

## Importance of MoS<sub>2</sub>-Compound Sputtering even with Sulfur-Vapor Anneal for Chip-Size Fabrication

Shinya Imai, Takuya Hamada, Masaya Hamada, Takanori Shirokura, Iriya Muneta, Kuniyuki Kakushima, Tetsuya Tatsumi, Shigetaka Tomiya, Kazuo Tsutsui and Hitoshi Wakabayashi

Tokyo Institute of Technology,  
4259 Nagatsuta-cho, Midori-ku, Yokohama 226-8502, Japan, Phone: +81-045-924-5847, E-mail: imai.s.ah@m.titech.ac.jp

### Abstract

High quality MoS<sub>2</sub> film was synthesized by RF magnetron sputtering and sulfur-vapor annealing. The crystallinity of MoS<sub>2</sub> film was controlled by sputtering conditions. And we revealed a relationship of the crystallinity between as-sputtered and sulfur-annealed MoS<sub>2</sub> films. It is found that a crystallinity improvement after the sputtering is mandatory to achieve the excellent quality of MoS<sub>2</sub> film after sulfur-vapor anneal.

### 1. Introduction

Molybdenum disulfide (MoS<sub>2</sub>) film is a promising material for human interface FET applications, because of its transparency, flexibility and high mobility (~ 380 cm<sup>2</sup>/V-s) even in atomic thickness [1,2]. The MoS<sub>2</sub> film with continuous large area and high quality is essential for the high performance of MoS<sub>2</sub> device. For these preconditions, several research groups have synthesized the CVD-MoS<sub>2</sub> film using perylene-3,4,9,10 tetracarboxylic acid tetrapotassium salt (PTAS) to promote lateral growth, which influences an intrinsic MoS<sub>2</sub> properties [3]. On the other hand, a sputtered-MoS<sub>2</sub> film is superior to others in terms of uniformity, chip-size deposition and high thickness controllability. It has been reported that MoS<sub>2</sub> film was formed by using Mo- or MoS<sub>2</sub>-target sputter and sulfurization [4-6]. It is easily speculated that the MoS<sub>2</sub> target sputter is better than Mo one, because MoS<sub>2</sub> one leads higher quality of final MoS<sub>2</sub> film through metastable state of just-after-sputtered MoS<sub>2</sub> film. Although sputtered-MoS<sub>2</sub> film has been investigated by controlling the sputter parameters [7,8], the high quality of MoS<sub>2</sub> film in metastable state is expected as high as possible to obtain the fine final MoS<sub>2</sub> film.

In this work, we investigate the further improvement of MoS<sub>2</sub> film quality with discussing on the film crystallinity before and after sulfurization.

### 2. Experiments

MoS<sub>2</sub> films were formed by RF magnetron sputtering with 4N-MoS<sub>2</sub> compound target on SiO<sub>2</sub>/n-Si substrate, as shown in Fig. 1 (a). In order to prepare various MoS<sub>2</sub> films, the sputtering conditions were changed as RF power of 30 to 50 W, substrate temperature of 200 to 500°C under Ar pressure of 0.55 Pa, Ar flow rate of 7 sccm and target-substrate distance of 150 mm. Figure 1 (b) shows a sulfur vapor annealing apparatus for sulfur compensation to MoS<sub>2</sub> film. The sulfur powder was placed at the zone1 heated at 250°C for 40 min, and the samples were placed at zone2 heated at 700°C for 40 min. The MoS<sub>2</sub> films were analyzed by the Raman

spectroscopy for which the full width at half maximum (FWHM) values of  $A_{1g}$  and  $E_{2g}^{\prime}$  peaks were extracted.

### 3. Results and Discussion

Figure 2 shows  $A_{1g}$  and  $E_{2g}^{\prime}$  FWHM values of MoS<sub>2</sub> films varying RF power. The crystallinity improves with an increase in RF power up to 40 W, in which particle's energy enhances migration on substrate. On the other hand upper than 40 W, the degradation of crystallinity is seen, in which higher particle's energy and the large number of species cause sulfur defects and increment of nucleation density, respectively. This trend is also seen even after S-annealing.

Figure 3 shows FWHM values depending on substrate temperature. For as-sputtered MoS<sub>2</sub> film below 300°C, MoS<sub>2</sub> crystallinity was improved along with an increase in the substrate temperature. However, the as-sputtered MoS<sub>2</sub> crystallinity deteriorated with higher than 300°C. Figures 4 (a) and (b) show the X-ray photoelectron spectroscopy (XPS) of Mo 3d for the as-sputtered MoS<sub>2</sub> films at 500 and 300°C, respectively. Mo-Mo bonds are observed corresponding to sulfur defects due to an excessive sulfur desorption [7]. On the other hand at 300°C, there is no peak of Mo-Mo bonds and Mo-O bonds, which is expected to be sulfurized by following SVA process.

In Fig. 3 with S-annealed data, FWHM values depending on temperature are almost the same as as-sputtered data. The excessive sulfur desorption might influence crystal recovery during sulfur annealing. From the X-ray diffraction (XRD) shown in Fig 5, (002) peak of MoS<sub>2</sub> film are found and the crystallites of MoS<sub>2</sub> is improved after sulfur annealing.

Figure 6 shows the correlation between FWHM values before and after sulfur annealing, based on the data in Figs. 2 and 3. All MoS<sub>2</sub> films except 400 and 500°C are improved along with correlation coefficient of one. From this result, the improvement of as-sputtered film quality directly affects to the MoS<sub>2</sub> crystallinity after S-annealing, as shown in Fig. 7. Based on these discussions, a two-step process consisting of compound sputter using MoS<sub>2</sub> target and sulfur annealing is appropriated for the MoS<sub>2</sub> film improvement.

### 4. Conclusions

The crystallinity of the MoS<sub>2</sub> were controlled by sputtering conditions, and we demonstrated the mechanism of controlling the MoS<sub>2</sub> crystallinity in various sputtering conditions. The crystallinity of the sputtered MoS<sub>2</sub> film before S-annealing was retained after S-annealing. It was revealed an importance of improvement of compound sputtering even after sulfur compensation process. Further high performance of FET using thin MoS<sub>2</sub> film is expected to be applied for human

interface FET applications.

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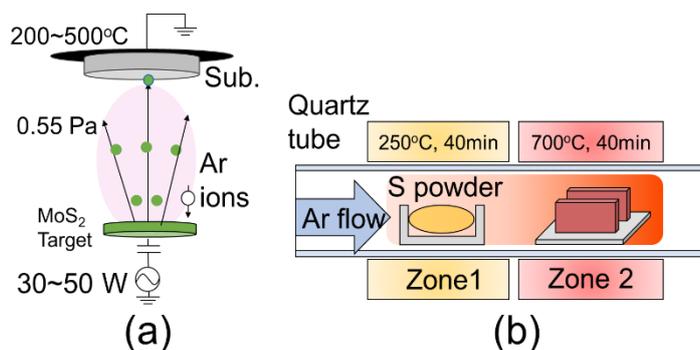


Fig. 1. Schematic diagrams of (a) RF sputtering system and (b) sulfur vapor annealing process tool.

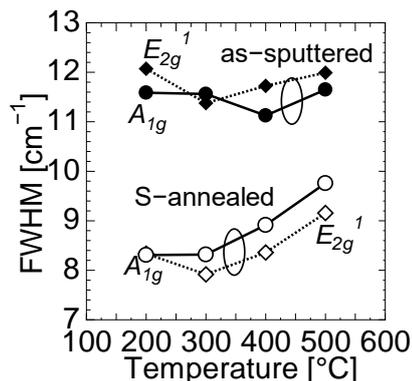


Fig. 3. FWHM values in  $E_{2g}^1$  and  $A_{1g}$  of the Raman spectra for as-sputtered and S-annealed MoS<sub>2</sub> films depending on substrate temperature.

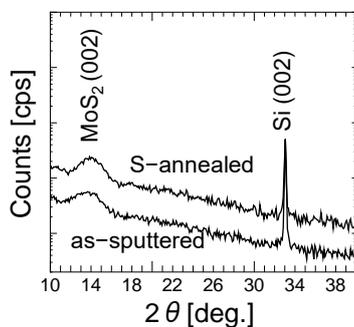


Fig. 5. XRD pattern of as-sputtered and S-annealed MoS<sub>2</sub> film. The films were sputtered with 40 W, 0.55 Pa at 300°C. Intensities have been normalized by silicon peak.

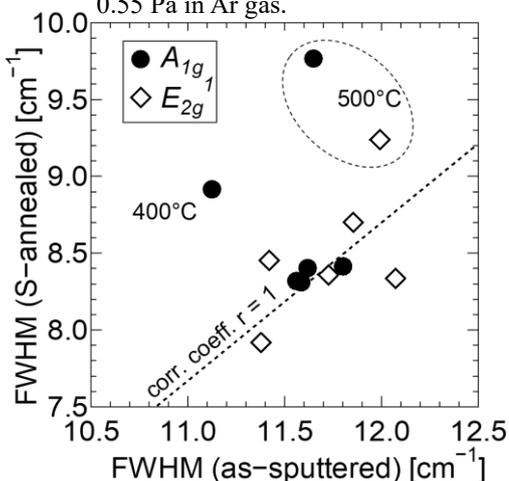


Fig. 6. Relationship of FWHM values in  $A_{1g}$  and  $E_{2g}^1$  modes between as-sputtered and S-annealed MoS<sub>2</sub> films.

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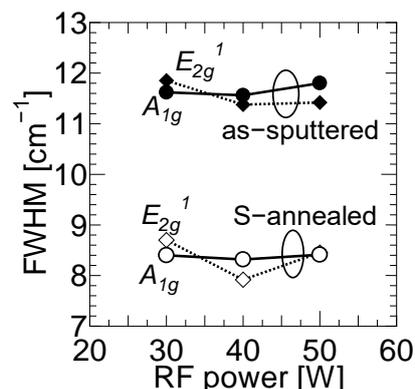


Fig. 2. FWHM values in  $E_{2g}^1$  and  $A_{1g}$  modes from the Raman spectra for as-sputtered and S-annealed MoS<sub>2</sub> films depending on RF power, 0.55 Pa, 300°C.

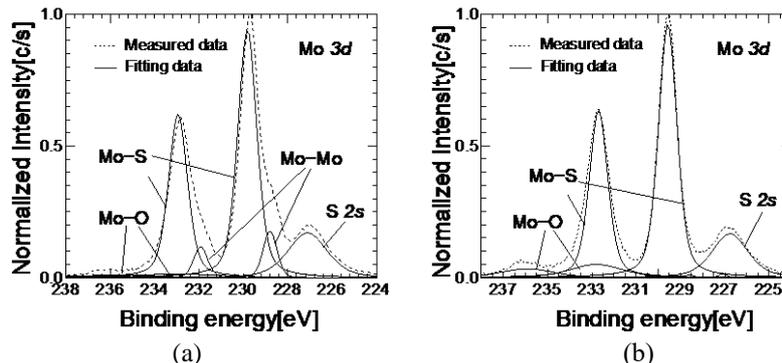


Fig. 4. XPS spectra of molybdenum 3d in as-sputtered MoS<sub>2</sub> film with the substrate temperature of (a) 500°C and (b) 300°C, RF power of 40 W and 0.55 Pa in Ar gas.

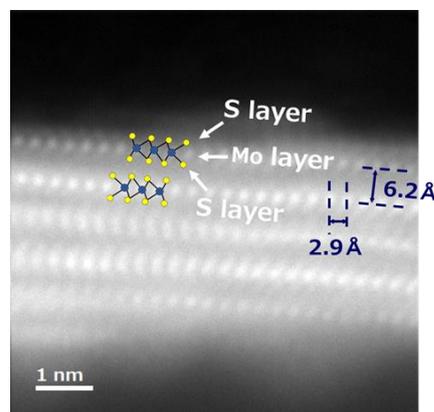


Fig. 7. Cross-sectional HAADF-STEM image of sputtered-MoS<sub>2</sub> film with 40 W at 300°C after sulfur annealing.