

# Antiferroelectricity and cycling behavior of ALD ZrO<sub>2</sub> ultra-thin films

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## Abstract

Antiferroelectricity (AFE) in ZrO<sub>2</sub> thin films is systematically investigated with different film thicknesses and PMA temperatures. Excellent AFE characteristics are observed for the ZrO<sub>2</sub> thickness from 5.3 to 9.5 nm. It is found that higher PMA temperature can provide better AFE properties in the thick ZrO<sub>2</sub> films, while the optimum temperature becomes lower in the thin films. In addition, the AFE ZrO<sub>2</sub> films are found to show a significant wake-up effect together with a reduction of the critical transition electric field during gate pulse cycling.

## 1. Introduction

ZrO<sub>2</sub> has been extensively studied as a high-k material in the past few years, while the discovery of the existence of antiferroelectricity (AFE) has given more choices on applications of this dielectric material [1]. Compared with its counterpart HfO<sub>2</sub> which has attracted much attention since the report of ferroelectricity (FE) in Si-doped ones [2], ZrO<sub>2</sub> possesses several superior properties such as the lower temperature crystallization, improved cycling endurance for non-volatile memory applications [3] and no need for dopants. Recently, scalability of AFE ZrO<sub>2</sub> has been proved in a wide thickness range (4.3-14.7 nm) [4], which strengthened the applicability to future CMOS devices. However, the AFE properties of ZrO<sub>2</sub> thin films including the effects of annealing and the cycling behaviors have less been studied so far.

In this work, we systematically examine factors that affect AFE in ZrO<sub>2</sub> thin films, including the film thickness and post metallization annealing (PMA) temperature. Besides, the cycling behavior of AFE ZrO<sub>2</sub> capacitors is studied with an emphasis on the wake-up effect of AFE in ultra-thin ZrO<sub>2</sub> films.

## 2. Experiment

The process flow is shown in Fig. 1. TiN/ZrO<sub>2</sub>/TiN structure capacitors were prepared with changing the ZrO<sub>2</sub> film thickness from 3.7 nm to 9.5 nm. 32 nm TiN was sputtered on a heavily-p-doped Si substrate as the bottom electrode, followed by atomic layer deposition (ALD) of pure ZrO<sub>2</sub>. Subsequently, 32 nm TiN was sputtered as the top electrode. PMA was then carried out at 400°C, 500°C and 600°C for 1 min in N<sub>2</sub> ambient.

## 3. Results and discussion

### Antiferroelectricity in ALD ZrO<sub>2</sub> films

AFE characteristics were observed in ZrO<sub>2</sub> films from 5.3 nm with different PMA temperatures. In the present study, 3.7 nm and 4.2 nm ZrO<sub>2</sub> films are quite leaky and, thus, the AFE cannot be evaluated. Fig. 2 shows P-E and I-E characteristics before and after gate pulse cycles of 10<sup>4</sup>. We have observed symmetric and clear AFE P-E curves and clear

switching current, compared with the results in the other recent report on AFE ZrO<sub>2</sub> [4]. Fig. 3 summarizes the thickness dependence of (a) maximum polarization (P<sub>max</sub>), (b) remnant polarization (2P<sub>r</sub>), (c) critical transition electric field (E<sub>AFE-FE</sub>, E<sub>FE-AFE</sub> and (d) the distance between E<sub>AFE-FE</sub> and E<sub>FE-AFE</sub> (ΔE<sub>c</sub>). The significant wake-up effect such as the increase of P<sub>max</sub> and 2P<sub>r</sub>, and the decrease of the critical transition electric field, which has not been pointed out yet, are observed after 10<sup>4</sup> cycles. Fig. 4 shows the amount of the hysteresis loop, ΔP, defined by the difference between the backward and forward polarization at a same electric field. The gate pulse cycle is found to open up the hysteresis loop in AFE.

Another finding in Fig. 2 is the different trends of the PMA temperature dependence of AFE characteristics for the different film thickness. For 5.3 nm and 6 nm, lower PMA temperature (400°C) is enough to obtain strong AFE, while increasing PMA temperature leads to leaky P-V curves. On the other hand, the PMA temperature dependence is different for 9.5 nm ZrO<sub>2</sub>, where the strongest AFE is obtained for 600°C PMA and the films with lower PMA temperature show leaky P-V. This trend is quite consistent with the DC I-V characteristics that the gate current in 5.3 nm and 6 nm ZrO<sub>2</sub> increases with increasing PMA temperature, while that in the 9.5 nm one decreases in a relatively high electric field region. Thus, AFE characteristics were evaluated for the 5.3 and 6 nm ZrO<sub>2</sub> with 400°C PMA and 9.5 nm ZrO<sub>2</sub> with 600°C PMA. As shown in Fig. 4, ΔP increases with increasing the film thickness. On the other hand, the distance between E<sub>AFE-FE</sub> and E<sub>FE-AFE</sub> also increases with the film thickness.

### Cycling behavior of AFE ZrO<sub>2</sub> films

The cycling behavior and the wake-up effect in AFE ZrO<sub>2</sub> films have not been investigated yet so far, whereas it has been reported [5] that some doped-HfO<sub>2</sub> films need the wake-up process to obtain the ferroelectric properties from the pristine antiferroelectric-like characteristics. The cycling behavior is thus examined for the AFE ZrO<sub>2</sub> films in order to quantitatively evaluate the wake-up effect.

Fig. 5(a) and (b) show the cycling behaviors of P-E and switching I-E characteristics of 5.3 nm ZrO<sub>2</sub> films with 400°C PMA up to 10<sup>4</sup> cycles. The emergence of AFE and a significant wake-up effect are clearly observed. Fig. 5(c) shows the ΔP-E characteristics as a parameter of the cycle number. All the above results have proved the existence of the wake-up effect in ultra-thin AFE ZrO<sub>2</sub> films. On the other hand, the critical electric field at which the switching current has a peak is reduced with an increase in the cycle number, as seen in Fig. 5(d). This result suggests that the energy needed for the change from the tetragonal to the orthorhombic phase [6] can be smaller with increasing the cycles. The reduction of critical electric field is also observed in the other samples. Both

$E_{\text{AFE-FE}}$  and  $E_{\text{FE-AFE}}$  in the positive gate voltage are reduced after  $10^4$  cycles. This fact suggests that the crystalline structure change of AFE  $\text{ZrO}_2$  films might be easier during the cycling.

### 3. Conclusions

$\text{ZrO}_2$  films in the thickness range of 3.7-9.5 nm have been systematically studied from the viewpoint of AFE with an emphasis on the PMA condition. Clear AFE characteristics were observed for the thickness from 5.3 to 9.5 nm. The thinner  $\text{ZrO}_2$  films show the AFE properties with low leakage current under lower thermal budget, while the thicker film tends to show a wider hysteresis loop under higher thermal budget. The strong wake-up effect has been observed in AFE  $\text{ZrO}_2$  films together with a reduction of the critical transition electric field during cycling.

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**References** [1] J. Muller et al., Nano Lett. **12**, 4318 (2012).

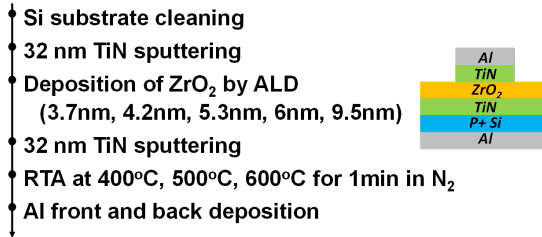
[2] T. S. Boscke et al., Appl. Phys. Lett. **99**, 102903 (2011).

[3] M. Pesic et al., Adv. Funct. Mater. **26**, 7486 (2016).

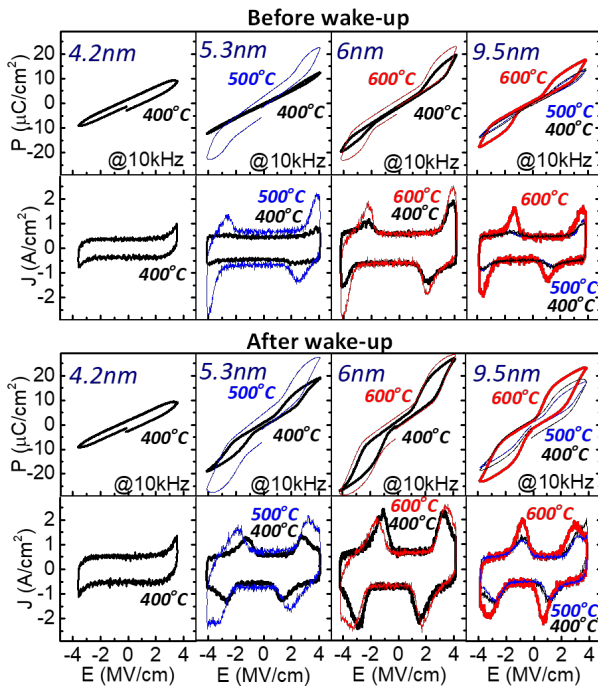
[4] P. D. Lomenzo et al., IEEE International Memory Workshop (IMW), 5.2., (2020).

[5] M. Hoffmann et al., J. Appl. Phys. **118**, 072006 (2015).

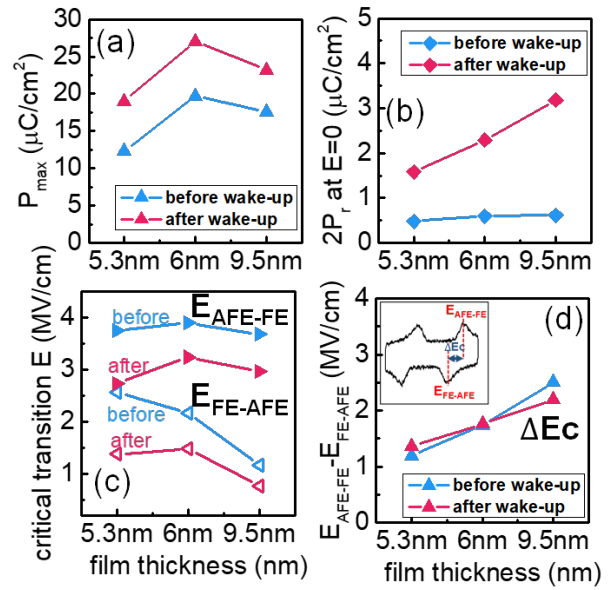
[6] S. E. Reyes-Lillo et al., Phys. Rev. B **90**, 140103 (2014).



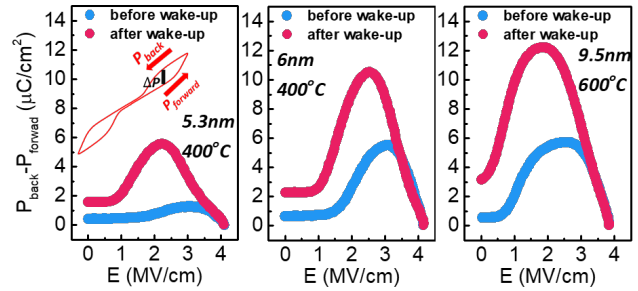
**Fig. 1** Fabrication process flow.



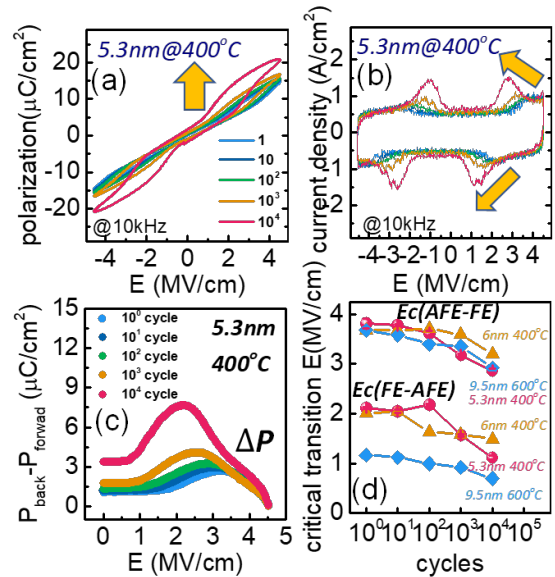
**Fig. 2** P-V hysteresis loops and switching current characteristics for TiN/ $\text{ZrO}_2$ /TiN capacitors with different thicknesses measured before and after  $10^4$  gate pulse cycles for wake-up.



**Fig. 3** Thickness dependence of (a) maximum polarization  $P_{\text{max}}$ , (b) remnant polarization  $2P_r$ , (c) critical transition  $E$  and (d) the distance between  $E_{\text{AFE-FE}}$  and  $E_{\text{FE-AFE}}$ .



**Fig. 4** The difference ( $\Delta P$ ) between backward polarization and forward polarization as a function of electric field for  $\text{ZrO}_2$  films with different thicknesses.



**Fig. 5** Cycling behaviors of (a) P-V characteristics, (b) switching current characteristics, (c)  $\Delta P$  for 5.3 nm  $\text{ZrO}_2$  with 400 °C PMA. (d) critical transition  $E$  changed during cycling for  $\text{ZrO}_2$  films with different thicknesses.