# A Study of Impact of Incorporating Locations of Yttrium on the Properties of GeO<sub>x</sub>/HfO<sub>2</sub>-based/TiN Gate Stack Based on Material Reaction

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#### Abstract

In this paper, an influence of incorporating location of Yttrium (Y) into GeO<sub>x</sub> interfacial layer (IL) with HfO<sub>2</sub>-based/TiN gate stack on p-Ge was studied. Various incorporating Y processes presented contrasting electrical and material results. Depositing Y metal before IL formation improved interface quality, however, degraded the dielectric constant ( $\kappa$ ) of the HfO<sub>2</sub>. We found non-fully oxidized Y atoms would capture nearby O atoms in Ge-O and/or Hf-O bonds, which depended on the position of Y. A series of XPS analyses indicated that Y doped GeO<sub>x</sub> (YGO) induced Hf-N bonds in HfO<sub>2</sub> with N<sub>2</sub> plasma treatment, which degraded the  $\kappa$  of HfO<sub>2</sub>.

### 1. Introduction

For further scaling MOSFET, Ge is one of high-mobility materials for enhancing the device on current. However, an essential challenge for Ge devices is how to form a high-quality interfacial layer (IL) on Ge. C. Lu *et al.* proposed a potential solution to solve the interface issue by doping Yttrium (Y) into GeO<sub>2</sub> (YGO) [1]. Based on the values of Gibbs free energy in Table I, Y-O bonds are more stable than Hf-O and Ge-O bonds. In fact, adding stable Y-O bonds into GeO<sub>x</sub> benefits to improve thermal stability of IL and its quality [2]. However, the influence of YGO IL for HfO<sub>2</sub>-based/TiN gate stack should be discussed in detail. In this paper, we demonstrated various processing conditions of incorporating Y into GeO<sub>x</sub> ILs and presented a series of material analyses to deeply realize the material interaction as various processes.

# 2. Device Preparation

Fig. 1 shows the process flow and structures of p-Ge capacitors for various ILs processes of Y incorporation. Standard IL process used oxygen plasma oxidation (PO) for GeO<sub>x</sub> formation [3]. Y-first/PO IL was formed by depositing a 0.6 nm Y metal followed by PO. PO/Y-later IL was formed by depositing a 0.6 nm Y after PO. High- $\kappa$  layers were grown by PEALD and TiN metal was deposited by sputtering Ti with N<sub>2</sub> plasma.

# 3. Results and Discussion

In Fig. 2, equivalent oxide thickness (EOT) and interface trap density ( $D_{it}$ ) of various ILs are shown. We found Y-first/PO case possessed lower  $D_{it}$  (4.9 × 10<sup>11</sup> eV<sup>-1</sup>cm<sup>-2</sup>) and higher EOT (1.9 nm) values than those in PO/Y-later case. Surprisingly, the PO/Y-later case showed the highest  $D_{it}$  value. Fig. 3 shows the corresponding TEM images and

EDX depth profiles. In fact, the IL of Y-first/PO case was only slightly thicker than the others. However, the EOT value was significantly degraded. The extracted dielectric constant ( $\kappa$ ) of the HfO<sub>2</sub>-based layers of Y-first/PO case was 10 only. For further investigating the difference between Y-first/PO and PO/Y-later cases, a simple test structures with various ILs under a 1 nm HfO<sub>2</sub> layer through a PDA of 500 °C were used for XPS analysis (Fig. 4(a)). Corresponding XPS spectra of O 1s and oxide composition are shown in Fig. 4. XPS spectra of Hf 4f and O/Hf atomic ratio are shown in Fig. 5. We found the peak of Hf 4f of Y-first/PO case was located at lower binding energy, which means the amount of Hf-O bonds in the HfO2 of Y-first/PO case was less. We, thus, concluded that Y-first/PO case had more Y-O bonds than PO/Y-later case, however, which had a lower O/Hf atomic ratio  $\sim$  1.403. This means pre-depositing Y followed by PO can form more Y-O bonds, nevertheless, the formation of stable Y-O bonds in the GeO<sub>x</sub> layer might rob the O atoms from upper HfO<sub>2</sub> layer. Next, we transferred the test structures through a N2 plasma treatment of TiN deposition. The XPS results of Hf 4f are shown in Fig. 6, indicating the Y-first/PO case had more Hf-N bonds formed by N<sub>2</sub> plasma treatment. This phenomenon was consistent with our EDX results, which showed N pile-up near IL. Postulated material interaction models were proposed in Fig. 7. Pre-depositing Y before IL oxidation resulted in more Y-O bonds in IL, and then, if Y atoms were not fully oxidized, they would rob the nearby O atoms from the HfO<sub>2</sub> layer and induce oxygen vacancies in HfO2. The oxygen vacancies would be filled by N atoms to form Hf-N bonds after N<sub>2</sub> plasma treatment. To further confirm the relation between Hf-N bonds and  $\kappa$  degradation, a pure Ti metal was used to substitute TiN metal of capacitors. Actually, we did observe the  $\kappa$  degradation of HfO<sub>2</sub> could be significantly improved.

# 3. Conclusions

In summary, the influence of incorporating Y into IL with HfO<sub>2</sub>-based/TiN gate stack was investigated. Doping Y into IL could improve interface quality; however, it also induced oxygen vacancies in HfO<sub>2</sub> and indirectly formed Hf-N bonds as TiN deposition, which degraded the  $\kappa$  of HfO<sub>2</sub>. If incorporating Y after the GeO<sub>x</sub> formation, non-fully oxidized Y atoms would rob the O atoms of IL and further degrade the interface quality.

### References

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Table I. Free Gibbs Energy values of thermal dynamic for conventional oxides.

Oxide	Gibbs Free Energy (kJ/mol)
GeO <sub>2</sub>	-497.06
HfO <sub>2</sub>	-1027.17
Y <sub>2</sub> O <sub>3</sub>	-1905.31

Cyclic DHF clean of p-Ge (100) bulk







Fig. 3. TEM images and EDX depth profiles for (a) Standard, (b) PO/Y-later and (c) Y-first/PO cases. Especially, N pile-up could be found in Y incorporation cases. The extracted  $\kappa$  of the HfO<sub>2</sub>/AlO<sub>x</sub>/HfO<sub>2</sub> stacks of standard and Y-first/PO cases were 15 and 10, respectively.



Fig. 5. XPS spectra of Hf 4f for Standard, PO/Y-later and Y-first/PO cases. Corresponding O/Hf atomic ratio was extracted by XPS spectra of Hf 4f and O 1s. Y-first/PO case had less Hf-O bonds and lowest O/Hf atomic ratio ~ 1.4 (O-less HfO<sub>2</sub>).



Fig. 2. Corresponding  $D_{it}$  and EOT values for various IL cases. Y-first/PO method could suppress  $D_{it}$  value; however, PO/Y-later method presented the highest  $D_{it}$  value.



Fig. 4. XPS spectra of O *1s* for (a) Standard, (b) PO/Y-later and (c) Y-first/PO cases. (d) Oxide composition for various test structures. Y-first/PO case had more Y-O bonds than PO/Y-later case. Standard  $+ N_2$  plasma



Fig. 6. Corresponding XPS spectra of Hf 4f for (a) Standard and (c) Y-first/PO case, (b) Standard and (d) Y-first/PO case through  $N_2$  plasma treatment. The Y-first/PO case had more Hf-N bonds formed by  $N_2$  plasma treatment. It means more oxygen vacancies were filled by N atoms in Y-first/PO case.



Fig. 7. Cartoons of the postulated sequential material interactions for Y-first/PO and PO/Y-later cases. (a) Pre-depositing Y before plasma oxidation was prone to make Y atoms located in IL. After HfO<sub>2</sub> deposition and PDA processes, non-fully oxidize Y atoms might rob the O atoms of upper HfO<sub>2</sub> and form oxygen vacancies in HfO<sub>2</sub>. During a N<sub>2</sub> plasma treatment, Hf-N bonds were formed in HfO<sub>2</sub>. (b) In the process of Y deposition after plasma oxidation, Y atoms oppositely located in HfO<sub>2</sub> layer. Hence, through a PDA process, Y atoms might rob the O atoms of under GeO<sub>x</sub> layer. This mechanism would degrade the IL quality, even if through a N<sub>2</sub> plasma treatment.