

## Influence of Oxygen Partial Pressure on Electrical and Magnetic Characteristics of Spinel CoFe<sub>2</sub>O<sub>4</sub> Thin Films

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### Abstract

Recently, semiconducting spinel ferrites have attracted increasing attention for the potential as room temperature spintronics devices. In this study, we investigated the effect of O<sub>2</sub> atmosphere during depositions on CoFe<sub>2</sub>O<sub>4</sub> film for controlling the electrical transport property. The electrical and magnetic properties were evaluated for the CoFe<sub>2</sub>O<sub>4</sub> films fabricated in various O<sub>2</sub> partial pressures.

### 1. Introduction

Spinel ferrite is one of the most promising materials in the spintronics field owing to half-metallicity, spin filter effect and high Curie temperature. Since such ferrites, CoFe<sub>2</sub>O<sub>4</sub>, MnFe<sub>2</sub>O<sub>4</sub> and ZnFe<sub>2</sub>O<sub>4</sub> etc., are the electric insulators, the uses in the spintronics are restricted. If we succeed in controlling the electric conductivity, it extends potential as room temperature spintronics devices. In recent research, semiconducting characteristics of the spinel ferrites were realized by varying composition of cations[1][2][3]. On the other hands, NiFe<sub>2</sub>O<sub>4</sub> grown by magnetron sputtering in Ar atmosphere (without oxygen gas) showed low resistivity [4]. Such low resistivity could be attributed to the coexisting of Fe<sup>2+</sup> and Fe<sup>3+</sup> in B sublattices, which caused by composition modulation or oxygen defects. Therefore, sophisticated controls of the O<sub>2</sub> partial pressure during the deposition is considered to realize fine-tuning of the resistivity in the spinel ferrite. However, few researches about controlling electrical properties by O<sub>2</sub> pressure have been reported so far. Furthermore, in previous researches, spinel ferrites were grown by PLD or sputtering from "oxide sources". It is considered to be difficult to control the oxidation states of the films by such fabrication technique.

In this research, we fabricated CoFe<sub>2</sub>O<sub>4</sub> film, which is ferrimagnetic insulator originally, by reactive molecular beam epitaxy method in various O<sub>2</sub> partial pressure using metal Fe and Co sources. This method is expected to realize the control of the oxidation state accurately, leading fine-tuning of the transport property. We also investigated the magnetic characteristics of the films because the oxygen defects may affects magnetic property.

### 2. Experiments

CoFe<sub>2</sub>O<sub>4</sub> films were grown on MgO(100) substrate by reactive molecular beam epitaxy method (Base Pressure : ~10<sup>-8</sup> Pa). The film structures were MgO(100)/CoFe<sub>2</sub>O<sub>4</sub>(30 or 50 nm)/Al<sub>2</sub>O<sub>3</sub>(3 nm). CoFe<sub>2</sub>O<sub>4</sub> thin films were grown by co-evaporation of Fe and Co at 300°C in O<sub>2</sub> atmosphere(0.1~4×10<sup>-4</sup> Pa). Growth rate of CoFe<sub>2</sub>O<sub>4</sub> was 0.4 Å/s. Al<sub>2</sub>O<sub>3</sub> was deposited at room temperature for preventing excess oxidation of CoFe<sub>2</sub>O<sub>4</sub> in the air. The epitaxial growth and surface morphology was observed by RHEED and AFM. We measured electrical resistivity by Van der Pauw method and four terminal method. We confirmed magnetization process by MOKE measurement. The oxidation states and composition ratio of the CoFe<sub>2</sub>O<sub>4</sub> were estimated by XPS. We determined charge carrier from Seebeck coefficient.

### 3. Results and discussion

Fig. 1 shows the RHEED images for (a) 4×10<sup>-4</sup> Pa, and (b) 3×10<sup>-5</sup> Pa in the [001] azimuth. The CoFe<sub>2</sub>O<sub>4</sub> film showed typical diffraction patterns from the (100) plane of the spinel structure. The sharp streaks suggest high crystallinity and flat surface in both films. The diffraction patterns of the films grown under 2×10<sup>-5</sup> Pa were spotty and were not like the spinel structure (not shown). From AFM measurements, the surface roughness was less than 0.3 nm.

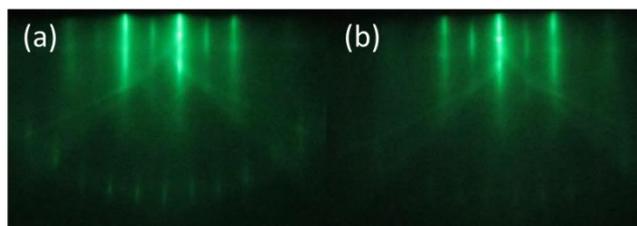


Fig.1. RHEED patterns for heteroepitaxial growth of CoFe<sub>2</sub>O<sub>4</sub> thin films on MgO (100) in O<sub>2</sub> atmosphere (a) 4×10<sup>-4</sup> Pa (b) 3×10<sup>-5</sup> Pa

Fig.2 shows XPS results for the films grown in various  $O_2$  pressure. Peak shift in Fig.2.(a) suggests that the deposition under low oxygen pressure yielded increase of  $Fe^{2+}/Fe^{3+}$  ratio. Fig.2.(b) exhibited clear peak shifts, which suggests the presence of the un-oxidized metallic Co. These results mean that the oxidation states are controllable by  $O_2$  atmosphere during the deposition. Composition ratio between Fe and Co was also evaluated from the spectra. The result is Fe:Co = 1:1.92~1.96. Since composition ratio was constant, change of electrical resistivity came from the  $O_2$  pressure, not from the composition fluctuation.

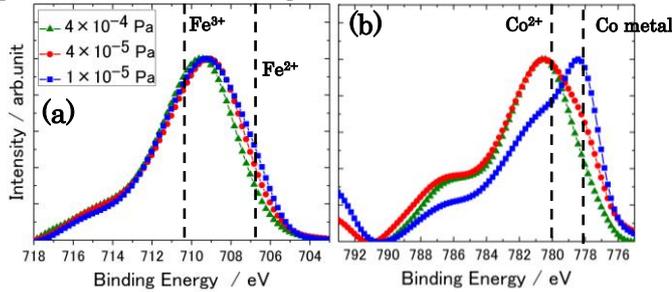


Fig.2. XPS spectra of (a) Fe 2p<sub>3/2</sub> (b) Co 2p<sub>3/2</sub> core level for the CoFe<sub>2</sub>O<sub>4</sub> films.

Fig.3 shows  $O_2$  partial pressure dependence of electrical resistivity of the films. CoFe<sub>2</sub>O<sub>4</sub> films grown in low  $O_2$  pressure exhibited low resistivity. The tendency indicates that the electrical transport is enhanced by oxygen defects, namely coexisting of  $Fe^{2+}$  and  $Fe^{3+}$  in B sublattices or insufficient oxidation of Co.

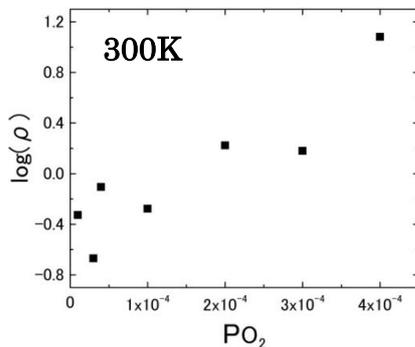


Fig.3.  $O_2$  partial pressure dependence of Resistivity of the CoFe<sub>2</sub>O<sub>4</sub> films grown in various  $O_2$  pressures

Fig.4 shows (a) temperature dependence of resistivity  $\rho$  of the CoFe<sub>2</sub>O<sub>4</sub> films grown in  $4 \times 10^{-4}$  Pa and (b) the Arrhenius plot. Semiconducting behavior was observed. The activation energy was estimated at 55 meV from the Arrhenius plot. That value is rather small compared to the previous study in which the films were fabricated by PLD[1]

Seebeck measurements were also carried out to determine the carrier type. The Seebeck coefficient for films grown in  $4 \times 10^{-5}$  Pa and  $3 \times 10^{-5}$  Pa were  $-140.6 \mu V / K$  and  $-164.4 \mu V / K$ , respectively. Negative value means that charge carrier was electron. Increment of Seebeck coefficients implied increasing of electron density.

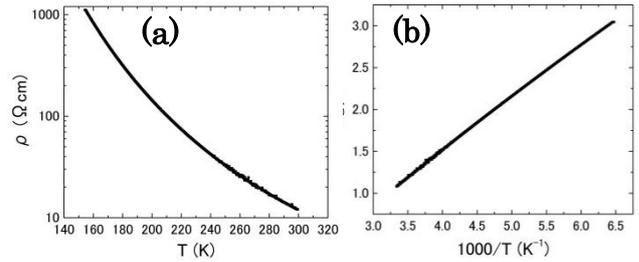


Fig.4. (a) Temperature dependence of resistivity  $\rho$  of the CoFe<sub>2</sub>O<sub>4</sub> films grown in  $4 \times 10^{-4}$  Pa. (b) The  $\rho$  vs  $1000/T$  plots are fitted to the Arrhenius law

Fig.5 shows magnetic hysteresis loop measured by MOKE in out-of-plane magnetic field (1.6T). The in-plane MOKE measurements in magnetic field of 0.2T showed no hysteresis loop (not shown). Although in-plane magnetic field was not sufficient, these results suggest that CoFe<sub>2</sub>O<sub>4</sub> films have perpendicular magnetic anisotropy. All the loops in Fig. 5 have similar shapes, so that the magnetic property was not affected by  $O_2$  partial pressure during deposition.

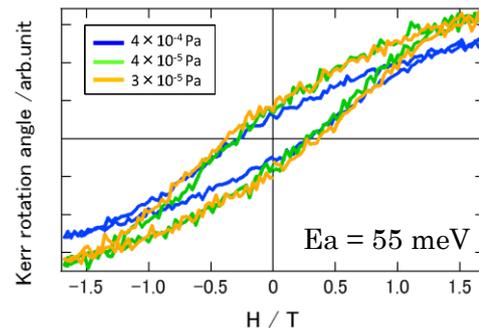


Fig.5.  $O_2$  partial pressure dependence of Resistivity of the CoFe<sub>2</sub>O<sub>4</sub> films grown in various  $O_2$  pressures

#### 4. Summary

We fabricated CoFe<sub>2</sub>O<sub>4</sub> films in various  $O_2$  pressure. From RHEED and AFM measurements, CoFe<sub>2</sub>O<sub>4</sub> films have high crystallinity and vary flat surface. The films grown in low  $O_2$  pressure showed low electrical resistivity, which is attributed to the presence of  $Fe^{2+}$  and metal Co. The temperature dependence of the resistivity obeyed the Arrhenius law. Such semiconductive transport properties extend the potential of spinel ferrites as new spintronics materials.

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#### References

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