# Electrical property of single-phase HMS grown by temperature gradient solution growth method using Ga and Sn solvent

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# Abstract

We have investigated the electrical property of single-phase HMS crystals grown by a temperature gradient solution growth (TGSG) method using Ga and Sn solvent. We found the clear difference of resistivity between  $Mn_{11}Si_{19}$  and  $Mn_4Si_7$ , which would be caused by the difference in their electronic structure. The hole concentration of  $Mn_{11}Si_{19}$  was approximately one order of magnitude higher than that of  $Mn_4Si_7$ . This experimental result supports the theoretical prediction that the Fermi level of only  $Mn_4Si_7$  stays in the energy gap while those of the other HMSs, namely  $Mn_{11}Si_{19}$ ,  $Mn_{15}Si_{26}$ , and  $Mn_{27}S_{47}$ , stay inside the valence band.

#### 1. Introduction

Higher manganese silicides (HMS)s have attracted attention as interesting materials for the thermoelectric devices and infrared optical detection devices because they have a large Seebeck coefficient, high chemical stability and narrow band gap energy [1-3]. There are at least four different tetragonal phases in HMSs, including  $Mn_4Si_7$ ,  $Mn_{11}Si_{19}$ ,  $Mn_{15}Si_{26}$ , and  $Mn_{27}Si_{47}$  [1]. Migas et al., studied electronic structures of these HMSs by the ab initio calculation and predicted that only the Fermi level of  $Mn_4Si_7$ stays in the energy gap while those of the other HMSs, namely  $Mn_{11}Si_{19}$ ,  $Mn_{15}Si_{26}$ , and  $Mn_{27}S_{47}$ , stay inside the valence band regardless of the presence of the similar energy gaps (0.77 – 0.78 eV) [1].

In addition, a number of experimental studies on the band gap nature and electrical property of each HMS phase have been carried out by optical and electrical measurements. Iioka et al., investigated the optical absorption of single phase  $Mn_{11}Si_{19}$  and found the indirect transition type absorption with the energy of approximately 0.64 eV at 5 K [3]. Nishida and Kawasumi et al. reported the anisotropy of the electrical property [4, 5]. However, difference in electrical property of each HMS phase is still unclear because of the difficulty in obtaining single phase HMS crystal.

Recently, we have succeeded in growing single-phase  $Mn_{11}Si_{19}$  and  $Mn_4Si_7$  bulk crystal by the temperature gradient solution growth (TGSG) method using Ga and Sn solvent, respectively [3, 6]. In this paper, we investigate the electric property of single-phase  $Mn_{11}Si_{19}$  and  $Mn_4Si_7$  crystals by Hall effect measurement between 5 and 300 K.

# 2. Experimental procedures

Crystal growth of single-phase Mn<sub>11</sub>Si<sub>19</sub> and Mn<sub>4</sub>Si<sub>7</sub> bulk was carried out by the TGSG method using Ga and Sn solvent, respectively. The solute of Mn-Si alloy ingot was pre-synthesized by melting the certain amount of Mn (3N) and Si (10N) (Mn:Si = 1:1.7) in a purified quartz ampoule. Then, the alloy solute and Ga (6N) or Sn (5N) solvent were encapsulated in a purified quartz ampoule with a graphite template under a high vacuum ( $< 5 \times 10^{-6}$  Torr) condition. Inner diameter of the template and ampoule were 15 mm and 16 mm, respectively. The thickness of the solvent, i.e. distance between the alloy ingot and the template, was fixed at 10 mm. The encapsulated growth ampoule was placed in a resistive heating vertical furnace with temperature gradient of  $\Delta T = 40$  °C/cm. Typical growth temperature and dissolution temperature were 900  $^{\circ}$ C and 950  $^{\circ}$ C, respectively. Typical growth period was 1 week (168hours). After growth, the Ga and Sn solvent were removed by aqua regia.

Electrical property of the grown crystals was evaluated by the Hall effect measurement and the van der Pauw method using the Physical Properties Measurement System (PPMS, Quantum Design). The measured temperature range and applied magnetic field were 5 - 300 K and 0 - 5 T, respectively.

# 3. Results and discussion

Figs.1 (a) and (b) show a typical single-phase  $Mn_{11}Si_{19}$  crystal grown from Ga solvent. The crystal was about 14mm in diameter and about 0.8-1.4mm in thickness. Figs. 2 (a) and (b) show a typical single-phase  $Mn_4Si_7$  crystal grown from Sn solvent. The crystal diameter was identical with the  $Mn_{11}Si_{19}$  crystal grown from Ga solvent and the thickness was about 0.3mm.



Fig.1. Top view (a) and side view (b) of typical Mn<sub>11</sub>Si<sub>19</sub> crystal grown from Ga solvent.



Fig.2. Top view (a) and side view (b) of typical  $Mn_4Si_7$  crystal grown from Sn solvent.

Fig. 3 shows the temperature dependence of electrical resistivity of  $Mn_{11}Si_{19}$  (sample #1 and #2) and  $Mn_4Si_7$  (sample #3 and #4) crystals. The resistivity of #1 and #2 ( $Mn_{11}Si_{19}$ ) was lower than that of #3 and #4 ( $Mn_4Si_7$ ) in measured temperature range, although all samples showed metallic temperature dependence. The resistivity of  $Mn_{11}Si_{19}$  at 300K was 1.2 x 10<sup>-3</sup> and 1.1 x 10<sup>-3</sup> for #1 and #2, respectively. The values are close to our previous report (0.9 - 1.1 x 10<sup>-3</sup> \Omega cm)[6].

Figs. 4 (a) and (b) show the temperature dependence of hole concentration and Hall mobility of  $Mn_{11}Si_{19}$  and  $Mn_4Si_7$  crystals, respectively. Interestingly, clear difference of hole concentration was found between  $Mn_{11}Si_{19}$  (#1 and #2) and  $Mn_4Si_7$  (#3 and #4). The hole concentration of  $Mn_{11}Si_{19}$  was (1.6 - 3.6) x  $10^{21}$  cm<sup>-3</sup>, whereas that of  $Mn_4Si_7$  was (1.1 - 2.7) x  $10^{20}$  cm<sup>-3</sup>. The big difference of electronic structure between  $Mn_{11}Si_{19}$  and  $Mn_4Si_7$  as predicted by ab-initio calculation [1]. The Hall mobility was 1.6 - 2.3 cm<sup>2</sup>/Vs for  $Mn_{11}Si_{19}$  and 2.4 - 4.1 cm<sup>2</sup>/Vs for  $Mn_4Si_7$  at 300K. The values are close to each other. Therefor, the difference of resistivity between  $Mn_{11}Si_{19}$  and  $Mn_4Si_7$  is mainly due to the difference of hole concentration in each crystal.



Fig.3 Temperature dependence of resistivity of single-phase  $Mn_{11}Si_{19}$  and  $Mn_4Si_7$  crystals.



Fig.4 Temperature dependence of (a) hole concentration and (b) Hall mobility of single-phase  $Mn_{11}Si_{19}$  and  $Mn_4Si_7$  crystals.

### 4. Conclusion

We have studied the electrical property of single-phase  $Mn_4Si_7$  and  $Mn_{11}Si_{19}$  crystals grown by the temperature gradient solution growth (TGSG) method using Ga and Sn solvent, respectively. Clear difference of resistivity that would be caused by the difference in their electronic structure was found between  $Mn_{11}Si_{19}$  and  $Mn_4Si_7$  crystals. The hole concentration of  $Mn_{11}Si_{19}$  was approximately one order of magnitude higher than that of  $Mn_4Si_7$ . The higher hole concentration in  $Mn_{11}Si_{19}$  supports the theoretical results that the Fermi level of only  $Mn_4Si_7$  stays in the energy gap while those of the other HMSs, namely  $Mn_{11}Si_{19}$ ,  $Mn_{15}Si_{26}$ , and  $Mn_{27}S_{47}$ , stay inside the valence band.

### Acknowledgements

This work was partially supported by the Grant-in-Aid for Scientific Research (B), Japan Society for the Promotion of Science (JSPS).

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