

Grain-Boundary-Limited Polycrystalline-Silicon Mobility with Negative Temperature Dependence ~ Modeling of Carrier Conduction through Grain-Boundary Traps Based on Trap-Assisted Tunneling ~

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Abstract

Based on the thermionic emission (TE) theory, the so-called Seto's model explains that the grain-boundary(GB)-limited carrier mobility in polycrystalline-silicon (poly-Si) is dependent positively on a temperature. Although this model is widely accepted as a standard model of the GB-limited mobility in poly-Si, many experiments support extremely-weak or negative temperature dependence. In this report, we have formulated carrier conduction through GB traps by utilizing the so-called trap-assisted tunneling (TAT) model based on the non-radiative multi-phonon transition theory. Self-consistent calculation has revealed that in contrast to the Seto's model, our novel model can reproduce the GB-limited mobility dependent negatively on a temperature. From the viewpoint of consistency with experiments, our novel model seems more appropriate than the Seto's model.

1. Introduction

The Seto's model [1], which is based on the TE theory, has been widely accepted as a standard model of the GB-limited carrier mobility in poly-Si [2, 3]. It has been already known that because a GB barrier is not very high in poly-Si films, the quantum-mechanical tunneling through the GB barrier is not significant compared with TE [4, 5]. The most essential consequence of the Seto's model is that the GB-limited mobility is temperature-activated and dependent positively on a temperature. Therefore, when the measured mobility shows extremely-weak or negative temperature dependence, we do not usually consider the mobility limited by GBs but that controlled by in-grain or MOS-interface defects instead [3]. We note that a low GB barrier energy less than the thermal energy (kT), which can reproduce the GB-limited mobility dependent negatively on a temperature [1], tends to overestimate a value of that mobility (see Sec. 3).

The circumstances explained above seem to suggest that the GB-limited mobility proportional to a grain size cannot be interpreted consistently with negative temperature dependence. This inconsistency will be, however, dissolved if we have developed a *novel* GB-limited mobility model that can reproduce negative temperature dependence. In this report, we formulate carrier conduction through GB traps by utilizing the TAT model [6 - 11], and then reveal that our model reproduces extremely-weak temperature dependence of the GB-limited mobility.

2. Grain-Boundary Modeling Based on Trap-Assisted Tunneling

In this section, carrier conduction through GB traps is formulated by utilizing the TAT model. In general, each localized electronic state is stabilized by the self-trapping mechanism, namely by local polarization due to lattice displacement [12]. Supposing the lattice displacement, the TAT model considers non-radiative multi-phonon transition [12]. Examples to which the TAT model has been applied extend over various kinds of simulation including SILC [6, 7] and RTN [8 - 10] as well as MONOS [11].

According to Refs. [1, 2], trapped charges on a GB have been expected to induce a parabolic potential barrier. Therefore, we can expect that the TAT model for GB traps is derived by replacing a trapezoidal potential barrier usually assumed in the TAT model for SILC [6, 7] with a parabolic one (see Fig. 1). A potential barrier induced by trapped charges on a GB (Q_t [1/cm²]) is schematically shown in Fig. 2. Figure 2 also shows equations for lengths and heights of the potential barrier, as well as an equation for an applied bias (V_a [V]) between anode and cathode sides of the potential barrier. All equations shown in Fig. 2 have been derived by using the literatures [1, 2] as reference. We note that a grain size of poly-Si is assumed to be larger than a sum of barrier lengths for both cathode and anode sides, namely Q_t/N where N [1/cm³] is a volume concentration

of free carriers (see Sec. 3). In Fig. 3, energies of trapped carriers on a GB and of de-trapped carriers are shown on a parabolic energy profile of the conduction band edge for a cathode or anode side ($E_C(x)$ [eV]). An energy difference between de-trapped and trapped carriers is attributed to the multi-phonon emission or absorption ($p\hbar\omega$ [eV]) during carrier-trapping. p is a number of phonons emitted during carrier-trapping (negative p indicates phonon absorption). $\hbar\omega$ [eV] is a single phonon energy typical for displaced lattice-vibration-modes.

We have formulated TAT rates for capture and emission of carriers, based on following two concepts: A short-range interaction [9] for the transition matrix element of carriers, originally proposed by Lundstrom and Svensson [13]; simplified statistics [7, 8, 10] for energies of tunneling carriers, originally introduced by Herrmann and Schenk [6]. Assuming large grains, those rates have been expressed as

$$\begin{aligned} \tau_{c,y}^{-1} &= \frac{2\pi}{\hbar} \cdot 8\pi\Phi_t^2 r_t^3 \cdot \frac{\sqrt{2}}{\pi^2 \hbar^3} m_{dos}^{3/2} \sum_p^{E \geq 0} \left[\sqrt{E} \cdot f_{FD}(E, E_F, kT) \right. \\ &\quad \left. \cdot T_{WKB}(x_1, x_2) \cdot P_{MP}(p, S, \hbar\omega, kT) \right], \\ \tau_{e,y}^{-1} &= \frac{2\pi}{\hbar} \cdot 8\pi\Phi_t^2 r_t^3 \cdot \frac{\sqrt{2}}{\pi^2 \hbar^3} m_{dos}^{3/2} \sum_p^{E \geq 0} \left[\sqrt{E} \cdot [1 - f_{FD}(E, E_F, kT)] \right. \\ &\quad \left. \cdot T_{WKB}(x_1, x_2) \cdot P_{MP}(-p, S, \hbar\omega, kT) \right], \end{aligned} \quad (1)$$

$$\left[r_t = \hbar / \sqrt{2m_{nm}\Phi_t}, \quad E = E_C(x_1) \right]$$

where "y" is "c" (cathode) or "a" (anode) and τ_{cy}^{-1} and τ_{ey}^{-1} [1/s] are capture and emission rates of carriers by GB traps, respectively. r_t is a localization length of GB traps, m_{dos} is the density-of-states effective-mass for a poly-Si grain, $f_{FD}()$ is the Fermi-Dirac distribution function, and E_F is the Fermi level of carriers in a grain whose concentration is N . $P_{MP}()$ is the multi-phonon transition probability under the Condon approximation (see Eq. (6.66) of Ref. [12]), S is the Huang-Rhys factor (see Sec. 6.1 of Ref. [12]), and m_{nm} is the tunneling mass of carriers in a grain. See Fig. 3 for parameters such as Φ_t , x_1 , and x_2 .

According to Eqs. (6 - 9) of Ref. [5], the tunneling probability of carriers across a parabolic GB barrier, denoted by $T_{WKB}()$ in Eqs. (1), has been expressed as

$$T_{WKB} = \exp \left[-\sqrt{\frac{q^2 m_{nm} N}{\hbar^2 \epsilon}} \left[x_2 \sqrt{x_2^2 - x_1^2} - x_1^2 \cdot \ln \left((x_2 + \sqrt{x_2^2 - x_1^2}) / x_1 \right) \right] \right]. \quad (2)$$

Using Eqs. (8) and (15) of the literature [7] as reference, a carrier occupancy of GB traps (f_t) and a TAT current density through GB traps (J_{TAT} [C/cm²/s]) have been expressed at a steady-state as

$$f_t = \frac{\tau_{c,c}^{-1} + \tau_{c,a}^{-1}}{\tau_{c,c}^{-1} + \tau_{c,a}^{-1} + \tau_{e,c}^{-1} + \tau_{e,a}^{-1}}, \quad J_{TAT} = qN_t \left[(1 - f_t) \tau_{c,c}^{-1} - f_t \tau_{e,c}^{-1} \right], \quad (3)$$

where N_t [1/cm²] is a surface concentration of GB traps.

3. Self-consistent Calculation and Results

Using equations shown in Figs. 2 and 3, $E_C(x)$ for both cathode and anode sides can be derived from Q_t . On the other hand, $Q_t (= f_t N_t)$ can be calculated from $E_C(x)$ for the both sides, according to Eqs. (1), (2), and (3). In order to self-consistently carry out those derivation and calculation, we have developed the EXCEL program based on a scheme shown in Fig. 4.

We then tried to confirm that the above program reproduces experimental temperature dependence of the electron mobility in poly-Si shown in Ref. [3]. Parameters used in self-consistent calculation are summarized in Table I. Reference [3] has reported that a median size of poly-Si grains is approximately 0.8 μm . We note that according to preceding calculation for the 0.8 μm grain size, a low GB potential barrier

less than kT/q yields the unphysical GB-limited mobility larger than 2000 $\text{cm}^2/\text{V}\cdot\text{s}$. Figure 5 shows calculation results of temperature dependence of the electron mobility for total currents including both TAT and TE. Calculation results for a TE current only are also shown in Fig. 5 as reference. In contrast to strong positive temperature dependence of TE, the total electron mobility dominated by TAT shows extremely-weak negative temperature dependence, which has been attributed to the inverted region effect of the so-called Marcus theory [14]. In Fig. 5, N is supposed to be $2 \times 10^{18} \text{ [1/cm}^3\text{]}$. If an induced channel thickness [2] is assumed to be 5 nm, the corresponding surface concentration ($N_s \text{ [1/cm}^2\text{]}$) is estimated to be $1 \times 10^{12} \text{ [1/cm}^2\text{]}$. One can see that extremely-weak temperature dependence of the experimental electron mobility about 50 $\text{cm}^2/\text{V}\cdot\text{s}$, which has been shown for $N_s = 1 \times 10^{12} \text{ [1/cm}^2\text{]}$ in Ref. [3], is comparable with our calculation results shown for $N = 2 \times 10^{18} \text{ [1/cm}^3\text{]}$ in Fig. 5. We note that the $0.8 \mu\text{m}$ grain size is larger than the maximum Q_t/N , namely 30 nm.

4. Conclusions

Utilizing the TAT model based on the non-radiative multi-phonon transition theory, we have formulated carrier conduction through GB traps. Self-consistent calculation has revealed that in contrast to the conventional Seto's model based on the TE theory, our novel model can reproduce the GB-limited mobility dependent negatively on a temperature. From the viewpoint of consistency with experiments, our novel model seems more appropriate than the Seto's model.

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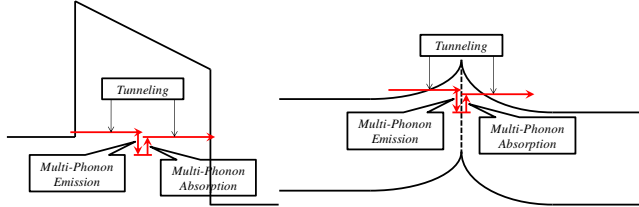


Fig. 1. A trapezoidal potential barrier is usually assumed in the TAT model for SILC [6, 7], as schematically shown in the left side. On the other hand, a parabolic potential barrier is expected for TAT carrier conduction through GB traps, as shown in the right side.

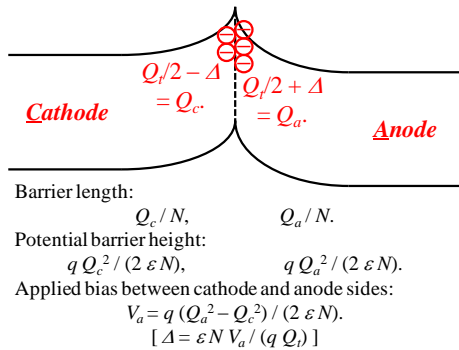


Fig. 2. Energy-band-bending due to trapped charges on a GB ($Q_t \text{ [1/cm}^2\text{]}$) is schematically shown in the upper figure. Below that figure, we also list equations concerning barriers of both cathode and anode sides. Q_c and $Q_a \text{ [1/cm}^2\text{]}$ are surface concentrations of GB-trapped-charges that

should be assigned to cathode and anode sides, respectively. $q \text{ [C]}$ and $\epsilon \text{ [C/V}\cdot\text{cm]}$ are the elementary electronic charge and the permittivity in poly-Si, respectively. A difference between Q_a and Q_c , namely $2\Delta \text{ [1/cm}^2\text{]}$, is related to V_a .

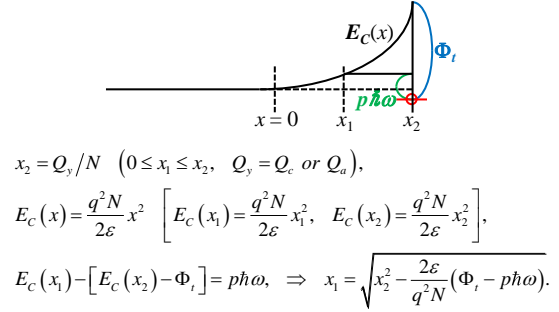


Fig. 3. The upper figure shows $E_c(x)$ for a cathode or anode side as a parabolic function of x . $x_1 \text{ [cm]}$ is a position of a classical turning point for de-trapped carriers. $x_2 \text{ [cm]}$ is a position of a GB, and also a barrier length. $\Phi_t \text{ [eV]}$ is the trap depth. See Fig. 2 for $Q_y = Q_c$ or Q_a . x_1 is related to x_2 and $p\hbar\omega$.

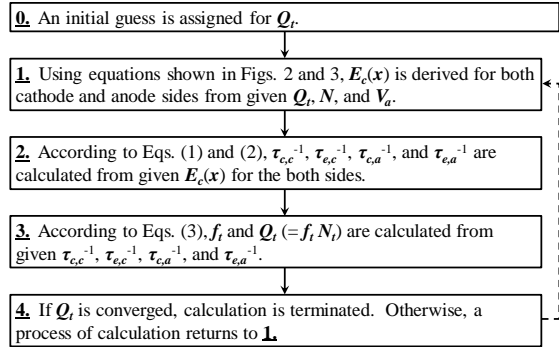


Fig. 4. A scheme is shown for self-consistent calculation of TAT through GB traps.

Trap-Type 1: $\hbar\omega = 0.02 \text{ eV}$, $\Phi_t = 0.315 \text{ eV}$, $S = 1$, $N_t = 2 \times 10^{12} \text{ cm}^{-2}$.
Trap-Type 2: $\hbar\omega = 0.02 \text{ eV}$, $\Phi_t = 0.400 \text{ eV}$, $S = 2$, $N_t = 2 \times 10^{12} \text{ cm}^{-2}$.
Trap-Type 3: $\Phi_t \sim 0.5 \text{ eV}$, $N_{\text{fixed-charges}} = 2 \times 10^{12} \text{ cm}^{-2}$.
$\epsilon = 11.7 \times \epsilon_0 \text{ C/V}\cdot\text{cm}$, $m_{\text{dos}} = 6^{2/3} \times 0.33 m_0$, $m_{\text{nan}} = 0.08 m_0$.

Table I. Parameters used in self-consistent calculation are listed. We have assumed three types of GB traps including deep levels, namely “Trap-Type 3”. Preceding calculation has ensured that “Trap-Type 3” behaves like fixed charges.

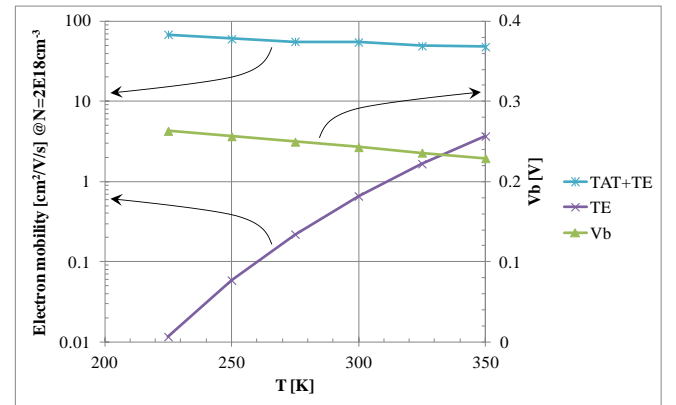


Fig. 5. “TAT+TE” and “TE” indicate the GB-limited electron mobility for total currents and that for a TE current only, respectively. $V_b \text{ [V]}$, namely a GB potential barrier height measured at a cathode side, is about 10 times larger than kT/q .