easily formed during set-process. In case of Ta₂O₅-HP stack, it is difficult to form filament due to insufficient V_o drift energy at low operation current (CC <20µA) (Fig. 11(a)). On the other hand, for Si-doped Ta₂O₅ stack, V_o migrates not only from Ti₄O₇ but also from Si-doped Ta₂O₅ films (Fig. 11(b)). Thus, Si-doped Ta₂O₅ could form filament and increase on/off current ratio at even low current.

4. Conclusion

We reported, for the first time, the switching behaviors of Si-doped Ta₂O₅ ReRAM grown by ALD. It exhibited low power switching (I_{on}/I_{off} >5 at 5 μ A CC) and good endurance (>8000cycles). Systematic analysis clearly explains the switching mechanism of Si-doped Ta₂O₅ that the conformal distribution of V_o in Si-doped Ta₂O₅ film allows a boost in on/off ratio and operation current scaling, which makes this ALD Si-doped Ta₂O₅ promising for future high density and 3D type ReRAM applications.

References

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Fig. 3. XPS spectra of Ta_2O_5 , Si-doped Ta_2O_5 , and Ta_2O_5 -HP. (a) Ta 5f, (b) Si 2p.

TABLE I. The composition of Si doped Ta_2O_5 with Ta_2O_5 and SiO_2 subcycle ratio.

TaO/SiO Cycle Ratio	Atomic percent (%)		
	0	Si	Та
2:1	71.9	6.6	21.5
5:1	73.1	2.4	24.5
8:1	72.9	1.6	25.5



Fig. 4. The switching results after the fifth I-V sweeps for Ta_2O_5 -HP with compliance current (CC). (a) 20 μ A CC, (b) 5 μ A CC. On/off current ratio was measured at 0.7V.



Fig. 6. On/off current ratios as a function of CC for Si-doped Ta_2O_5 and $Ta_2O_5\text{-HP}$ cells.



Fig. 8. Linear fitting of HRS from 5th cycle of switching I-V. (a) Ta_2O_5 -HP, (b) Si doped Ta_2O_5 .



AC Cycle Number

Fig. 7. AC endurance characteristics of Si-doped $Ta_2O_5\ stack.$



Fig. 9. Schematic band diagrams of (a) Ta_2O_5 -HP and (b) Si-doped Ta_2O_5 stack in HRS.



Fig. 2. (a) Crosssection TEM image of the fabricated ReRAM structure and (b) process flow. The bottom electrode contact (BEC) size is 50 nm





Fig. 5. The switching results after the fifth I-V sweeps for Si-doped (2.4% atomic percent) Ta_2O_5 with CC. (a) 20 μ A CC, (b) 5 μ A CC.



Current@0.7V (µA)

Fig. 10. HRS and LRS currents of Si doped Ta_2O_5 and Ta_2O_5 -HP cells.



Fig. 11. Switching mechanism (a) Ta_2O_5 -HP resistor, (b) Si doped Ta_2O_5 resistor.